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Effect of TiO₂ cathode performance on preparation of Ti by electro-deoxidation

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Abstract: To resolve the problem of low current efficiency and low reaction rate when preparing metal by electro-deoxidization method, various contents of CaO and CaCO₃ were added into the TiO_2 powder, which were compacted into pellets and sintered, and then conducted as the cathode. The morphology and contents of the cathode and products were studied by scanning electron microscopy and X-ray diffractometry, the effect of the additives were studied. The results show that formation of calcium titanate and change of microstructure within the cathode enhance the velocity of the direct electrochemical reduction of solid TiO_2 to Ti.

Key words: titanium; molten salt; electro-deoxidization; additive; electrolysis activity

1 Introduction

Titanium has been widely used in aerospace, chemical engineering and pharmacy because it has high strength, low density, excellent corrosive resistance. Now used preparation technologies, eg the Kroll process, have drawbacks of high cost, complicated process, high energy consumption, pollution of toxic chlorine gas, so we need develop new preparation technology of titanium which is low cost, short process flow and environmentally friendly.

Electro-deoxidization of titania in molten salt is a method to prepare metal titanium by novel electrochemistry. The process flow is short and friendly to environment. Both of domestic and abroad researcher do much research work about preparation of titanium metal by electro-deoxidization of titania in molten salt[1-9], the cathode of FFC method is titania powder pressed via cool isostatic pressing and then sintered to pellet, because the electrolytic process involves with solid phase mass transfer and phase transition, the reaction rate is slow, and efficiency of electrolytic reaction is low, so the current efficiency of preparation of metal titanium is low. Many researchers do much work to resolve these problems. In this paper, various contents of CaO and CaCO₃ were doped into TiO₂ powders and the doped powders pressed to pellets respectively to enhance the activity of the electrochemistry reaction of the cathode pallets, improve the electrolysis efficiency, lower the energy consumption. Microstructure transformation of the doped TiO_2 which were compacted and sintered, and the effect of the dopants to electro-deoxidization reaction of the TiO_2 cathode were studied.

2 Theoretical consideration

The electro-deoxidization in CaCl₂ melt of TiO₂ is a much complex reaction process. The overall reaction involves not only with molten salt electrochemistry, chemical thermodynamics, mass transfer, but also with phase transformation dynamics, etc. Recently, SCHWANDT et al[2–3] proposed that during different stages of preparation of metal titanium by electro-deoxidization, the control step of the reaction is also different.

Integrate the theory with CHEN et al[4–5], YAN et al[6–7], XU[10–11], DU[8–9,12] proposed factors affecting molten electro-deoxidization reaction rate proposed as follows.

2.1 Electrochemistry reaction factor

Electron transfer process between ionic melt and

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 TiO_2 cathode accompanies the reduction process of the TiO_2 cathode:

$$4\text{TiO}_2 + \text{Ca}^{2+} + 2e = \text{Ti}_3\text{O}_5 + \text{Ca}\text{TiO}_3 \tag{1}$$

$$3TiO_2 + Ca^{2+} + 2e = Ti_2O_3 + CaTiO_3$$
 (2)

$$2\text{TiO}_2 + \text{Ca}^{2+} + 2e = \text{TiO} + \text{Ca}\text{TiO}_3 \tag{3}$$

$$2CaTiO_3 + 2e = CaTi_2O_4 + Ca^{2+} + 2O^{2-}$$
(4)

$$CaTi_{2}O_{4}+2e = Ca^{2+}+2TiO+2O^{2-}$$
(5)

$$TiO+2(1-\delta)e = TiO_{\delta}+(1-\delta)O^{2-}$$
(6)

2.2 Chemical reaction factor

The chemical reaction is as follows:	
CaTiO ₃ +TiO=CaTi ₂ O ₄	(7)

2.3 Mass transfer processes

During the electro-deoxidization process of TiO_2 cathode, mass transfer process in solid phase includes diffuseness of oxygen ion and titanium ion, diffuseness of molten salt in porous cathode, oxygen ion diffuseness and transfer in molten salt.

