

Available online at www.sciencedirect.com



Trans. Nonferrous Met. Soc. China 20(2010) 2320-2325

Transactions of Nonferrous Metals Society of China

www.tnmsc.cn

### Preparation and photoelectric effect of Zn<sup>2+</sup>-TiO<sub>2</sub> nanotube arrays

ZHOU Yi(周 艺)<sup>1,2</sup>, SHI De-hui(石德晖)<sup>2</sup>, LI Hong(李 宏)<sup>2</sup>, DANG Ming-ming(党铭铭)<sup>2</sup>, LÜ Cai-xia(吕彩霞)<sup>2</sup>, HUANG Ke-long(黄可龙)<sup>1</sup>

1. School of Chemistry and Chemical Engineering, Central South University, Changsha 410083, China;

2. School of Chemical and Biological Engineering, Changsha University of Science and Technology,

Changsha 410114, China

Received 27 November 2009; accepted 26 April 2010

**Abstract:**  $Zn^{2+}$ -  $TiO_2$  nanotube arrays were prepared by anodic oxidation method. The current—time curves were used to investigate their growth mechanism. Scanning electron microscopy and X-ray diffractometry were applied to characterizing their structures and properties. The photoelectrochemical properties were studied by electrochemical impedance spectrum (EIS). The optimised working conditions for  $TiO_2$  nanotube arrays were found to be pH 1, 0.5% HF (mass fraction), 20 V oxidation voltage and for 2 h. The produced sample was in anatase form, with length of 70–100 nm, thickness of 10 nm, uniform diameter and structure that does not collapse under the preparation conditions. The EIS results show that  $TiO_2$  nanotube arrays prepared with 0.5% HF (mass fraction) present a low impedance and  $TiO_2$  nanotube arrays loaded by  $Zn^{2+}$  could have a decreased resistance. This decrease could likely accelerate the transfer of carriers and even increase photoelectric conversion.

Key words:  $Zn^{2+}$ -TiO<sub>2</sub> nanotube array; anodic oxidation method; photoelectric effect; growth mechanism

### **1** Introduction

TiO<sub>2</sub> is an important inorganic functional material with good photoelectronic, photosensitive, gas sensing, and pressure-sensitive characteristics. It has recently attracted much attention in terms of its promising application prospects, such as in the fields of photoelectricity (for solar energy), electronics (types of sensors), and biology (bone growth)[1-3]. Highly ordered TiO<sub>2</sub> nanotube arrays fabricated by anodization constitute a material architecture that offers large internal surface area without a concomitant decrease in geometric and structural order. And the precisely oriented nature of the nanotube arrays makes them excellent electron pathways for vectorial percolation charge transfer between interfaces[4-5]. ZWILLING et al[6] prepared porous TiO<sub>2</sub> films by the anode oxidation method using a metal titanium sheet. This provided the research idea for preparing TiO<sub>2</sub> nanotube arrays. GONG et al[7] prepared TiO<sub>2</sub> nanotube arrays with the electrochemical anode oxidation method. The TiO<sub>2</sub>

nanotube arrays with uniform distribution were previously prepared by the anode oxidation method, but the conversion efficiency still needed improvement[8–9]. During the procedure for improving the photoelectronic activity of TiO<sub>2</sub>, various methods such as element doping[10-11], noble metal deposition[12], surface modification[13] and semiconductor composition[14], have been carried out. Among them, doping element into TiO<sub>2</sub> lattice is an effective way. And at present, nanotubes treated by loading modification have little report in photoelectric material fields. This study is based on the nanotube arrays treated by Zn<sup>2+</sup> loading modification. Zn(NO<sub>3</sub>)<sub>2</sub> was added during the oxidation process to load a certain amount of  $Zn^{2+}$  to the nanotubes. The influence of the oxidation voltage, temperature, time, electrolyte concentration, and content of Zn, as well as the photoelectric effect, on the structure and morphologies of the TiO<sub>2</sub> nanotubes was investigated. And the goal of this study is to prepare  $Zn^{2+}$ -TiO<sub>2</sub> nanotube arrays on Ti sheet and further investigate their photoelectric properties.

