CORROSION MECHANISM OF SnO₂-BASED INERT ANODES®

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ARSTRACT

According to the extent of their damage in cryolite melts, the corrosion of SnO₂-based inert anodes can be divided into three types, slight corrosion, marked corrosion and catastrophic corrosion. The conditions under which these three types take place and their mechanisms are different. Our work shows that depletion of oxygen containing ions in melts, high density, electrolyte penetration into the electrode and intergranular corrosion contribute greatly to marked corrosion and catastrophic corrosion and that the reduction reaction by aluminium only attacks the electrode slightly.

Key words: SnO3-based inert anodes corrosion corrosion mechanism aluminium eletrolysis

1 INTRODUCTION

The corrosion resistance of the inert anode is one of its most important performance, because it is not only related to its service time, but also to the metal quality. It can be said that the corrosion resistance of an inert anode is one of the key factors, which determines whether if it can be used in Hall-Heroult cell for aluminium production.

Some work has been done on what influences the corrosion rate of inert anodes. However only several hypotheses on the corrosion mechanisms were proposed, and fewer experimental studies have been conducted^[1-8]. Obviously, the further study of this topic will be helpful for understanding the corrosion mechanisms and improving the performance of inert anode, as well as for adjusting the operation parameters to extend their service times. Based on the available literature, we did several supplemented experiments to further investigate the corrosion mechanisms of the SnO₂-based inert anode.

2 POSSIBLE CAUSES OF CORROSION OF THE SnO₂-BASED INERT ANODES

Based on available experimental results and thermodynamic calculations, several possible corrosion mechanisms for SnO₂-based inert anodes in molten cryolite melts were proposed.

2. 1 Reductive Reaction

Because tin has a greater positive electrode potential than dissolved aluminium, sodium and aluminium monofluoride, the tin oxide tends to be reduced. These reactions can be expressed as follows.

$$SnO_2 + 4/3Al = Sn + 2/3Al_2O_3$$
 (1)
 $\Delta G_{950C}^* = -534.76 \text{ kJ}$

$$SnO_{2} + 2AIF = Sn + 2/3AI_{2}O_{3} + 2/3AIF_{3}$$
 (2)
 $\Delta G_{9590}^{0} = -395.13 \text{ kJ}$
 $SnO_{2} + 4Na = Sn + 2Na_{2}O$ (3)

 $\Delta G_{asor}^{o} = -163.35 \text{ kJ}$

If there are no kinetic obstacles, the above reactions are autokinetic. Static corrosion testes [5]. 8] indicated that the electrode corrosion rate in the presence of aluminium is 6 – 65 times as fast as that in the absence of aluminium. It is also thought that the anodic polarization and the oxygen bubbles evolved by anodic reactions around the anode could inhibit the approach of aluminium and other reducing species dissolved in the bath to the anode surface. Therefore the corrosion rate would be slower. A set of experiments have been carried out to verify the above hypothesis. The results are shown in Table 1. The SnO.— based anodes

Table 1 The corrosion rate of the SnO₂-based electrodes

Test No	Experimental conditions	Electrolyte compositions	Corrosion rate* / g · cm ⁻² · h ⁻¹
	no Al. no	C. R. 2.8, Al ₂ O ₃ 8%,	
1	polarization.	CaF ₂ 4%.	9.183×10 ⁻⁴
	no Ar bubble	MgF ₂ 4%, 960℃	
	Al present.	C. R. 2.8, Al ₂ O ₃ 8%,	
2	no polarization.	CaF ₂ 4%.	2.165 × 10 ⁻³
1/0	Ar bubble	MgF ₂ 4%, 960℃	
	Al persent.	C. R. 2.8. Al ₂ O ₃ 8%,	
3	polarization.	CaF, 4%.	4.942 × 10 ⁻³
	Ar bubble	MgF ₂ 4%, 960℃ .	
4	Electrolysis.	C. R. 2.8. Al ₂ O ₃ 8%.	7.795 × 10 ⁻³
	$D_{anode} = 1A / cm^2$.	CaF2 4%.	
	Duration of runs, 4 h	MgF ₂ 4%, 960°C	
5	Electrolysis,	C. R. 2.4, LiF8%,	1.278 × 10 ⁻²
	$D_{\text{anode}} = 2A / \text{cm}^2$	CaF ₂ 4%, MgF ₂	
	Duration of runs, 4 h	4%, Al ₂ O, 4%	

The corrosion rate was measured according to the method in ref [6], and the data listed above are the average value of several experimental results.

in this study were of the same composition and similar properties.