2.4 New phase formation and phase transfer processes

Because TiO_2 in cathode reacts with Ca^{2+} in molten salt by electrochemistry mechanism, important intermediate products $CaTiO_3$ and $CaTi_2O_4$ were created, which consists of perovskite and pseudo-pyromelane structure respectively; on the later period of electro-deoxidization reaction, the structure of the product is face-centered cubic TiO and close packed hexahedron Ti(O). The above process includes the process of nucleation and growth up of new phase. At the same time, because of partial Ti⁴⁺ are deoxidized, TiO₂ are transformed from rutile to magneli phase, so small amount CaO or CaCO₃ are doped into TiO₂, the dopants would accur reaction as follows:

$$CaO \xrightarrow{TiO_2} Ca_{Ti}^{"} + O_O^{\times} + V_O^{\bullet}$$
(8)

Rutile TiO₂ will form cation impurity defect Ca"_{Ti}, at the same time, on the basis of electroneutrality principle, the oxygen ion vacancy V"_O will be formed in TiO₂ crystal lattice, this will be propitious to diffuse of oxygen ion in solidoid during the electro-deoxidization process of cathode, along with the increase of defect of cathode, the amount of active site in solid cathode will be increased, which will promote the electrochemical reaction and chemical reaction. When CaO or CaCO₃ are doped into TiO₂, the cathode will probably form perovskite structure after sintering, which may act as crystal nucleus of intermediate product CaTiO₃ of the electro-deoxidization reaction. This will improve the formation of CaTiO₃ in cathode, and are likely to restrain over growth of CaTiO₃ crystal.

Furthermore, decomposing and emitting of CO_2 during sintering will affect pore distribution and pore size of TiO₂ cathode doped CaCO₃.

It is very important that the reaction of electrodeoxidization occurs in $CaCl_2$ molten, both $CaTiO_3$ and $CaTi_2O_4$ are just the intermediate product of electrodeoxidization, so comparing with pure TiO₂ cathode, doped CaO or CaCO₃ in cathode could not bring out impurity in final product.

3 Experimental

Various contents of CaO or CaCO₃ were doped into TiO₂ powder (AR grade), then it were compacted and sintered at different temperature to make into 2 g TiO₂ pallets. Sintering temperature were 750, 900 and 1 200 °C separately. Using CaCl₂ molten as electrolyte, sintered TiO₂ as cathode, graphite rod as anode, the pallets were electrolyzed at 900 °C under 3.0 V for 6 h. Argon was admitted into the sealed reactor continuously when temperature was higher than 300 °C. After electrolysis, the product was rinsed with distilled water, dried at low temperature of 100 °C.

The microstructure and morphology of the cathode were studied by JSM-6400 scanning electron microscope (SEM). The contents of the product were analysed by INCA X-sight energy dispersive spectrum (EDS). Phase compositions of cathode were studied by D/max-2200PC X-ray diffraction (XRD).

4 Results sand discussion

The intensity of the TiO_2 cathode is markedly increased after sintering, which prevents the cathode pulverization by molten salt during electrolysis, at the same time, the junction effect of powder particles of cathode is enhanced by high temperature sintering, which will improve constructing conduct net in TiO_2 cathode.

Fig.1 shows the SEM images of TiO₂ cathode which was compacted under 100 MPa, and sintered for 4 h at 750, 900 and 1 200 °C separately. It can be observed from the SEM images that the particle size of the cathode pallet sintered at 750 °C is much small, the particles are densely arranged, the particle size of the pallet sintered at 900 °C is larger, pore size between particles also grow larger, the particle size of the pallet sintered at 1 200 °C is about several 10 times larger than the pallet sintered at 750 °C, pore size of the pallets is much larger. The densities of the three pallets sintered at different temperature are listed in Table 1. The sintering temperature greatly affects the volume density of the pallets, the volume density of the pallets sintered at 1 200 °C is the maximal. The density of the cathode pallets is controlled by the size of particle and array compactness of the pallets. Because the particles size increases and array compacts, the volume of the pallet is contracted after being sintered, and the density of the pallets increases. The cathode pallets sintered separately at 750, 900 and 1 200 °C were electrolyzed 6 h, microstructure and contents of the cross section were studied by SEM and EDS line scanning (Fig.2).

