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High temperature DSC study on interfacial reaction of aluminum borate whisker reinforced aluminum alloys[®]

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Abstract: Aluminum borate $(9Al_2O_32B_2O_3)$ whisker reinforced ZL109 alloy and pure Al were fabricated by squeeze casting technique. By means of high temperature DSC, the interfacial reaction of these composites was studied in detail, and reaction products were analyzed by using XRD and SEM. It was shown that in the $9Al_2O_32B_2O_3$, w/Al composites, the pure Al can react with the $9Al_2O_32B_2O_3$ whisker and the critical temperature is $756\,^{\circ}\mathrm{C}$. In the composites, the chemical reaction occurs at the interface and the product is identified to be spinel $MgAl_2O_4$. The magnesium is one of main factors that cause the interfacial reaction, oxygen promotes the interfacial reaction, but Cu, Ni elements in ZL109 alloy don't take part in the interfacial reaction. The temperature is another main factors that cause interfacial reaction. When the temperature is higher than $690\,^{\circ}\mathrm{C}$, both Mg and Al elements can react with the whisker. When the temperature ranges from 449 $^{\circ}\mathrm{C}$ to $690\,^{\circ}\mathrm{C}$, only Mg element reacts with the whisker. T_6 heat treatment causes more serious interfacial reaction.

Key words: composites; aluminum borate whisker; interfacial reaction Document code: A

1 INTRODUCTION

Recently, the ceramic whisker reinforced metals have been paid more attention because these composites possess excellent behaviors such as low density, good strength, high specific stiffness and specific modulus, excellent fatigueresistance and wear-resistance, and whisker/metal composites become an important branch of metal matrix composites. The early developed whisker reinforced metal are mainly expensive SiC and Si₃N₄ series composites, so that the wide applications of this kind of composites are limited. At present, a state of art whisker-aluminum borate whisker, has been developed on the base of improved production technique. The aluminum borate whisker has excellent physical properties and the outstanding advantage is that its commercial price only ranges from 1/10 to 1/20 of SiC and Si₃N₄ whiskers. The aluminum-based composites with aluminum borate whisker have excellent strength and modulus which are in the same range as that of SiC_w/Al or Si₃N_{4, w}/Al composites, but their thermal expansion is lower and abrasion resistance is better^[1~5]. It makes this new kind of composites to get wider applications in the future. Interfacial reaction between matrix alloys and reinforcers have considerably strong influence on interfacial microstructure and mechanical properties of the composites $[6^{-8}]$. In this kind of composites, interfacial reaction is very serious, but the excessive reaction causes whiskers surface degradation and reduction[2,3,9,10]. It is necessary to study the interfacial reaction on aluminum borate whisker reinforced aluminumbased composites for their properties improving, interface design, and application developing.

2 EXPERIMENTAL

2.1 Materials

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Aluminum borate whisker (NICHIAS Co. Ltd., Japan) was employed as reinforcer in this study. The crystal structure of aluminum borate whisker(AlBO $_{\rm w}$) is of orthorhombic, similar to that of Al $_2$ O $_3$ -SiO $_2$ series substance such as andalusite, sillimanite and mullite. Its space group is A2 $_1$ am and physical properties are listed in Table 1 $^{[3,11]}$. The commercial ZL109 alloy and pure Al were used as the matrices, the chemical composition of the alloy is listed in Table 2.

Table 1 Material properties of AlBO_w

Property	Value
Average length/μm	10~30
Average diameter/µm	$0.5 \sim 1.0$
Density/(g·cm ⁻³)	2.93
Tensile strength/GPa	8
Elastic modulus/GPa	400
Mohs hardness	7
Thermal expansion coefficient/°C -1	4.8

 Table 2 Chemical compositions of ZL109 alloy (%)

 Si
 Mg
 Cu
 Ni
 Al

 11.0~13.00.8~1.3 0.5~1.5 0.8~1.5 Balance

2.2 Preparation of composites

Aluminum borate whisker reinforced aluminum alloys were produced by squeeze casting process. The whisker preform was preheated at 750 °C, then inseted in a casting die of 300 °C. The molten aluminum of 750 °C was poured into the die and a high squeeze pressure of 100 MPa was applied immediately, then aluminum melt was infiltrated into the whisker preform, solidified and cooled down under the pressure for 30 s. Finally the AlBO_w reinforced aluminum composites were separated from the die.

In order to investigate the effect of heat treatment on interfacial reaction of the composites, the T_6 treatment were done under the conditions as follows: 530 °C, 3 h for solid solution, and quenching at 75 °C, then 160 °C, 24 h for aging treatment.

