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# Synthesis and electrochemical behavior of LiCoO<sub>2</sub> recycled from incisors bound of Li-ion batteries

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Abstract:  $LiCoO_2$  was separated from Al foil with dimethyl acetamide(DMAC), and then polyvinylidene fluoride(PVDF) and carbon powders in active material were eliminated by high temperature calcining. The content of the elements in the recovered powders were analyzed. Then the  $Li_2CO_3$  was added in recycled powders to adjust molar ratio of Li to Co to 1.00, 1.03 and 1.05, respectively. The new  $LiCoO_2$  was obtained by calcining the mixture at 850 °C for 12 h in air. Structure and morphology of the recycled powders and resulted sample were observed by XRD and SEM technique, respectively. The layered structure of the  $LiCoO_2$  is improved with the decrease of molar ratio of Li to Co. The charge/discharge performance, and cyclic voltammograms performance were studied. The recycle-synthesized  $LiCoO_2$  powders, whose molar ratio of Li to Co is 1.0, is found to have the best characteristics as cathode material in terms of charge—discharge capacity and cycling performance. And the cyclic voltammograms(CV) curve shows the lithium extraction/insertion characteristics of the  $LiCoO_2$  well.

Key words: LiCoO<sub>2</sub>; synthesis; electrochemical behavior; cyclic voltammograms

#### **1** Introduction

Layered LiCoO<sub>2</sub> is the most widely used positive electrode material in commercial lithium secondary batteries due to ease of preparation, high electronic conductivity, and excellent cycling performance, etc. The market for Li-ion battery(LIB) has expanded rapidly because of the increase in demand for mobile electronics. It is reported that the total world sales of the Li-ion cells have reached more than 10 billion dollars in 2006[1]. It demands for large amount of Co to meet the market. So in one hand, the LiMn<sub>2</sub>O<sub>4</sub>[2–3], LiFePO<sub>4</sub>[4–5] and Li[NiCoMn]O<sub>2</sub>[6–7] with low or no content of Co have attracted more and more attention because of their low price, abundant resources and safety; and in other hands, many researchers have paid attention to the recycle of LIB[8–12].

JINSIK et al[8] reported that cobalt ions, extracted from waste  $LiCoO_2$  by using a nitric acid leaching solution, were transformed into cobalt hydroxide on a titanium electrode and cobalt oxide was then obtained via a dehydration procedure. ZHANG et al[9] reported the recycle of the valued metal such as cobalt and lithium from the spent Li-ion secondary batteries. MICHEAL[10] recycled the valued metal from waste cell using a special processing technology. The above investigators just recycled the valuable metal from the waste batteries. But CHURL et al[11] and CESTAILE et al[12] not only recycled the valuable metal but also prepared new LiCoO<sub>2</sub> just by using the recycled valuable metal.

In this work, a new LiCoO<sub>2</sub> recycle and synthesis technology from the incisors bound of Li-ion battery is studied. Wherein the impurities of cathode such as polyvinylidene fluoride(PVDF) and carbon powders are eliminated by calcining at high temperature; after recycle the  $Li_2CO_3$  is added in the recycled powders to adjust the molar ratio of Li to Co to 1.00, 1.03 and 1.05; the structure, morphology and electrochemical behaviors of the recycle-synthesized LiCoO<sub>2</sub> are studied.

## 2 Experimental

At first, a certain incisors bound of Li-ion battery was immersed in dimethyl acetamide(DMAC), and the aluminum foil was picked out several hours later. The

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solution was filtrated after several-hour deposition. The cathode material without aluminum was gained after filtration. Then the cathode material was heated at 120  $^{\circ}$ C in air for 12 h. The dry powders were firstly heated at 450  $^{\circ}$ C in air for 2 h, then heated at 600  $^{\circ}$ C for 5 h to gain the recycled powders. The structure and morphology of the recycled powders were characterized with XRD and SEM technique. The content of the Co was analyzed by titrating, and the content of the element Li was analyzed by atomic absorption spectrometry.

The  $Li_2CO_3$  was added in the recycled powders to adjust the molar ratio of Li to Co to 1.00, 1.03 and 1.05, respectively. The final LiCoO<sub>2</sub> was synthesized by calcaining at 850 °C for 12 h in air, and cooled to room temperature in a tube-furnace.

The powders X-ray diffraction(XRD, Rint-2000, Rigaku) measurement using Cu K<sub>a</sub> radiation was employed to identify the crystalline phase of the synthesized materials. And the particle size and morphological character of the LiCoO<sub>2</sub> powders were measured by scanning electron microscope (JEOL, JSM-5600LV) with an accelerating voltage of 20 kV.