Table I shows that the corrosion rate in the presence of AI is four times larger than in the absence of AI and that the anodic polarization does not decrease the corrosion rate, and that under the same conditions, the corrosion rate makedly increases with the current density. It will be necessary to further verify whether the oxygen bubbles evolved from anodic reaction can decrease the corrosion rate. However, the corrosion of anodes by the reductive reaction is slight(10⁻³g/cm² · h⁻¹) and is affected by mass transfer.

2.2 Fluorination

The SnQ₂-based anodes were fluorinated in the cryolite melt possibly as follows:

$$SnO_2 + 4/3AlF_3 = SnF_4 + 2/3Al_2O_3$$
 (4)

 $\Delta G_{950C}^{\circ} = 115.6 \text{ KJ}$ The equilibrium constant can be expressed as

$$K = \frac{a_{\text{SaF}_4} \cdot a_{\text{Al_2O}_3}^{27/3}}{a_{\text{Al_F}_3}^{4/3}} = 1.155 \times 10^{-3}$$

Assuming $a_{\text{SnF}_4} = 1$, i. e. SnF_4 is saturated in melt, when the cryolitic ratio is 2.2, $a_{\text{AlF}_3} = 1.827 \times 10^{-3}$, then

$$a_{\rm Al,\,O} = 1.31 \times 10^{-13}$$

According to Dewing formula, we have

$$a_{\text{Al}_2\text{O}_3} = \left(C / C_s\right)^{2.77}$$

If $C_{\rm s} = 0.045$, then

$$C_{\text{Al}_2\text{O}_3} = 1.006 \times 10^{-6}$$

which indicates that only when the concentration of alumina is lower than 1.06 × 10⁻⁶, can the fluorination be automatic and attain equilibrium at once with little alumina production. However, when the electrode is anodized and electrolyzed, especially with a high current density, an utterly different situation occurs. As the oxgenontaining ions are continuously

^{* *} So-called polarization means switching on the DC supply, but the anode potential is less than the oxygen evolution potential in the case where the current approaches zero.

consumed in electrolysis and depleted in the melt layer near the anode, it is possible that the concentration of alumina near the anode could attain or be lower than 1.006×10⁻⁶. Then the above reaction would proceed continuously, and the reaction could be read as follows.

$$S_{n}O_{2}+4F^{-}-4e = S_{n}F_{4}+O_{2}$$
 (5)

$$SnO_2+4/3AlF_3=4/3Al+O_2 \uparrow +SnF_4$$

 $\Delta G_{9SOC}^{\circ} = 974.1 \text{ kJ}$

The standard electromotive force for the reaction is

$$E^{\circ} = -2.52 \text{ V}$$

At a current density of 1A / cm2, the over potential for O2 evolution at the SnO2-based anodes is about 0.1 V. Hence the actual decom- position voltage above is about 2.62 V, while the actual decomposition voltage of alumina is about 2.3 V. The potential difference between them is about 0.32 V, which can be easily attained. According to the calculation, when the concentration of alumina is lower than 1.75× 10⁻⁵, the decomposition voltage can also attain 2.62 V, under which circumstances Al2O3 and SnO2 would simultaneously decompose. The SnO2-based inert anodes would quickly be subjected to damage, which explains why the static corrosion rate of the SnO2-based electrode is very small. Once electrolysis began, the corrosion rate increased abruptly as the current density increased. Recently, the steady state polarization curves and the corrosion experiment in pure cryolite for the SnO3-based anodes made by Xiao Haiming et al91 verified the above hypotheses.

Some of the SnF₄ produced from reaction (5) was evapoated, and the rest reacted with alumina to produce SnO₂ again.

SnF₊₂ / 3Al₂O₂ = SnO₂+4/3Al₃ (7)

$$SnF_4+2/3Al_2O_3 = SnO_2+4/3Al_3$$
 (7
 $\Delta G_{expr}^0 = -115.6 \text{ kJ}$

The SnO₂ precipitated from the above reaction was needle-like, which can be seen in the corresponding SEM photos. Otherwise, SnF₄ reacted with NaF to SnF₅ as $SnF_4+NaF=Na_2SnF_5$ $\Delta G_{SNF}^0=-1,652.9 \text{ kJ}$

Wetting and Penetration of Electrolyte in to Electrodes

The microstructure of the SnO_2 -based electrode is clearly shown in Fig. 1. The cracks among particles are about $0.3 \sim 1~\mu$, which makes penetration by electrolytes possible.