Corresponding author: HUANG Ke-long; Tel: +86-731-88879850; E-mail: huangkelong@163.com DOI: 10.1016/S1003-6326(10)60648-6

Foundation item: Project(20976016) supported by the National Natural Science Foundation of China; Project(09JJ606) supported by the Natural Science Foundation of Hunan Province, China; Project(08FJ1002) supported by Key Science Research Project of the Hunan Provincial Natural Science, China

### **2** Experimental

### 2.1 Materials

The materials used for this experiment were hydrofluoric acid (HF, AR), titanium foils (Ti), sodium sulfate (Na<sub>2</sub>SO<sub>4</sub>, AR), zinc nitrate [Zn(NO<sub>3</sub>)<sub>2</sub>, AR], isopropanol (C<sub>3</sub>H<sub>3</sub>O, AR), acetone (CH<sub>3</sub>COCH<sub>3</sub>, AR), and ethanol (CH<sub>3</sub>CH<sub>2</sub>OH, AR).

#### 2.2 Synthesis

### 2.2.1 Synthesis of TiO<sub>2</sub> nanotube arrays

1) The titanium foils were anodized constantly at pH 4 with 1% HF (mass fraction, the same below if not mentioned) under different oxidation voltages. A potentiostat-galvanostat in a two-electrode setup was used to apply constant anodizing voltages of 15, 20, and 30 V typically for 2 h at 20 °C.

2) The titanium foils were anodized at a constant oxidation voltage of 20 V with 0.3%, 0.5%, 0.7%, and 1% HF at 20 °C. The increase rate of voltage was 0.1 V/s.

3) The titanium foils were anodized at a constant HF mass fraction of 0.5% in electrolyte systems with pH values of 1, 4, 7, and 9 at 20  $^{\circ}$ C.

Finally, the anodized samples were washed with de-ionised water, dried in air, and then calcined at 450  $^{\circ}$ C for 2 h.

2.2.2 Synthesis of Zn<sup>2+</sup>-TiO<sub>2</sub> nanotube arrays

The titanium foils were anodized under conditions of 0.5% HF, pH 4, and an oxidation voltage of 20 V. The electrolytes contained 1.0% Zn(NO<sub>3</sub>)<sub>2</sub>. The anodized samples were washed with de-ionised water, dried in air after reacting for 2 h, and then calcined at 450 °C for 2 h.

#### 2.3 Characterisation

The surface morphology and dimension characterisation of the anodized samples were observed on scanning electron microscope (SEM, JEOL JSM-6700F). The crystallographic structures of the produced nanotube arrays were characterised on X-ray diffractometer (XRD, Siemens D5000).

### 2.4 Photoelectric analysis

Photoelectric test was carried out with a threeelectrode system, including a  $TiO_2/Zn^{2+}$ - $TiO_2$  nanotube arrays working electrode, a Pt counter electrode, and a saturated calomel reference electrode. A 0.1 mol/L Na<sub>2</sub>SO<sub>4</sub> solution was used as the electrolyte solution. The electrochemical impedance was measured at room temperature on a CHI660C Electrochemical Workstation working under 8 W ultraviolet lamp ( $\lambda$ =365 nm) with 5 cm-distance between the lamp and electrode.

# 2.5 Analysis of growth mechanism of TiO<sub>2</sub> nanotube arrays

The titanium foils were anodized with 1.0% HF, an oxidation voltage of 5 V, and a duration of 20 min. A CHI660C Electrochemical Workstation with a threeelectrode system, including a  $TiO_2/Zn^{2+}-TiO_2$  nanotube arrays working electrode, a Pt counter electrode, and a saturated calomel reference electrode, was used to measure the current and voltage in order to analyze the growth mechanism of the TiO<sub>2</sub> nanotube arrays.

### **3 Results and discussion**

#### 3.1 XRD analysis

XRD was used to characterize the crystalline structure of the  $TiO_2$  nanotube arrays. By comparing the two XRD patterns shown in Fig.1, it is easy to see that the crystalline structures of the  $TiO_2$  nanotube arrays prepared in different electrolyte systems are not obviously different. The main diffraction peaks at 25.29° and 38.41° are indexed as the (101) and (004) reflections of the crystalline anatase phase.