The results shows that the cathode pallets is layered after electrolysis, the main content of the outer layer is Ti, O content is very lower in out layer. But in inner layer,

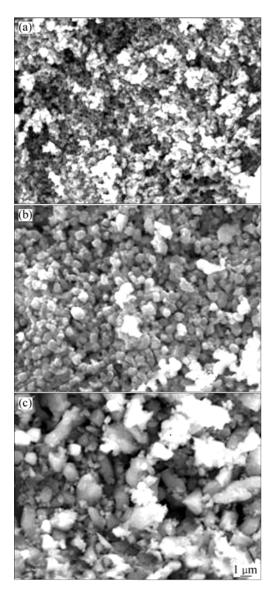


Fig.1 SEM images of TiO₂ pellets after sintering at various temperatures for 4 h: (a) 750 °C; (b) 900 °C; (c) 1 200 °C

 Table 1 Densities of cathode under various sintering temperatures for 4 h

Sintering temperature of sample/°C	700	900	1 200
Density/ $(g \cdot cm^{-3})$	1.97	2.03	3.31

the oxygen content is higher, Ti content is relatively low, the thickness of metal layer of electrolyzed cathode pallets which were sintered at 750, 900 and 1 200 °C are 0.75, 0.70, 0.52 mm. Although the metal layer of the pallets sintered at 750 °C has the maximum thickness, but it's intensity is so poor that it is incline to be pulverized by molten salt. So 900 °C is the optimal sintering temperature for the doped cathode pallets.

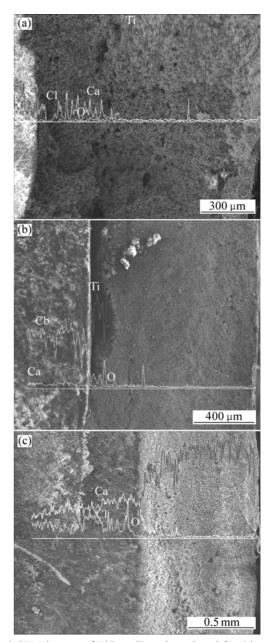


Fig.2 SEM images of TiO₂ pellets electrolyzed for 6 h at 900 $^{\circ}$ C after sintering at various temperatures for 4 h: (a) 750 $^{\circ}$ C; (b) 900 $^{\circ}$ C; (c) 1 200 $^{\circ}$ C

Fig.3 shows the SEM images of the sintered cathode pallets which doped with different concentration of CaO (mass fraction, %), macroscopically, volume of the pallets contract obviously after being sintered, and it's intensity is enhanced too. It can be observed from Fig.3

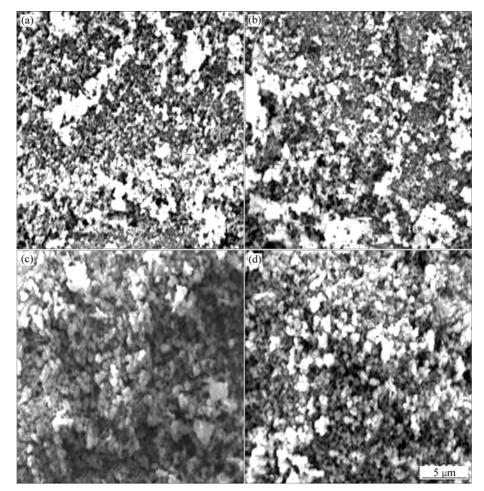


Fig.3 SEM images of TiO₂ pellets with various contents of CaO after sintering at 900 °C for 6 h: (a) 1%; (b) 5%; (c) 10%; (d) 15%

that the particle size of the sintered cathode is smaller when the content of CaO is 1% and 5%, the particle size of the sintered cathode is obviously larger when content of CaO is 10% and 15%.

Fig.4 shows the XRD pattern of the sintered TiO_2 cathode pallets doped 10% CaO. There is diffraction peak of CaTiO₃ in the XRD spectrum, indicating that

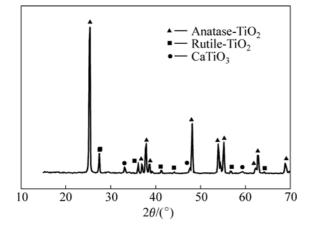


Fig.4 X-ray diffraction pattern of TiO₂ pellets with 10% CaO after sintering at 900 $^{\circ}$ C for 6 h

 TiO_2 doped CaO would form new phase CaTiO₃ after being sintered. Formation of CaTiO₃ may be an important factor of growth of cathode pallets which doped 10% or 15% CaO after being sintered.

From Fig.4 we found that there is two kinds of diffraction peaks of TiO_2 for the cathode pallets after being sintered, rutile- TiO_2 and Anatase- TiO_2 , no CaO peak is found, this may be explained as majority CaO reacted with TiO_2 to produce CaTiO_3. If all CaO (10%) dopes into TiO_2 and forms CaTiO_3, mass fraction of CaTiO_3 in cathode is 24.29%.