2.3 Analysis methods

A Netzsch DSC-404 high temperature thermal analyzer was used to measure heat behaviors on interfacial reaction of the composites in detail. The AlBO whisker was etched from the composites by using the chemical dissolution method and the chemical constitution as well as phase identification of reaction products were analyzed by X-ray diffraction analysis (Simens D-500 type X-ray diffractometer, copper target). The morphological changes of the AlBO_w were observed in a Philips XL-30 scanning electron microscope.

3 RESULTS AND DISCUSION

3.1 Interfacial reaction of AlBO_w/Al compos-

As reported in Ref. [3,9], no any reaction product was found at the interface of AlBO_w/Al composites by means of SEM, TEM and HREM. It is generally accepted that pure Al can not react with AlBO whisker in the composites. In the present study, the reactivity between pure Al and AlBO_w was checked based on the thermodynamic calculation (data based on Ref. [12]) and the Gibbs free energy change associated with the reaction (1) is

4Al+9Al₂O₃2B₂O₃→11
$$\gamma$$
-Al₂O₃+4B (1)
 $\Delta G \approx (0.15 T - 500.94)$ kJ/mol<0
(25 $\mathbb{C} \leq T \leq 827 \mathbb{C}$).

Besides, the Gibbs function of formation of oxidization reactions are calculated between B, Al, Mg, Cu, Ni and $O_2(O_2)$ is the per mole) and the curve of $\Delta G - T$ was obtained (as shown in Fig. 1). It is known from Fig. 1 that B element can be reduced from its oxide by Al element, which means the chemical reaction maybe exist in the composites according to Eqn. (1).

When reaction occurs at the AlBO_w/Al matrix interface, interfacial reaction products formed as a result of the reaction will alter the chemical composition of the matrix alloy, resulting in changes in thermal responses of the matrix alloy. In this study, the thermal behaviors of the as-cast AlBO_w/Al composites were measured by high temperature DSC. The specimen was heated from 200 °C to 950 °C under protection of Ar gas, and the heating rate was 10 °C/min. The

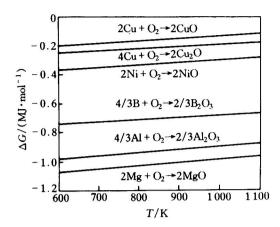


Fig. 1 Temperature dependence of Gibbs function of formation of oxidization reaction

curve of the DSC analysis is shown as Fig.2, the endothermal peak during $641 \sim 687 \, ^{\circ} \! \text{C}$ corresponds to the melt of the Al matrix and the exothermic peak during $756 \sim 908 \, ^{\circ} \! \text{C}$ should be caused by the chemical reaction in the composites.

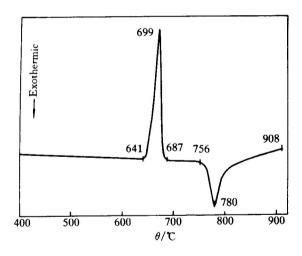


Fig. 2 DSC curve of as-cast AlBO_w/Al composites

It can be seen from Fig. 3(b) by XRD that AlBO whisker is stable after heat treatment at $850\,^{\circ}\mathrm{C}$ for 12 h, which indicates the chemical reaction does not related to the discomposition of AlBO whisker. In order to inquire this reaction, the AlBO_w/Al composites specimens were treat-

ed by T₆ process and at 780 °C for 15 min, respectively. When the size and volume fraction of the interfacial reaction products in the composites are relatively small, it is difficult to determine the exact phase identification by means of X-ray diffraction. In order to overcome this problem, AlBOw as well as the interfacial reaction products were extracted from the composites, so that X-ray diffraction was performed on the extracted whisker. Based on XRD results (Fig. 3(c) and (d)), it is known that no new substance was found in T₆-treated composites. But the composites which treated at 780 °C for 15 min obviously showed that the pure Al should be reacted with the whisker since γ-Al₂O₃ was formed. In other words, the interfacial reaction Eqn. (1) can take place when the temperature is higher than 756°C. In this work, no interfacial reaction is found in the AlBOw/Al composites because the maximum processing temperature is 750 ℃.