The electrochemical characterizations were carried out using a CR2025 coin type cell. For cathode fabrication, the prepared powders were mixed with 10% (mass fraction) of carbon black and 10% (mass fraction) of polyvinylidene fluoride in N-methyl pyrrolidinone until slurry was obtained. The blended slurries were pasted onto an aluminum current collector, and the electrode was dried at 100 °C for 12 h in vacuum. The test cell consisted of the cathode and lithium foil anode coated by a porous polypropylene film, cellgard 2300 as the separator and 1 mol/L LiPF<sub>6</sub> in EC:EMC:DMC  $(C_{3}H_{4}O_{3}:C_{4}H_{8}O_{3}:C_{3}H_{6}O_{3}, 1:1:1$  in volume) as the electrolyte. The assembly of the cells was carried out in a dry Ar-filled glove box. Capacity measurements and cycling tests of the coin-type cells were carried out between 3.0 V and 4.2 V at current density of 0.2 C and temperature of 25°C.

The cyclic voltammetry(CV) measurements were performed on  $LiCoO_2$  electrode at a sweep rate of 0.1 mV/s over a volt of 3.0–4.4 V.

#### **3** Results and discussion

The surface morphology of the recycled  $LiCoO_2$  powders is shown in Fig.1. Some small particles, with diameter of 1  $\mu$ m are observed in Fig.1. Some particles were smashed during the recycle process of high temperature calcining and washing.

XRD patterns of recycled  $LiCoO_2$  powders are shown in Fig.2. The diffraction peaks of  $Co_3O_4$  at 31.3°, 36.8° are observed in Fig.2. To analysis the reaction and validate the reaction results, the recycled powders were

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Fig.1 SEM micrograph of recycled powders



Fig.2 XRD patterns of recycled LiCoO<sub>2</sub> powders

washed by hot water, and then filtrated. White deposition appeared after adding  $CaCl_2$  solution into the hot filtrate. The white deposition should be  $CaF_2$ . We just think that PVDF gives out HF when it is heated. And the HF reacts with LiCoO<sub>2</sub> to produce Co<sub>3</sub>O<sub>4</sub> and LiF.

The content of the elements in the recycled powders were analyzed. The results are listed in Table 1. From Table 1, it can be known that the content of the C and F are little, and the molar ratio of Li to Co is 0.914, which s904

Table 1 Content of elements in recycled LiCoO2 (mass fraction,%)

Li	Co	С	F
6.76	62.88	0.018	0.021

is not consistent with that of the normal LiCoO<sub>2</sub>.

SEM micrographs of the LiCoO<sub>2</sub> synthesized at different molar ratios of Li to Co are shown in Fig.3. Compared with Fig.1 the small particles disappeared and the particle size raised with the increase of the molar ratio of Li to Co. The average particle size is over 5  $\mu$ m. This is because that the small particle reunites when the residual Li<sub>2</sub>CO<sub>3</sub> melts at high temperature.

The final LiCoO<sub>2</sub> was obtained by adding a certain amount of Li<sub>2</sub>CO<sub>3</sub> in the recycled LiCoO<sub>2</sub> and calcining at 850 °C for 12 h. Fig.4 shows the XRD of the LiCoO<sub>2</sub> synthesized at different molar ratios of Li to Co. A pure phase of LiCoO<sub>2</sub> could be identified. The high intensity of the (003) peak and the clear splitting between the (006)/(012) and (018)/(110) peaks can be observed in Fig.4. The spinel phase of  $Co_3O_4$  has disappeared indicating that  $Co_3O_4$  reacts with  $Li_2CO_3$  at high temperature, and well-crystallized single phase  $LiCoO_2$  could be obtained after calcining for 12 h at 850 °C.

The ratio of  $I_{(003)}/I_{(104)}$  of LiCoO<sub>2</sub> synthesized at different molar ratios of Li to Co is listed in Table 2. It shows that the ratio of  $I_{(003)}/I_{(104)}$  of LiCoO<sub>2</sub> is increased with the decrease of molar ratio of Li to Co. The intensity ratio of (003) to (104) is known to be closely related to the degree of cation disordering in the doped LiCoO<sub>2</sub> crystals[13]. We think that the degree of cation disordering is increased by redundant Li inasmuch as higher  $I_{(003)}/I_{(104)}$  ratio corresponds to lower degree of cation mixing and lower R-factor. According to Ref.[14], the lower the R-factor, the better the hexagonal ordering and hence the electrochemical performance.