**Fig.1** XRD patterns of TiO<sub>2</sub> nanotubes prepared with different electrolyte systems

The XRD patterns of pure TiO<sub>2</sub> and Zn<sup>2+</sup>-TiO<sub>2</sub> nanotube arrays are shown in Fig.2. The pattern of the Zn<sup>2+</sup>-TiO<sub>2</sub> nanotube arrays shows a diffraction peak at 56.70°, which is indexed as the (111) reflection of ZnO. The result demonstrates that Zn<sup>2+</sup> has been loaded on the TiO<sub>2</sub> nanotube arrays, and ZnO is the main form of Zn<sup>2+</sup> in the Zn<sup>2+</sup>-TiO<sub>2</sub> nanotube arrays. Therefore, the half peak width of the sample increases and the lattice constant decreases. The patterns shown here have obvious diffraction peaks at 25.29° and 38.41°, indicating that those samples have a regular anatase crystal structure and the addition of Zn<sup>2+</sup> has no effect on the crystalline lattice of the TiO<sub>2</sub> nanotube arrays.

# **3.2 SEM analysis of TiO<sub>2</sub> nanotube arrays prepared under different conditions**

The SEM images of the TiO<sub>2</sub> nanotube arrays



Fig.2 XRD patterns of  $TiO_2$  nanotube arrays and  $Zn^{2+}-TiO_2$  nanotube arrays

prepared on the surface of titanium in electrolyte systems with different PH values are shown in Fig.3. When pH=1, the nanotube arrays with diameter about 70 nm were uniformly distributed on the surface. However, the  $TiO_2$ nanotube arrays have different lengths, creating an irregular arrangement. A  $TiO_2$  nanotube arrays with a sponge structure can be obtained when prepared under pH 4. The reason for the phenomenon is the low mass fraction of H<sup>+</sup> in the reaction during its progress. And a low mass fraction of H<sup>+</sup> in the reaction leads to the decrease of the etching rate and creates an incomplete array structure. There are no nanotube arrays, but some corrosion pits appear in the electrolyte system with pH 7 (Fig.3(b)).

In the 0.5% HF reaction system, although the  $TiO_2$  arrays have different lengths, it is obvious that the  $TiO_2$  nanotube arrays have a better morphology, a regular arrangement, and an average tube diameter of 70 nm (Fig.4). Therefore, the formation rate and the dissolution rate of the  $TiO_2$  nanotube arrays reach equilibrium when the mass fraction of HF is 0.5%.

Fig.5 shows that the nanotube arrays have different morphologies under different anodic oxidation voltages. When the anodic oxidation voltage is 20 V, the nanotube arrays have a regular arrangement, and the average tube diameter is about 80 nm. An unorganised arrangement is obtained when the voltage is 30 V. When it is 15 V, there are spots of nanotube arrays and some corrosion pits on the surface of titanium. The results exhibit that the anodic oxidation voltage has a significant influence on the morphology of the TiO<sub>2</sub> nanotube arrays. When the voltage is too low, there are no TiO<sub>2</sub> nanotube arrays generated, but rather some corrosion pits are formed. When it is too high, the nanotube arrays had an unorganised arrangement and easily collapsed.



Fig.3 SEM images of TiO<sub>2</sub> nanotube arrays prepared in electrolyte systems with different pH values: (a) pH = 1; (b) pH = 7; (c) pH = 4



**Fig.4** SEM images of TiO<sub>2</sub> nanotube arrays prepared in electrolyte systems with different mass fractions of HF: (a) 0.3% HF; (b) 0.5% HF; (c) 1.0% HF

Fig.6 shows the SEM images of  $TiO_2$  and  $Zn^{2+}$ - $TiO_2$  nanotube arrays. It is obvious that the two morphologies are similar. The tube diameter is in the range of 70–100 nm, the tube wall is about 10 nm, and the  $TiO_2$  nanotube arrays are homogeneously distributed on the surface of titanium. The results indicate that  $Zn^{2+}$  loading has little effect on the morphology of the  $Zn^{2+}$ - $TiO_2$  nanotube arrays.