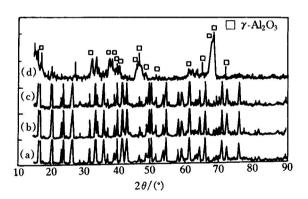


Fig. 3 X-ray diffraction patterns of original and extracted AlBO whisker (a)—Original whisker; (b)—Whisker at 850 ℃ for 12 h;

(b) Whisker at 650 C for 12 h;

(c)—Extracted whisker of T₆-composites;

(d)—Extracted whisker of composites at 780 ℃ for 15 min

3.2 Interfacial reaction of AlBO_w/ZL109 composites

The as-cast $AlBO_w/ZL109$ composites was studied by means of the same high temperature DSC analysis, two exothermic responses are observed (Fig. 4(a)). This means that chemical

reaction occurs in the AlBO_w/ZL109 composites. Fig.4(b) shows the DSC curve of ZL109 alloy under the same conditions as the composites. Compared Fig. 4(a) with Fig. 4(b), it shows that the exothermic responses do not result from the reaction or precipitation among compositions in the matrices but due to the interaction between AlBO_w and matrix alloys. Moreover, it is evident from Fig. 1 that both Mg and Al element can reduce B element from its oxide, but Cu and Ni can't displace B element. As a result, it is reasonable to consider that the chemical reaction is caused by interaction between Mg, Al and Al-BO whisker while Cu and Ni elements don't take part in the interfacial reaction.

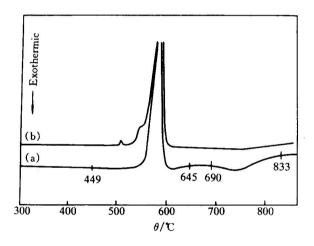


Fig. 4 DSC scans of as-cast AlBO_w/ZL109 composites and matrix alloy
(a)—AlBO_w/ZL109 composites; (b)—ZL109 matrix

In order to determine the exact phase identification by means of X-ray diffraction, AlBO whisker as well as the interfacial reaction products were extracted from the T₆ heat-treated composites by using 10% HCl solution. From the X-ray diffraction pattern (Fig. 5(b)) performed on the extracted whisker, the interfacial reaction products are identified as spinel MgAl₂O₄ and Si which is formed by eutectic reaction, and usually nucleate and grow at whisker/Si interface^[9], which means the MgAl₂O₄ is the interfacial reaction products only. This result indicates that magnesium could

react with the whisker during the interfacial reaction of $AlBO_{\rm w}/ZL109$ composites.

Morphologies of the interfacial reaction products were shown in Fig. 6. It is easy to find that there are more interfacial reaction products in the T₆ heat-treated composites than that in the as-cast one. Because the solid solution temperature of T_6 heat treatment is 530 °C, it indicates that magnesium reacts easily with the whisker at the temperature of $449 \sim 645 \, \text{C}$, so that T_6 treatment causes more serious interfacial reaction. According to the DSC curve (Fig. 4(a)), it is founded that no thermal responses exist in the composites when the temperature ranges from 645 ℃ to 690 ℃, it means that Mg element at the interface is almost exhausted. Therefore, campared the DSC curves shown in Fig. 4(a) and Fig. 1, it is reasonable to consider that the exothermic peak of 690 ~ 833 ℃ is caused by the reaction between Al and the AlBO whisker.

In general, although it was accepted by most researchers that Mg and O elements in the matrix take part in the interfacial reaction in the AlBO_w reinforced aluminum-based composites, there were different views about interfacial reaction mechanism. For example, Sugumara et $al^{[1,2,13]}$ put forward that due to the existence of Mg element, the interfacial reaction is as

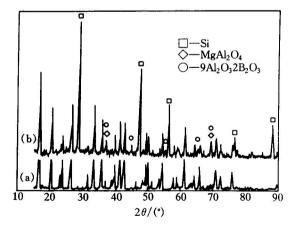


Fig. 5 X-ray diffraction patterns of original and extracted AlBO whisker
(a)—Original wisker;
(b)—Extracted wisker of T₆-composites

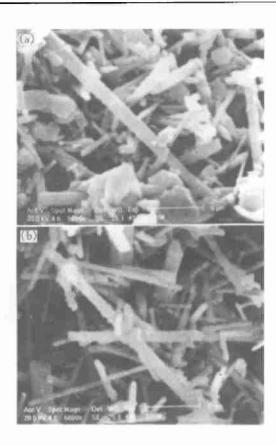


Fig. 6 SEM micrographs of whisker extracted from AlBO_w/ZL109 composites

(a)—Whisker extracted from as-cast composites;

(b)—Whisker extracted from

T₆-treated composites

follows:

$$9Al_2O_32B_2O_3 + 4Al \rightarrow 11 \gamma - Al_2O_3 + 4B$$

 $\gamma - Al_2O_3 + Mg + [O] \rightarrow MgAl_2O_4$

But Pan et al^[3] proposed the following process: 9Mg + 9Al₂O₃2B₂O₃ + 3[O] → MgAl₂O₄ + 4B