Fig.5 shows SEM images of electrolyzed cathode pallets which doped different concentration of CaO. From Fig.5 we can know that after 6 h electrolysis, particle array of the cathode pallets doped 1% and 5% CaO is still compactly, the pore size of intergranular is relatively smaller. After 6 h electrolysis, intergranular pore size of the cathode pallets doped 10% CaO is obviously larger, and it's microstructure is spongy, and the intergranular pore size of the cathode pallets doped 15% CaO is much larger.

Table 2 lists the comparison of oxygen content of pure TiO_2 cathode pallets and TiO_2 cathode doped

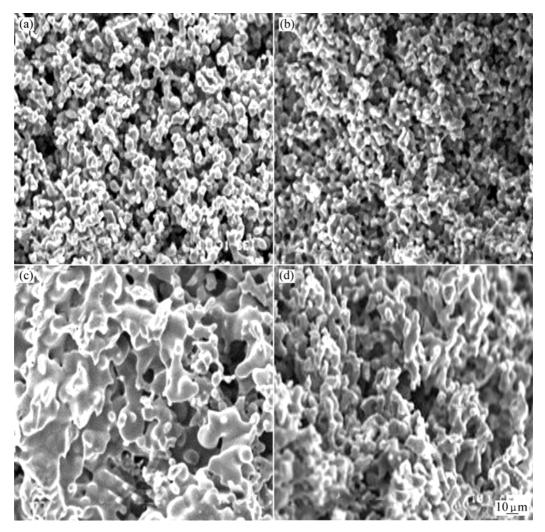


Fig.5 SEM images of TiO₂ pellets with various contents of CaO after electrolysis at 900 °C for 6 h: (a) 1%; (b) %; (c) 10%; (d) 15%

different concentration CaO after electrolysis for 6 h, oxygen content of different cathodes reduced in different degrees, deoxidation affection of the cathode doped 15% CaO is the best, residual oxygen in product is 0.39%. In consideration that all doped CaO reacts with TiO₂ to form CaTiO₃, if concentration of 15% doped CaO is conversed to concentration of CaTiO₃, its value is 36.43 %, the research of GEORGE et al show that using CaTiO₃ as raw material to prepare Ti by molten electro-deoxidation has higher electrolytic efficiency.

Table 2 Oxygen contents of reduced TiO_2 pellets with variouscontents of CaO after electrolysis at 900 °C for 6 h

Content of CaO/%	0	1	5	10	15
Oxygen content of reduced TiO ₂ /%	17.43	14.27	1.00	0.52	0.39

Fig.6 shows XRD patterns of electrolyzed TiO_2 cathode pallets doped with different concentration of CaO. From Fig.6 we know that when the doping amount is 1% and 5%, diffraction peaks of the product have

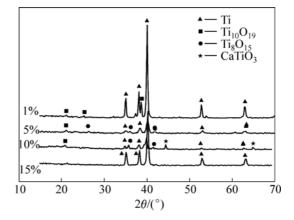


Fig.6 X-ray diffraction patterns of TiO_2 pellets with various contents of CaO after electrolysis for 6 h

some Ti protoxide and intermediate product CaTiO₃. Beside Ti and very small quantity of impurity, when doped concentration is 15%, diffraction peak of the product only has Ti. In addition, appearance of many kinds of protoxide of Ti in XRD pattern illustrates that the electro-deoxidation reaction of TiO₂ is a progressive

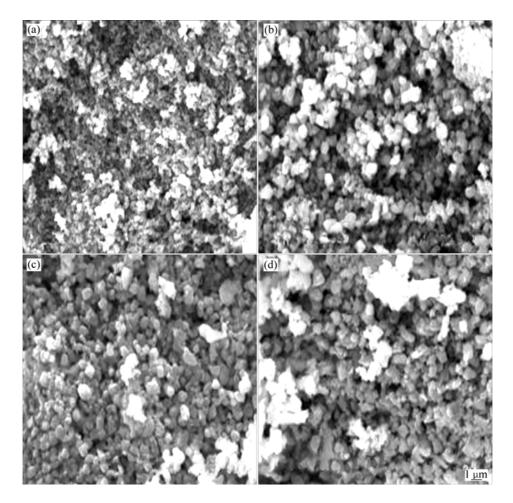


Fig.7 SEM images of TiO₂ pellets with various contents of CaCO₃ after sintering at 900 °C for 6 h: (a) 1%; (b) 5%; (c) 10%; (d) 15%

