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# Effects of nano-additive TiO<sub>2</sub> on performance of micro-arc oxidation coatings formed on 6063 aluminum alloy

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**Abstract:** Ceramic coatings were fabricated on aluminum alloy substrates by micro-arc oxidation (MAO) in silicate electrolytes doped with different concentrations of  $TiO_2$  nano-additive. Effects of nano-additive concentration on the structural and mechanical properties of the MAO coatings were analyzed. The results revealed that some nano-particle were incorporated into the resulting coating during the MAO process, while there was a reasonable concentration for the  $TiO_2$  nano-additive. With increasing the nano-additive concentration to 3.2 g/L, the adhesion value increased, while mean friction coefficient and mass loss decreased. A further increase of nano-additive deteriorated the adhesion and mean friction coefficient values, which was consistent with the micro-hardness tests.

Key words: Al alloy; micro-arc oxidation; nano-additive; micro-hardness; friction and wear

# **1** Introduction

Micro-arc oxidation (MAO) has attracted much interest as an effective technique to improve the hardness and wear resistance of aluminum alloys by forming a relatively thick and hard ceramic coating on aluminum alloys through plasma discharging under high voltage [1-3]. Thus, the technique shows a wide prospect for its application [4,5]. However, the ceramic coatings generally possess a foam-like structure with high bulk porosity and relatively poor mechanical properties [6,7], which restrict them from even wider technical applications [8]. Up to now, researches mainly focused on the effects of the processing parameters [9-12], such as current density [13]; however, nano-additive doping in the electrolyte also had great influence on the properties of the ceramic coatings [14,15]. JIN et al [8] found that the MAO Al<sub>2</sub>O<sub>3</sub> coatings with Fe micro-grains were much denser and harder, and the wear resistance was also improved significantly since ceramic coatings became denser with a lower population of the micro-pores [8]. SHI et al [16] fabricated coatings in Na<sub>2</sub>SiO<sub>3</sub>-Na<sub>3</sub>PO<sub>4</sub> solution system by doping two kinds of additives (Na<sub>2</sub>B<sub>4</sub>O<sub>7</sub> and EDTA). The results revealed that doping of additives had little effect on the elemental composition, while it influenced the morphological feature of the coating. However, few data can be available concerning on the effect of nano-additive concentrations on the properties of the prepared MAO coatings.

In this work, in order to improve the structural and mechanical properties of ceramic coatings prepared by MAO method, a new electrolyte based on silicate solution with different concentrations of  $TiO_2$  nano-additive was used to fabricate ceramic coatings on 6063 alloy. Effects of different nano-additive concentrations on the structural and mechanical properties of the MAO coatings were analyzed.

## **2** Experimental

The substrate materials selected for this study were commercial 6063 aluminum alloys (0.45%-0.90% Mg, 0.2%-0.6% Si, 0.35% Fe, 0.10% Cu, 0.10% Mn, 0.10% Cr, 0.10% Zn, 0.10% Ti and balance Al in mass fraction). The samples with dimensions of 30 mm×25 mm×3 mm were polished with abrasive paper and degreased with acetone followed by rinsing with distilled water before coating formation. The electrolyte was a dilute aqueous

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solution of 1 g/L KOH and 10 g/L Na<sub>2</sub>SiO<sub>3</sub>. Rutile TiO<sub>2</sub> nano-additive with size about 10 nm was dispersed evenly in the solution and the concentration of nano-additives was in a range of 0.8-4.0 g/L. The electrolyte was agitated with a mechanical stirrer. A fivelitter bath from the stainless steel was used. The specimen was served as the anode and the bath wall was served as the contrary electrode. The system was cooled by cold water pumped through double walls of the bath. The electrolyte temperature was controlled at 25-30 °C throughout the process. The power supply was pulsed DC MAO equipment (DSM30IM) with the maximum voltage amplitude of 1000 V. During the MAO process, the current density was fixed at 15 A/dm<sup>2</sup> by controlling the voltage amplitude and the pulse width was 5000 µs. Time of coating formation was 1 h according to our experimental experience, under which the prepared coatings had optimal properties. After the treatment, the samples were rinsed in distilled water and dried in air.

