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## Electrochemical codeposition of Mg-Li-Gd alloys from LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub> melts

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**Abstract:** Mg-Li-Gd alloys were prepared by electrochemical codeposition from LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub> melts on molybdenum electrode with constant current density at 823 and 973 K. The microstructure of the Mg-Li-Gd alloys was analyzed by X-ray diffraction (XRD), optical microscopy (OM) and scanning electron microscopy (SEM). The results show that magnesium and gadolinium deposit mainly in the first 30 min, and the alloy obtained contains 96.53% Mg, 0.27% Li and 3.20% Gd (mass fraction). Then, the reduction of lithium ions occurs quickly. The composition of alloy can be adjusted by controlling electrolysis time or  $Gd_2O_3$  concentration in LiCl-KCl melts. With the addition of Gd into Mg-Li alloys, the corrosion resistance of the alloys is enhanced. XRD results suggest that Mg<sub>3</sub>Gd and Mg<sub>2</sub>Gd can be formed in Mg-Li-Gd alloys. The distribution of Gd element in Mg-Li-Gd alloys indicates that Gd element mainly distributes at the grain boundaries of Mg-Li-Gd alloys.

Key words: electrochemical codeposition; Mg-Li-Gd alloy; chloride melt; galvanostatic electrolysis; Gd<sub>2</sub>O<sub>3</sub>

## **1** Introduction

Mg-Li alloys are the lightest metallic materials with high specific strength, good machining property, good magnetic screen and shock resistance ability. They are widely used in the fields of aerospace, aircraft and weapons[1–3]. Because Li is a very active element, the stability of Mg-Li alloys is relatively poor[4–6].

To improve the mechanical properties and corrosion resistance of Mg-Li based alloys, rare earth elements are added; consequently, Mg-Li-RE alloys are widely studied[7–11]. The addition of rare earth elements can effectively increase the mechanical properties and high temperature performance of Mg-Li alloys, and can improve the corrosion resistance and the performance of creep. The effect of Gd on mechanical properties of Mg alloys was investigated by directly melting high purity magnesium and Mg-Gd master alloys in an electric resistance furnace. After the addition of Gd, the grain

size of Mg alloys was refined, and the strength, the ultimate tensile strength and hardness were increased [12-14].

The electrochemical codeposition has been widely used to prepare binary or ternary alloys[15–19]. Electrochemical deposition of Gd has been investigated. ZHENG et al[20] synthesized Gd-Co film using an electrodeposition method. LI et al[21] codeposited Gd-Co film in urea-acetamide-NaBr-KBr-GdCl<sub>3</sub>-CoCl<sub>2</sub> baths by potentiostatic electrolysis at 353 K. ZHAN and WANG[22] investigated the possibility of gadolinium electrodeposition in different ionic liquids at 373 K.

However, electrochemical codeposition of Mg-Li-Gd ternary alloys has not been investigated. In this work, Mg-Li-Gd ternary alloys were prepared on a molybdenum electrode in LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub> melts by constant current density electrolysis. XRD, ICP-AES, OM, SEM-EDS were used to characterize the Mg-Li-Gd alloys. Corrosion resistance of the alloy was investigated in sodium chloride solution.

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## 2 Experimental

The mixture of LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub> was melted in 150 cm<sup>3</sup> alumina crucible placed in a quartz cell inside an electric tubular furnace. The temperature of melt was measured with a nickel-chromium thermocouple sheathed by an alumina tube. The LiCl and KCl were dried in electric furnace for more than 48 h at 573 K and 873 K, respectively. Gadolinium and magnesium were introduced into the bath in the form of Gd<sub>2</sub>O<sub>3</sub> ( $\geq$ 99.0%) and dehydrated MgCl<sub>2</sub> powder. All experiments were carried out under an argon atmosphere.

X-ray diffraction (Rigaku TRR III), scanning electron microscopy (SEM JSM-6480A; JEOL Co., Ltd.), optical microscopy (DFC320, Leica Microsystems) and inductively coupled plasma atomic emission spectrometer (ICP-AES, IRIS Intrepid II XSP, Thermo Elemental) analyses were also used.