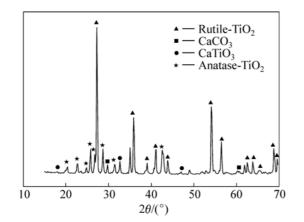


Fig.8 X-ray diffraction pattern of TiO_2 pellets with 10% CaCO₃ after sintering at 900 °C for 6 h

process. Fig.7 shows SEM images of the TiO₂ pellets doped various contents of CaCO₃ after sintering at 900 $^{\circ}$ C for 6 h. From Fig.6 we know when the concentration of CaCO₃ are 1% and 5%, the particle size of the sintered cathode is almost the same, when the content of CaO is 10% and 15%, the particle size of the sintered cathode grows larger. This is because the dopant CaCO₃

decomposes to CaO and CO₂ when it is sintered a 900 $^{\circ}$ C, the more CaCO₃ is doped, the more CaO is decomposed, the more new phase CaTiO₃ is formed. Formation of CaTiO₃ is verified by XRD patter (Fig.8) of sintered sample. The main phases of sintered cathode pallets are rutile-TiO₂, anatase-TiO₂, undecomposed CaCO₃ and new phase CaTiO₃.

Fig.9 shows SEM images of the TiO_2 pellets with various concentration of CaCO₃ electrolysis at 900 °C for 6 h. Fig.9 shows that after 6 h electrolysis there is little intergranular pore in the cathode pallets doped 1% and 5% CaCO₃; after 6 h electrolysis, pore size of intergranular of the cathode pallets doped 10% CaCO₃ and 5% CaCO₃ is obviously larger, and it's microstructure is spongy.

Table 3 shows the comparison of oxygen content of TiO_2 cathode doped with different participation CaCO₃ after 6 h electrolysis, oxygen content of different cathode reduces in different degrees, deoxidation affection of the cathode doped 15% CaCO₃ is the best, residual oxygen in product is not detected by EDS. If concentration of 15% doped CaCO₃ is conversed to concentration of CaTiO₃, it's value is 21.48%, relative to same doping amount of

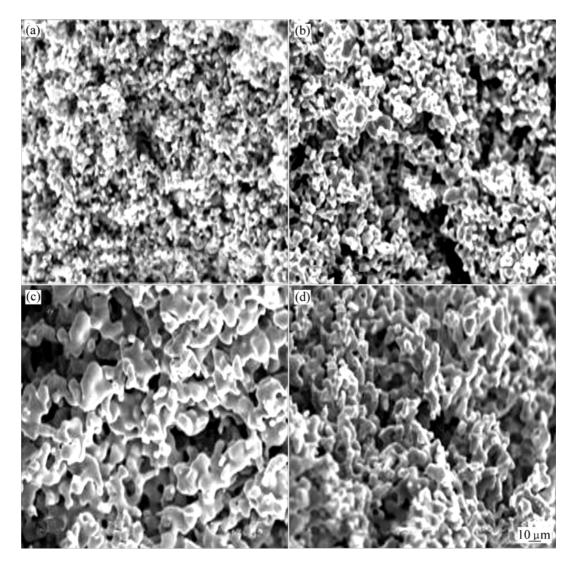


Fig.9 SEM images of TiO₂ pellets with various contents of CaCO₃ electrolysis at 900 °C for 6 h: (a) 1%; (b) 5%; (c) 10%; (d) 15%

CaO, deoxidation of 15% CaCO₃ additive is better. One of the prime reasons is that CaCO₃ is decomposed to liberating CO₂ gases, this makes intergranular pore size much larger, moreover, pylome is probably formed in cathode which is favorable to 'entry' of molten salt, the new formed phase CaO is much activated which is more prone to form CaTiO₃ with TiO₂.

Fig.10 shows the X-ray diffraction patterns of TiO_2 pellets with various content of CaCO₃ after electrolysis for 6 h. When doping amount of CaCO₃ is 1% and 5%, there are much impurity peak in product XRD patters, when doping amount of CaCO₃ is 10%. Electrolysis product are mainly Ti and some CaTiO₃, the latter content is very low, the XRD peak of CaTiO₃ is very weak. And when doping amount of CaCO₃ is 15%, electrolysis product is almost entirely Ti and there is very little impurity peak. Some kinds of sub-oxide peak in XRD pattern confirms that the reduction of TiO₂ is progressive.