The crystallographic characteristics of the coatings were investigated using a Thermo ARL X' TRA X-ray diffractometer (XRD) (Cu K<sub> $\alpha$ </sub> radiation) with step size of 0.04°. The X-ray generator settings were 45 kV and 40 mA, respectively. The surface morphology of the coatings was characterized by a Hitachi S-4700 scanning electron microscope (SEM). Energy dispersive X-ray spectroscopy (EDX) was used for qualitative elemental chemical analysis. The coating hardness was evaluated using a HMV-IT micro-hardness tester with Vickers indenter under a load of 1.96 N (200 g). The adhesion of the coatings to substrate was carried out on a conventional WS-2005 scratch tester at a constant linear velocity of 4 mm/min and a loading rate of 100 N/min. The corresponding critical load represented the adhesion to substrate for the coatings. The tribological properties of the coatings were performed on a WTM-2E ball-on-disk tribometer with a rotational speed of 336 r/min. The coating was served as the disk, and the counterpart was a Si<sub>3</sub>N<sub>4</sub> ceramic ball (4 mm in diameter, HV1550 in hardness). The tests were carried out at a normal load of 9.8 N with a wear track radius of 15 mm. The abrasion loss was measured after 1 h friction measurement using an electronic direct reading balance (LJBROR L-200, readability 0.01 mg).

# **3** Results and discussion

#### 3.1 Phase analysis

Figure 1 shows XRD patterns of the ceramic coatings prepared in silicate electrolyte with different concentrations of TiO<sub>2</sub> nano-additive. For the coating prepared in silicate solution without nano-additives, it can be seen that the prepared ceramic coatings consist of two crystal phases of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> and  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> in addition to

some diffraction peaks of 6063 alloy. It is similar for the coatings prepared in silicate solution with  $TiO_2$  nano-additive; however, the peak intensity at 46° with addition of  $TiO_2$  nano-powder is increased because of the appearance of  $TiO_2$  phase, which indicates that some nano-particles have entered into the prepared ceramic coatings. As well known, continuous and long time discharge at 15 A/dm<sup>2</sup> results in a high temperature in the discharge channels, which is propitious to the deposition of  $TiO_2$  nano-particles [15]. So  $TiO_2$  peaks at 46° can be detected in the coating. However, titanate compound was not formed during the process since no corresponding diffraction peaks were found.



**Fig. 1** XRD patterns of ceramic coatings prepared with different TiO<sub>2</sub> nano-additive concentrations: (a) 0; (b) 0.8 g/L; (c) 1.6 g/L; (d) 2.4 g/L; (e) 3.2 g/L; (f) 4.0 g/L

Generally, the in situ growth of the  $Al_2O_3$  film on the aluminum alloy by MAO technique includes the following processes.

Cathode reaction:

$$2H^{+}+2e \longrightarrow H_{2}\uparrow$$
 (1)

Anode reaction:

 $4OH^{-} \longrightarrow 2H_2O + O_2\uparrow + 4e \tag{2}$ 

$$Al \longrightarrow Al^{3+} + 3e \tag{3}$$

$$Al^{3+}+3OH^{-} \longrightarrow Al(OH)_{3}$$
(4)

$$2Al(OH)_3 \longrightarrow Al_2O_3 + 3H_2O \tag{5}$$

$$2\mathrm{Al}^{3+} + 3\mathrm{SiO}_3^{2-} \longrightarrow 3\mathrm{SiO}_2 + \mathrm{Al}_2\mathrm{O}_3 \tag{6}$$

$$2\mathrm{Al}^{3+}+3\mathrm{O}^{2-}\longrightarrow\mathrm{Al}_{2}\mathrm{O}_{3} \tag{7}$$

Due to the high temperature and upper voltage caused by discharge sparks,  $Al^{3+}$  from the substrate combines with OH<sup>-</sup> and O<sup>2-</sup> in the electrolyte and forms molten  $Al_2O_3$ . Simultaneously, the molten  $Al_2O_3$  erupts from the discharge channels and forms the ceramic coatings by quenching of electrolyte with a high cooling rate.