## **3** Results and discussion

### 3.1 Electrochemical codeposition of Mg-Li-Gd alloys

Electrochemical codeposition of Mg-Li-Gd alloys was carried out by galvanostatic electrolysis on a Mo electrode from LiCl-KCl molten salts containing 10% MgCl<sub>2</sub> and 2% Gd<sub>2</sub>O<sub>3</sub> (mass fraction, the same below if not mentioned) at 823 K. Figure 1 shows the change of the molten salt concentration and the metal content in the Mg-Li-Gd alloys. From Fig.1(a), the Mg (II) and Gd (III) concentrations in the 2%Gd<sub>2</sub>O<sub>3</sub>-10%MgCl<sub>2</sub>-LiCl-KCl molten system, determined by ICP-AES, are 5.73% and 0.19% before electrolysis, respectively. According to Ref.[23], MgCl<sub>2</sub> can react with Gd<sub>2</sub>O<sub>3</sub> at high temperature as follows:

$$MgCl_2 + Gd_2O_3 = 2GdOCl + MgO$$
(1)

$$GdOCl + MgCl_2 = GdCl_3 + MgO$$
(2)

The concentration of  $MgCl_2$  reduces to less than 10% and so we can detect Gd ions by ICP-AES in  $Gd_2O_3$ -MgCl<sub>2</sub>-LiCl-KCl melts at 823 K.

From Fig.1(a) it can be also seen that the MgCl<sub>2</sub> and  $Gd_2O_3$  concentrations in the molten salts decrease quickly in the first 30 min of electrolysis, and then more slowly. After 30 min, LiCl concentration starts to decrease. This indicates that magnesium and gadolinium are deposited in the initial 30 min, and then metal lithium is reduced.

The above results are also confirmed in Fig.1(b), which shows the metal contents of Mg-Li-Gd alloys obtained by constant current density at various electrolysis periods. The alloy contains 96.53% Mg,

0.27% Li and 3.20% Gd after electrolysis for 30 min. The Li content of the alloys increases rapidly, and 43.07% Li in alloy is deposited after electrolysis for 40 min. This suggests that the composition of alloy can be adjusted by controlling the electrolysis period.



**Fig.1** Change of molten salt concentration (a) and composition of alloy obtained by constant current density (b) at different periods

Galvanostatic electrolysis was also carried out in LiCl-KCl melts containing different concentrations of MgCl<sub>2</sub> and Gd<sub>2</sub>O<sub>3</sub> on a molybdenum electrode at 973 K (Table 1). Li content of Mg-Li-Gd alloys increases with the increase of cathode current density and electrolysis period. A higher Gd<sub>2</sub>O<sub>3</sub> concentration in LiCl-KCl melts results in a higher Gd content in Mg-Li-Gd alloys. According to these results, the Gd content and Li content of Mg-Li-Gd alloys can be adjusted by changing the cathode current density and Gd<sub>2</sub>O<sub>3</sub> concentration in LiCl-KCl melts.

#### 3.2 Microstructure and composition of alloys

The microstructures of the Mg-Li-Gd alloys listed in Table 1 are shown in Fig.2. According to Mg-Li phase diagram[24], alloys A and B exhibit dual-phase

**Table 1** ICP analysis of all samples obtained by galvanostatic electrolysis on Mo electrodes ( $S=0.322 \text{ cm}^2$ ) from LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub> melts at 973 K

mgel2-6d263 mens at 775 K						
Alloy	w(MgCl <sub>2</sub> )/	w(Gd <sub>2</sub> O <sub>3</sub> )/	w(Gd)/	w(Li)/	Current/	Time/
	%	%	%	%	Α	h
А	10	-	-	8.3	4.0	1
В	8	0.2	5.2	8.3	2.5	2
С	10	0.2	5.1	19.7	4.0	2
D	15	4.0	16.1	14.8	2.5	4



**Fig.2** Optical micrographs of Mg-Li-Gd alloys alloy A (a), alloy B (b) and alloy C (c)

microstructure ( $\alpha + \beta$  phase), alloys C and D are  $\beta$  phases. As given in Fig.2(a), the bright and dark zones of the optical micrograph correspond to  $\alpha$  and  $\beta$  phases, respectively. The dual-phase microstructure includes a  $\beta$  matrix and a distributed phase in lath form with a width of about 9 µm and a length of approximately 100 µm. Fig.2(c) exhibits only  $\beta$  matrix. Figure 3 shows the XRD patterns of Mg-Li-Gd alloys obtained by galvanostatic electrolysis on a Mo electrode from LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub> melts at 973 K (Table 1). The XRD results are in good agreement with the one obtained in the analysis of optical micrograph. In alloys A and C, there are dual-phase structures ( $\alpha + \beta$  phases). Gd has no effect on the phase structure of Mg-Li-Gd alloys because of the relatively low Gd content in Mg-Li-Gd alloys. When cathode current density is increased, phase transformation from  $\alpha + \beta$  to  $\beta$  is observed (alloy C in Fig.3). However, in alloy D, there are Mg-Gd intermetallic compound phases besides  $\beta$  phase.