# **3.3** Complex impedance analysis of TiO<sub>2</sub> nanotube arrays

The Nyquist diagrams (Fig.7) show that  $TiO_2$ nanotube arrays prepared with 0.5% HF exhibit low impedance, which might have been influenced by the structure of the  $TiO_2$  photoelectrodes. The SEM images of the  $TiO_2$  nanotube arrays show that the nanotubes are distributed in a disordered fashion when the mass fractions of HF are 0.3% and 1.0%. A collapse even takes place at the mass fraction of 1.0%. All of these could increase the resistance of the array. When the mass fraction of HF is 0.5%, an orderly arrangement of the  $TiO_2$  nanotube arrays is obtained. The order can reduce the interface resistance between the electrode and solution and be beneficial to the interface charge transfer. The general behaviour of the impedance spectra exhibited by the  $Zn^{2+}$ - $TiO_2$  nanotube arrays is similar to that of the TiO<sub>2</sub> nanotube arrays with 0.5% HF. TiO<sub>2</sub> nanotube arrays loaded with  $Zn^{2+}$  would not result in an increase of resistance. Some correspondences at low and high frequencies are revealed in Fig.8(a) and 8(b), respectively. It can be inferred that the response at high frequency is related to the interface between the electrode and the electrolyte, and the response at low frequency can be associated with the diffusion process of the carrier. Furthermore, TiO<sub>2</sub> nanotube arrays loaded by  $Zn^{2+}$  could decrease the resistance, which is likely to accelerate the transfer of the carrier and even increase the photoelectric conversion.

# **3.4** Current—time curve of TiO<sub>2</sub> nanotube array anodizing process

In the first few seconds of the anodic process, an exponential decay of the current occurs (Fig.9). Then the current increases slowly to a quasisteady state. The causes of the current drop are analyzed as follows. Initially,  $Ti^{4+}$  is produced by the reaction of Ti and HF. Then,  $Ti^{4+}$  is oxidized on the surface of the Ti foils. As the dense membranes of TiO<sub>2</sub> are formed on the surface of the Ti foils, a decrease in the anodic current takes place[15–16]. The oxide thickness increases with the increase of electric field intensity in the next stage. With the solubility of the TiO<sub>2</sub> in HF-containing electrolytes



Fig.5 SEM images of TiO<sub>2</sub> nanotube arrays prepared at different voltages: (a) 30 V; (b) 20 V; (c) 15 V



**Fig.6** SEM images of  $TiO_2$  (a) and  $Zn^{2+}-TiO_2$  (b) nanotube arrays



Fig.7 Nyquist diagrams of TiO<sub>2</sub> nanotubes



Fig.8 Frequency–phase of AC impedance diagram: (a)  $\rm TiO_2$  nanotubes; (b )  $\rm Zn^{2+}-\rm TiO_2$  nanotubes

and the electric field,  $TiO_2$  nanotubes with homogenous distribution could be formed on titanium oxide barrier with the disordered pore sites etched by HF. It is favorable for the transfer of  $Ti^{4+}$  from the oxide barrier to the solution during the process of the pitting cores to form titanium oxide nanopores, and  $Zn^{2+}$  is loaded on the  $TiO_2$  nanotubes simultaneously. Due to all of these, the current starts to increase in the stage, leading to the



Fig.9 Anodic oxidation process of TiO<sub>2</sub> nanotube arrays

random growth of  $TiO_2$  nanopores. At the last stage, a more stable and regular self-aligned  $TiO_2$  nanopore growth and dissolution equilibrium of  $Zn^{2+}$  were established; therefore, the current begins to decrease slowly again.

### **4** Conclusions

1) TiO<sub>2</sub> nanotube arrays with regular arrangements are obtained when the mass fraction of HF is 0.5%-1.0%, the pH value is 2–4, and the anodizing voltage is 20 V. The length of these tubes is 70–100 nm, and the thickness of these tubes is about 10 nm. Moreover, the adulterate of zinc does not affect obviously the array structure of the TiO<sub>2</sub> nanotubes.

2) It is known that the growth of the  $Zn^{2+}$ -TiO<sub>2</sub> nanotube arrays could be divided into three obvious phases in the preparation process. Firstly, a TiO<sub>2</sub> film is generated on the titanium substrate by HF acid etching. Secondly, the pitting and etching that occur on the oxide film result in microporous dent and microholes, which are evenly distributed in the basal surface gradually over time. Thirdly, nanotubes no longer elongate when the loading of  $Zn^{2+}$  and the current reach equilibrium. Finally, TiO<sub>2</sub> regions are dissolved through the pores among the holes and formed tubes.

3) According to the analysis on the complex impedance of nanotube arrays, it is obviously that  $TiO_2$  nanotube arrays loaded by  $Zn^{2+}$  could decrease the resistance, which is likely to accelerate the transfer of the carrier and even increase the photoelectric conversion.