#### 3.2 SEM observation and EDX analysis

Figure 2 represents the morphology of the coatings prepared in silicate electrolyte with different TiO<sub>2</sub> nano-additive concentrations. It was found that the surface of the MAO coating was porous and loose (Fig. 2(a)), while it changed greatly for the coatings prepared with nano-additives in comparison with that without nano-additives. Compositions and structures of the prepared coatings combine both substrates and electrolytes during MAO process. With increasing the concentration of TiO<sub>2</sub> nano-additives from 0.8 g/L to 4 g/L, the coating surface became denser and smoother and the number of pores decreased. It can be concluded that addition of nano-powder plays an essential role in fabricating ceramic coatings with a lower porosity. For the composite oxide coatings, TiO<sub>2</sub> nano-particles could embed into the micro-arc discharge channels through the diffusion by the introduction of TiO<sub>2</sub> nano-additive in the electrolyte during MAO. When the TiO<sub>2</sub> nano powders were added into the electrolyte, they were homogeneously distributed under vigorous stirring. During MAO, when

the molten  $Al_2O_3$  erupted from the discharge channels, TiO<sub>2</sub> particles would be mixed within the  $Al_2O_3$  coating. We thought that more and more dispersed nano particles entered into the pores with increasing the nano-additive concentrations, so the coating surface became denser and smoother. The coating color was darkened from grey white to blue black. By this method, high quality coatings with high micro-hardness, strong adhesion and excellent wear resistance could be synthesized.

The EDX spectrum collected from polished crosssections of ceramic coatings with TiO<sub>2</sub> nano-additive concentration of 4 g/L shows the elemental composition and element concentration (Fig. 3). It can be seen that the prepared coating mainly consists of elements Al, O, Si and Ti, which confirms that some TiO<sub>2</sub> nano-additive has melted into the resulting coating. Figure 4 shows the major elemental concentrations of ceramic coatings prepared with different TiO<sub>2</sub> nano-additive concentrations. From Fig. 4, it can be seen that the content of O changed little, while the content of Al decreased and the content of Ti increased with increasing



**Fig. 2** SEM images of ceramic coatings with different  $TiO_2$  nano-additive concentrations: (a) 0; (b) 0.8 g/L; (c) 1.6 g/L; (d) 2.4 g/L; (e) 3.2 g/L; (f) 4.0 g/L



Fig. 3 EDX spectrum of ceramic coating with nano-additive concentration of 4 g/L  $\,$ 



Fig. 4 Elemental concentrations of ceramic coatings prepared with different  $TiO_2$  nano-additive concentrations

nano-additive concentration. The mass fraction of Ti was 10.29% for the coating prepared in silicate solution with nano-additive concentration of 4.0 g/L, which increased 2 times more than compared with that prepared with nano-additive concentration of 0.8 g/L.

#### 3.3 Micro-hardness tests

Micro-hardness tests were carried out on the samples prepared with different nano-additive concentrations, which are shown in Fig. 5. To improve the accuracy of the micro-hardness value, several-point hardness tests were utilized to get an average hardness value. For the coating prepared without nano-additive, the average micro-hardness value is about HV1356.5. It TiO<sub>2</sub> increases with increasing nano-additive concentration, and is about HV1620 for the coating prepared with nano-additive concentration of 3.2 g/L, which increased by 19% compared with that prepared without nano-additive. The value decreased to HV1585 when the nano-additive concentration was further increased to 4.0 g/L, which indicates that there is a

reasonable concentration of TiO<sub>2</sub> nano-additive for the silicate solution. From above analysis, we can see that each micro-hardness value of the ceramic coating prepared with nano-additive is higher than that without nano-additive. Such a considerable increase in hardness is attributed to the better crystallization and smoother surface morphology with fewer defects. However, when the nano-additive concentration was further increased to 4.0 g/L, the concentration of TiO<sub>2</sub> embedded in the coatings becomes higher. The decrease of hardness value may be caused by the too much TiO<sub>2</sub> nano-additive incorporation since the hardness of TiO<sub>2</sub> is lower than that of ceramic  $Al_2O_3$  coating.