Fig.3 XRD patterns of alloys A, B, C and D

# 3.3 Gd element distribution and corrosion behavior of Mg-Li-Gd alloys

SEM and EDS mapping analysis of alloy B were employed to examine the distribution of Mg and Gd elements. The results show that elements Mg and Gd distribute in gray zones and bright zones in lath form, respectively (Fig.4). The gray zone in the SEM micrograph corresponds to the bright zone ( $\alpha$  phase) in optical micrograph, and the bright lath form in the SEM micrograph corresponds to the black zone in the optical micrograph. According to the optical micrograph (Fig.5), element Gd mainly distributes at the grain boundary of Mg-Li-Gd alloys.

The corrosion behavior of alloy B with increasing content of gadolinium was evaluated in sodium chloride solution and compared with that of alloy A without containing gadolinium. Scanning electron microscopy microanalysis was used to characterize the samples. The experiments were carried out in 3.5% NaCl aqueous solution for 30 min at 298 K. Figure 6 shows that alloy A (Mg-8.3Li) has a relatively low corrosion resistance. After immersion in sodium chloride aqueous solution for 30 min at 298 K, severe corrosion is evident. In contrast, SEM characterization of alloy В (Mg-8.3Li-5.2Gd) after immersion in sodium chloride



**Fig.4** SEM image (a), EDS mapping analysis Gd (b) and Mg (c) of alloy B



Fig.5 Microstructure of alloy B

aqueous solution for 30 min at 298 K shows a less accentuated corrosion attack (Fig.6(b)).

As commonly known, corrosion at the grain boundary is most likely to occur. From above results, element Gd mainly distributes at the grain boundary of Mg-Li alloys. Mg-Gd intermetallic compounds enhance corrosion resistance of Mg-Li alloys.



**Fig.6** SEM images of alloy A (a) and B (b) after immersion for 30 min in 3.5% NaCl solution at 298 K

## **4** Conclusions

1) Mg-Li-Gd alloys were successfully prepared by electrochemical codeposition from LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub> melts on molybdenum electrode with constant current density at 823 and 973 K. The Gd content and Li content of Mg-Li-Gd alloys can be adjusted by changing the cathode current density and Gd<sub>2</sub>O<sub>3</sub> concentration in LiCl-KCl melts.

2) The Gd element in Mg-Li-Gd alloys mainly distributes at the grain boundary of the alloy. Corrosion resistance of the alloy is enhanced with the addition of Gd in Mg-Li alloys.

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## LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub>熔盐共电沉积制备 Mg-Li-Gd 合金

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摘 要:在 823 K 和 973 K 的条件下,采用恒电流密度共电沉积法在 LiCl-KCl-MgCl<sub>2</sub>-Gd<sub>2</sub>O<sub>3</sub> 熔盐体系中制备 Mg-Li-Gd 合金,并运用 XRD、SEM、EDS 和 OM 对所制备合金进行微观组织分析。结果表明:在开始的 30 min 内,主要是 Mg 和 Gd 的沉积,所得合金含 96.53% Mg, 3.20% Gd 和 0.27% Li(质量分数),然后 Li 迅速沉积。可 以通过控制电解时间或改变 Gd<sub>2</sub>O<sub>3</sub> 的浓度调节 Mg-Li-Gd 合金的组成。XRD 分析可知,在 Mg-Li-Gd 合金中存在 Mg<sub>3</sub>Gd 相和 Mg<sub>2</sub>Gd 相。从 Gd 元素的面扫描分析可知,Gd 元素主要分布在 Mg-Li-Gd 合金的晶界处。Gd 的添加 增强了合金的抗腐蚀能力。

关键词: 共电沉积; Mg-Li-Gd 合金; 氯化物熔盐; 恒电流电解; Gd<sub>2</sub>O<sub>3</sub>

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