The penetration of electrode by electrolytes is closely related to wet ability. The wetting angle of the electrolyte to the SnO₂-based anodes was determined in different Al₂O₃ concentrations, as shown in Fig. 2.



Fig. 1 SEM morphology of the SnO₂-based electrode

Fig.2 shows that the wetting of the SnO₂—based electrode by cryolite-alumina melts is excellent. Whatever the Al₂O₃ concentration is, the melts can wet the whole electrode at quite high speed.

Excellent wetting of the SnO₂-based electrode by cryolite melts and capillary action made it possible for cryolite melts to penetrate the inside of the electrode and gradually fill in the cracks and holes among the SnO₂ particles, which resulted in a lack of ties among the particles and disintegration of tiny SnO₂ particles into the electrolyte. Under unfavorable conditions, such as low alumina concentration, high Life concentration and high current density, the disintegration would proceed more quickly.

Then the electrode would disintegrate and disappear in a few minutes, so it was called catastrophic corrosion. The penetration of the electrode by the electrolyte resulted in the disintegration is shown in Fig. 3.

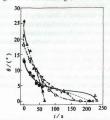


Fig. 2 The wetting angles (θ) of the SnO₂-based electrodes by cryolite-alumina melts as a function of time (t) at 1,012 °C. Alumina concentration (wt.-%):

●-0.0; △-0.5; ○-1.0; ×-2.0; ▲-5.0.

The sample in Fig. 3 was made by the following procedures. After electrolysis ended, the electrode was not taken out but cooled with the furnace. Then, the electrode was fetched with electrolyte to be cut open vertically, and, a SEM photo of the boundary between the electrode and electrolyte was taken. The white parts in Fig. 3 indicate the bulk and a piece of the SnO₂-based electrode, and the black are electrolyte. The disintegration of SnO₂ from the bulk electrode may be produced by reacting SnF₄ with Al₂O₃.

2. 4 Electrode Disintegration Due to the Selective First Solution of CuO

Out of all the constituents of the SnO₂-based electrodes, only fluorination of CuO has a negetive standard Gibbs energy, i. e. the following reaction automatic.

$$2CuO+4/3AlF_3 = 2CuF_2+2/3Al_2O_3$$
 (8)

$\Delta G_{\text{osor}}^{\text{o}} = -92.8 \text{ kJ}$

The electrode sample in the SEM photo displays a marked depletion of Cu near the electrode surface^[6]. The selective first solution of CuO enlarges the cracks and holes among the SnO₂ particles and finally causes disintegration of the electrode because the CuO was used as a binder in the SnO₂-based electrode.



Fig. 3 The vertical section SEM morphology of the electrolyzed SnO₂-based electrode. Electrolysis conditions, CR = 3; C_{Al,O} = 1 wt. -%; i = 1 A / cm²; T = 1.030 T; t = 4.25 h.

3 THE FACTORS INFLUENCING THE CORROSION RATE OF THE SnO₂-BASED INERT ANODES

3. 1 The Effect of Al2O3 Concentration

The effect of Al_2O_3 Concentration on corrosion rate is shown in Fig. 4.

The curves in Fig. 4 can be divided into three sections. The Al₂O₃ concentration in section CD is more than 4% of saturation. Under that condition the corrosion rate of the electrode is very small, and a change in Al₂O₃ concentration has no remarkable effect on the corrosion rate, which indicates that only if there are enough oxygen-containing ions to be transfered close to the anode, the SnO₂-based electrode would be stable. Reduction increases the corrosion rate only a little as Al₂O₃ concentration

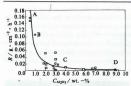


Fig. 4 The effect of Al₂O₃ concentration

Ca_{1,O}, on corrosion rate R of SnO₂-based inert anodes, CR = 2.8, 1A / cm².

decreases. The Al2O3 concentration in section AB is lower than 1 % and the corrosion rate increases abruptly. A tiny change in Al₂O₃ concentration would cause strong effects on the corrosion rate. SnO2 is unstable at that point. Due to a virtual absence of oxygen-containing ions, especially since the Al2O3 concentration in the diffusion layer near the electrode is lower than 10-6, the flurorination and even slight decomposition of SnO2 would occur, and the oxygens are escaped almost from SnO2. Section BC is a transition section. A change in the Al₂O₃ concentration would produce a marked effect. The decrease of the Al-Oconcentration not only preceeds the reduction, but also starts the flurorination of the electrode. In the mean time, one of the oxygen escaping from the anode reaction came from oxygen-containing ions in melts, and the other from SnO.