### References

- WANG Wen-zhong, VARGHESE O K, PAULOSE M, GRIMES C A, WANG Qing-lei, DICKEY E C. A study on the growth and structure of titania nanotubes [J]. Journal of Materials Research, 2004, 19(2): 417–422.
- [2] MOR G K, SHANKAR K, PAULOSE M, VARGHESE O K,

GRIMES C A. Transparent highly ordered  $TiO_2$  nanotube arrays via anodization of titanium thin films [J]. Advanced Functional Materials, 2005, 8(15): 1291–1296.

- [3] MOR G K, SHANKAR K, PAULOSE M, VARGHESE O K, GRIMES C A. Enhanced photocleavage of water using tiatania nanotube arrays [J]. Nano Letters, 2005, 5: 191–195.
- [4] LEE B G, CHOI J W, LEE S E, JEONG Y S, OH H J, CHI C S. Formation behavior of anodic TiO<sub>2</sub> nanotubes in fluoride containing electrolytes [J]. Transactions of Nonferrous Metals Society of China, 2009, 19(4): 842–845.
- [5] TANG Yu-xin, TAO Jie, ZHANG Yan-yan, WU Tao, TAO Hai-jun, ZHU Ya-rong. Preparation of TiO<sub>2</sub> nanotube on glass by anodization of Ti films at room temperature [J]. Transactions of Nonferrous Metals Society of China, 2009, 19(1): 192–198.
- [6] ZWILLING V, DARQUE-CERETTI E, BOUTRY-FORVEILLE A, DAVID D, PERRIN M Y, AUCOUTURIER M. Preparation and photoelectrocatalytic application of porous TiO<sub>2</sub> film by anodization at low voltage [J]. Surface and Interface Analysis, 1999, 27(7): 629–637.
- [7] GONG Da-wei, GRIMES C A, VARGHESE O K, CHEN Zhi, HU Wen-chong, DICKEY E C. Titanium oxide nanotube arrays prepared by anodic oxidation [J]. Journal of Materials Research, 2001, 16(12): 3331–3334.
- [8] CHEN Huan, WANG Hong, FU Gang. Overview of the titanium oxide nanotube arrays preparation and the mechanism research [J]. Journal of Guangzhou University: Natural Science Edition, 2007, 6(5): 9–12. (in Chinese)
- [9] LIU Ping, LI Xin-yong, WANG Yu-xin, JU Xiao-dong, CHEN Guo-hua. Construction and photoelectrocatalytic properties of TiO<sub>2</sub>

nanotubes arrays on titanium substrates [J]. Chemical Journal of Chinese Universities, 2006, 27(12): 2411–2413. (in Chinese)

- [10] CHEN X F, WANG X C, HOU Y D, HUANG J H, WU L, FU X Z. The effect of postnitridation annealing on the surface property and photocatalytic performance of N-doped TiO<sub>2</sub> under visible light irradiation[J]. Journal of Catalysis, 2008, 255: 59–67.
- [11] HORIKAWA T, KATOH M, TOMIDA T. Preparation and characterization of nitrogen-doped mesoporous titania with high specific surface area[J]. Microporous and Mesoporous Materials, 2008, 110: 397–404.
- [12] NISHIJIMA K, FUKAHORI T, MURAKAMI N, KAMAI T, TSUBOTA T, OHNO T. Development of a titania nanotube (TNT) loaded site-selectively with Pt nanoparticles and their photocatalytic activities[J]. Applied Catalysis A, 2008, 337: 105–109.
- [13] LIANG H C, LI X Z. Visible-induced photocatalytic reactivity of polymer-sensitized titania nanotube films[J]. Applied Catalysis B, 2009, 86: 8–17.
- [14] KIM J C, CHOI J, LEE Y B, HONG J H, LEE J I, YANG J W, LEE W I, HUR N H. Enhanced photocatalytic activity in composites of TiO<sub>2</sub> nanotubes and CdS nanoparticles[J]. Chemical Communications, 2006, 48: 5024–5026.
- [15] WILKE K, BREUER H D. The influence of transition metal doping on the physical and photocatalytic properties of titania [J]. Photoch Photobio A, 1999, 121(1): 49–53.
- [16] KAMAT P V. Photophysical, photochemical and photocatalytic aspects of metal nanoparticles [J]. The Journal of Physical Chemistry B, 2002, 106(32): 7729–7744.

(Edited by LI Xiang-qun)