However, [O] is not necessary factor that cause the interfacial reaction in spite that it can promote the interfacial reaction greatly (Fig. 7). In other words, magnesium can react with the Al-BO_w when there is no existence of [O] in the composites. The chemical reaction between Mg and AlBO_w can be given by Eqns. (2), (3):

$$6Mg + 9Al_2O_32B_2O_3 \rightarrow 6MgAl_2O_4 + 3\gamma$$
-
 $Al_2O_3 + 4B$ (2)

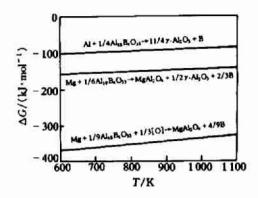


Fig. 7 Thermodynamic calculation between Mg, Al and AlBO whisker

$$3Mg + 4\gamma - Al_2O_3 \rightarrow 3MgAl_2O_4 + 2Al$$
 (3)

The reason for the two interfacial reaction equation can be explained as follows. Firstly, although both Mg and Al elements can displace B element from its oxide, Al element can be also reduced from its oxide by Mg element (Fig. 1). This indicates that the reaction between Mg element and AlBO_w is easier than Al element, means Eqn. (2) as well as Eqn. (1) can take place in the composites. Secondly, both Eqns. (3) and (4) may take place. But Lloyd [14] and Fishkis [15] pointed out that

3Mg + Al₂O₃→3MgO + 2Al (4) the interfacial reaction products are not MgO but MgAl₂O₄ in the Al₂O₃/Al-4% Mg composites based on the dynamic consideration. In the present study, the ZL109 alloy is selected as matrix and the mass fraction of magnesium in matrix ranges from 0.8% to 1.3%. Therefore, it is reasonable to consider that Eqn. (4) does not occur in the composites. Furthermore, because there exist Mg segregation at the interface between matrix and AlBO_w^[3,10], the reaction opportunity between Mg and AlBO_w are greater than that between Al and AlBO_w.

Based on the experimental results and analyses, the existence of interfacial reaction in the AlBO_w/ZL109 composites not only depends on the processing temperature, but also depends on the alloy elements in the matrix. Finally, the process of interfacial reaction in AlBO_w/ZL109 composites can be traced;

 $18Mg + 29Al_2O_32B_2O_3 + 3O_2 \rightarrow 18MgAl_2O_4 + 8B$

 $6Mg + 9Al_2O_32B_2O_3 \rightarrow 6MgAl_2O_4 + 3\gamma - Al_2O_3 + 4B$

 $2Mg + 2\gamma - Al_2O_3 + O_2 \rightarrow 2MgAl_2O_4$ $3Mg + 4\gamma - Al_2O_3 \rightarrow 3MgAl_2O_4 + 2Al$ $4Al + 9Al_2O_3 2B_2O_3 \rightarrow 11\gamma - Al_2O_3 + 4B$

Only Mg element can react with AlBO $_{\rm w}$ when the processing temperature ranges from 449 °C to 690 °C:

 $18Mg + 29Al_2O_32B_2O_3 + 3O_2 \rightarrow 18MgAl_2O_4 + 8B$

 $6Mg + 9Al_2O_32B_2O_3 \rightarrow 6MgAl_2O_4 + 3\gamma$ - $Al_2O_3 + 4B$

 $2Mg + 2\gamma - Al_2O_3 + O_2 \rightarrow 2MgAl_2O_4$ $3Mg + 4\gamma - Al_2O_3 \rightarrow 3MgAl_2O_4 + 2Al_2O_3 \rightarrow 2MgAl_2O_4 + 2Al_2O_3 + 2Al_2O_3$

4 CONCLUSIONS

- (1) The temperature of reaction between Al and AlBO_w is 756 °C in the AlBO_w/Al composites. The interfacial reaction product is γ -Al₂O₃.
- (2) In the $AlBO_{\rm w}/ZL109$ composites, Mg is one of the main factors that cause the interfacial reaction, Al also can react with $AlBO_{\rm w}$, O can promote the interfacial reaction, and Cu, Ni don't take part in the interfacial reaction.
- (3) For $9Al_2O_32B_2O_3$ /ZL109 composites, the processing temperature is another factor that causes interfacial reaction. When the temperature is higher than 690 °C, both Mg and Al ele-

ment react with whisker. When the temperature ranges from 449 °C to 690 °C only Mg element takes part in the interfacial reaction and the reaction product is $MgAl_2O_4$ with spinel structure.

(4) T₆ treatment causes very serious interfacial reaction between AlBO_w and ZL109 alloys.

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