3. 2 The Effect of Current Density

The effect of current density of the corrosion rate is shown in Fig. 5.

The first three samples were studied at a current density of $0 \sim 2A / cm^2$, and there were a minima at $0.5A / cm^2$ and $1A / cm^2$. Xiao Haiming studied the effect of current density to a greater extent. The above results together show that if the current density is larger than

1A / cm2, the corrosion rate increases abruptly as the current density increases. High current densities caused a greater depletion of oxygencontaining ions in the electrolyte near the electrode, and made SnO2 fluordize or decompose. Therefore, the anode current density could be increased a little only if the Al₂O₃ concentration was quite high in the electrolyte. Otherwise, when the current density was zero, i. e. static corrosion tests, even in cryolite melts with extremly few oxygen-containing ions, the corrosion rate was quite low. That was because the Al2O3 produced from fluorination did not escape in time. The reaction attained equilibrium very soon, and it was impossible for SnO2 to decompose directly.

3. 3 The Effect of Cryolite Ratio

The effect of the CR on corrosion rate of the SnO₂-based electrode is shown in Fig. 6.

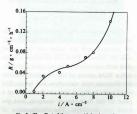


Fig. 5 The effect of the current (i) density on the corrosion rate (R) of the SnO₂-based electrodes

Reference resources; 1—Xue Jilai: 2—Wang Huazhang: 3—Cai Qifeng: 4—Xiao Haiming

The results in Fig. 6 show the corrosion rate increases as the CR decreases in acid electrolyte, when the CR decreases, the activity of AlF, increases, and the fluordination proceeds. In the meantime, a decrease in the CR reduce the solubility of Al₂O₃, limits the

increase of oxygen-containing ions in melts and strengthens the penetration of electrolyte, so further studies are needed.

3. 4 The Effect of LiF Concentration

When the LiF concentration in the electrolyte is lower than 4 %, the corrosion rate of SnO2- based electrodes is very slow, while at LiF concentrations greater than 8%, the corrosion rate increases abruptly[6].

4 CORROSION MECHANISMS SnO,- BASED INERT ELECTRODES

The corrosion of the SnO2-based elect-rode can be divided into three types; slight corrosion, serious corrosion and catastrophic corrosion. The differences among them are shown in Fig.7.

The three electrodes were of the same length before the experiment. Slight corrosion is visible in Fig. 7(a), and the corrosion rate is 0.0061 g/cm² · h. Serious corrosion is in Fig. 7(c), where the corrosion rate is 0.03g/cm². h. Catastrophic corrosion is in illustrated Fig. 7(b), where the corrosion rate is 11 g/cm² · h. The corrosion conditions and mechanisms of corrosions are different as shown in Table 2.

	slight corrosion	Serious corrosion	Catastrophic Corrosion
Corrosion rate / g · cm ⁻² · h ⁻¹	0.001~0.01	0.01~0.1	0.1~10
	Al ₂ O ₃ conc.4% up to sat.	Al ₂ O ₃ 1~4%	Al ₂ O ₃ < 1%
Electrolysis	C. R. 3 ± 0.2	C.R. < 2.6	C. R. < 2.2
conditions	i<1A/cm ²	i ₁ 1~2A/cm ²	i>2A/cm ²
	LiF < 4%	LiF 4~8%	LiF > 8%
	ATTOM A WAREST SON	Reduction	Penetration of electrolyte
Corrosion		Fluorodization	Corrosion between crystals
mechanism		Penetration of electrolyte	Strong fluorodization
		Corrosion between crystals	Direct decomposition of Sn

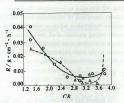


Fig. 6 The effect of the cryolite ratio (CR) on the corrosion rate (R) of the SnO2-based anodes Reference resources: A-Wang Huazhang

O-Xiao Haimig

The destruction due to catastrophic corrosion is very great and will be absolutely forbiden in industrial applications of inert anodes







Fig. 7 The corrosion of the SnO3-based electrode under different electrolysis conditions

in the future. It must be stated that catastrophic corrosion can only occur when subjecting a SnO2-based electrode to electrolysis and other conditions such as extremely low Al₂O₃ concentration or high LiF concentration. So far, it (To be continued on page 20)