Fig.3 SEM micrographs of LiCoO<sub>2</sub> synthesized at different molar ratios of Li to Co: (a)-(b) 1.0; (c)-(d) 1.03; (e)-(f) 1.05



**Fig.4** XRD patterns of LiCoO<sub>2</sub> synthesized at different molar ratios of Li to Co of 1.0, 1.03, 1.05 respectively

**Table 2** Ratio of  $I_{(003)}/I_{(104)}$  of LiCoO<sub>2</sub> synthesized at different molar ratios of Li to Co

Molar ratio of Li to Co	1.00	1.03	1.05
$I_{[003]}/I_{[104]}$	2.711	2.655	1.605

The electrochemical performances of  $LiCoO_2$  synthesized at different Li/Co ratios were tested with coin cell. The samples were cycled between 3.0–4.2 V at a constant current density 0.2 *C* rate. The first discharge curves of the samples are given in Fig.5. The initial discharge capacity of LiCoO<sub>2</sub> decreases with the increase of the molar ratio of Li to Co. The discharge capacities of samples with molar ratio of Li to Co of 1.0, 1.03 and 1.05 are 150, 147 and 142 mA·h/g, respectively. This is because that the overdosed Li decreases the hexagonal ordering, and results in lower discharge capacity.



**Fig.5** Discharge curve of LiCoO<sub>2</sub> synthesized at different molar ratios of Li to Co of 1.0, 1.03, 1.05 respectively

The discharge plateau of the three samples is same, just situating between 3.9-4.0 V. These are the same as

those for conventional HT-LiCoO<sub>2</sub>.

Fig.6 shows the cycling data of cells made of  $LiCoO_2$  synthesized at different Li/Co molar ratios. The test cells were cycled at 0.2 *C* rate between 3.0 and 4.2 V at room temperature. The  $LiCoO_2$  whose Li/Co ratio is 1.0 shows the best cycling performance. After 30 cycles, the capacity retention is 93.5%, while those of the other two samples are 93.0% and 91.5%. These results indicate that existence of overdose Li decreases the capacity of  $LiCoO_2$  during charge-discharge cycling. As mentioned before, the channel for  $Li^+$  diffusion is jammed by overdose Li and affects the hexagonal order. So the cycling behavior was affected.



**Fig.6** Cycling performance of LiCoO<sub>2</sub> synthesized at different molar ratios of Li to Co of 1.0, 1.03, 1.05 respectively

Cycle voltammograms technology was used to compare the difference between the recycle-synthesized  $LiCoO_2$  and the normal  $LiCoO_2$ . Fig.7 shows the CV curves of the recycle-synthesized and normal  $LiCoO_2$ . There are no obvious difference between the two curves. This indicates that the structure of the recyclesynthesized  $LiCoO_2$  is similar with that of the normal  $LiCoO_2$ . There are two obvious and one unconspicuous peaks in the oxidation process, and corresponding three



Fig.7 CV curves of LiCoO<sub>2</sub>

peaks in reductive process. The curve is the same as what in GEORGE's report[14]. The three cathodic and anodic peaks correspond to as many LiCoO<sub>2</sub> structure changes. The first peak at 4.045 V is related to the existence of a two-phase domain for Li<sub>x</sub>CoO<sub>2</sub>( $0.75 \le x \le 0.94$ ) and the third peak at 4.19 V corresponded to the presence of a monoclinic distortion for x=0.5 due to an interslab Li/vacancy ordering. The second peak at 4.09 V is unidentified[15].

### **4** Conclusions

1) Cathodic powders are separated from the Al foil with dimethyl acetamide(DMAC) easily, and then the impurity, polyvinylidene fluoride(PVDF) and carbon powders, in the active material are eliminated by high temperature calcining, and there are some  $Co_3O_4$  in the obtained powders.

2) The new LiCoO<sub>2</sub> is obtained after calcine for 12 h in air at 850  $^{\circ}$ C by adding a certain amount of Li<sub>2</sub>CO<sub>3</sub>. The well-crystallized single phase LiCoO<sub>2</sub> without Co<sub>3</sub>O<sub>4</sub> phase is obtained. The layered structure of the LiCoO<sub>2</sub> is improved when the Li/Co molar ratio is lower.

3) The recycle-synthesized  $LiCoO_2$  powders, whose Li/Co molar ratio is 1.0, is found to have the best discharge capacity and cycling behavior. The first discharge capacity is 150 mAh/g, and the capacity retention is 93.5% after 30 cycles.

4) The cyclic voltammograms curve shows that there is no structure difference in the synthesized  $LiCoO_2$  and normal  $LiCoO_2$ . And it also shows the lithium extraction/ insertion performances of the  $LiCoO_2$  well.

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s906