**Fig. 5** Average value of micro-hardness of ceramic coatings prepared with different TiO<sub>2</sub> nano-additive concentrations

#### 3.4 Adhesion and friction tests

Adhesion values of the coatings prepared with different  $TiO_2$  nano-additive concentrations are shown in Fig. 6. In this work, the addition of nano-powder to electrolyte led to a remarkable improvement of adhesion values. Adhesion value increased with increasing the nano-additive concentrations, and reached the highest value for the coating prepared with nano-additive concentration of 3.2 g/L. A further increase of nano-additive concentration deteriorated the bond of the coating to aluminum substrate, which is consistent with the micro-hardness tests.

Figure 6 also shows mean friction coefficients with test duration for the coatings prepared with different nano-additive concentrations. The addition of nano-powder to electrolyte led to reduction of friction coefficient. The coating prepared with nano-additive concentration of 3.2 g/L exhibited the lowest mean friction coefficient of 0.673 and the value was more stable during the tests. The mass loss of the coatings prepared with different nano-additive concentrations was measured after 1 h fiction test (Fig. 6). The value was decreased to 0.09 mg for the ceramic coating prepared with nano-additive concentration of 3.2 g/L, which indicates excellent wear resistance. It is mainly attributed to its dense surface, high hardness and relatively low friction coefficient. However, no obvious change was found when the nano-additive concentration was increased to 4.0 g/L.



Fig. 6 Adhesion value, average friction coefficient and mass loss of ceramic coatings prepared with different  $TiO_2$  nano-additive concentrations

# **4** Conclusions

1) Ceramic coatings were synthesized on 6063 aluminum alloy substrates in silicate electrolyte with different  $TiO_2$  nano-additive concentrations by MAO technique.

2) XRD and EDX analyses showed that  $TiO_2$  nano-particles were melted into the prepared ceramic coatings. The surface morphology of the coatings with nano-additive changed greatly in comparison with that without nano-additive.

3) The average micro-hardness value and adhesion value of the ceramic coatings increased with increasing nano-additive concentration to 3.2 g/L, while the values decreased for the coating prepared with nano-additive concentration of 4.0 g/L. The lowest mean friction coefficient of 0.673 and excellent wear resistance were measured for the coating prepared with nano-additive concentration of 3.2 g/L, which was thought the reasonable concentration of TiO<sub>2</sub> nano-additive in the silicate solution with current density of 15 A/dm<sup>2</sup> and oxidation time of 1 h. The effects of other fabrication parameters during MAO process on the performance of TiO<sub>2</sub> doped Al<sub>2</sub>O<sub>3</sub> coatings will be discussed next.

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# TiO<sub>2</sub>纳米添加剂对 6063 铝合金 微弧氧化陶瓷涂层性能的影响

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**摘 要:**通过在硅酸盐电解液中加入 TiO<sub>2</sub>纳米添加剂,研究纳米添加剂浓度的变化对 6063 铝合金微弧氧化陶瓷 涂层性能的影响。结果表明,纳米添加剂进入到陶瓷涂层中,而添加剂浓度的选取有一个较合理的范围。随着纳 米添加剂的浓度增加到 3.2 g/L,涂层的结合力逐渐增大,平均摩擦因数和质量损耗逐渐减小。当浓度增加到 4.0 g/L 时,涂层的结合力减弱,而平均摩擦因数增加,这与涂层显微硬度的测试结果一致。 关键词:铝合金;微弧氧化;纳米添加剂;显微硬度;摩擦磨损

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