Table 3 Oxygen contents of reduced TiO₂ pellets with various contents of CaCO₃ after electrolysis at 900 $^{\circ}$ C for 6 h

ontents of CaCO ₃ after electrolysis at 900°C for 0 fr					
Content of CaCO ₃ /%	0	1	5	10	15
Oxygen content of reduced TiO ₂ /%	17.43	12.20	0.51	0.22	0
1% 5% 10% 15% 10 20 3		*• *• *• *•	•— (*—)	$\begin{array}{c} \text{IiOCl}_2\\ \text{CaTiO}_3\\ \text{IiO}_2\\ \text{Ii}_8\text{O}_{15}\\ \end{array}$	0

Fig.10 X-ray diffraction patterns of TiO_2 pellets with various content of CaCO₃ after electrolysis for 6 h

5 Conclusions

1) Volume density of TiO₂ cathode varies with increment of sintering temperature. TiO₂ cathode has optimal intensity and electrochemical reaction activity when it is compacted under 100 MPa and sintered for 6 h at 900 °C, its volume density is 2.0 g/cm³, the pore size of TiO₂ cathode is about 0.5 μ m.

2) Doping CaO or CaCO₃ into TiO₂ would affect the microstructure of cathode, formation of new CaTiO₃ phase improves the electrochemical reaction activity, accelerates the reaction of electro-deoxidization.

3) When 1%-15% CaO is doped into TiO₂, the electrochemical reaction activity of the cathode is improved with the increase of the concentration of the doped CaO, the product for the cathode doped 15% CaO after electro-deoxidization for 6 h has 99.3% metal titanium.

4) Doping 1%-15 % CaCO₃ into TiO₂, the electrochemical reaction activity of the cathode is also improved with the increase of the concentration of doped CaO, the enhancement effect is preferable to that same doped concentration of CaO, the product for the cathode doped 15% CaCO₃ after electro-deoxidization for 6 h does not detect oxygen by EDS, principal product is titanium.

References

[1] CHEN G Z, FRAY D J, FARTHING T W. Direct electrochemical

reduction of titanium dioxide to titanium in molten calcium chloride[J]. Nature, 2000, 407: 361–364.

- [2] SCHWANDT C, FRAY D J. Determination of the kinetic pathway in the electrochemical reduction of titanium dioxide in molten calcium chloride[J]. Electrochimica Acta, 2005, 51: 66–76.
- [3] ALEXANDER D T L, SCHWANDT C, FRAY D J. Microstructural kinetics of phase transformations during electrochemical reduction of titanium dioxide in molten calcium chloride[J]. Acta Materialia, 2006, 54: 2933–2944.
- [4] CHEN G Z, GORDO E, FRAY D J. Direct electrolytic production of chromium powder[J]. Metal Mater Trans B, 2004, 35B: 223–233.
- [5] MA Meng, WANG Di-hua, WANG Wen-guang, HU Xiao-hong, JIN Xiao-bo, CHEN G Z. Extraction of titanium from different titania precursors by the FFC Cambridge process[J]. Journal of Alloys and Compounds, 2006, 420: 37–45.
- [6] YAN X Y, FRAY D J. Production of niobium powder by direct electrochemical reduction of solid Nb₂O₅ in a eutectic CaCl₂-NaCl melt[J]. Metall Mater Trans B, 2002, 33(B): 685–692.
- [7] YAN X Y, FRAY D J. Electrochemical studies on reduction of solid Nb₂O₅ in molten CaCl₂-NaCl eutectic(1)[J]. Journal of the Electrochemical Society, 2005, 152(1): D12–D21.
- [8] DU Ji-hong, XI Zheng-ping, LI Qing-yu, LI Zhen-xian, TANG Yong. Activity of cathode for impact in TiO₂ electrodeoxidation[J]. Journal of Rare Metals, 2007, 31(3): 336–340. (in Chinese)
- [9] DU Ji-hong, XI Zheng-ping, LI Qing-yu, LI Zhen-xian, TANG Yong. Impact of CaCO₃ doping on TiO₂ performances[J]. Rare Metal Materials and Engineering, 2007, 36(1): 96–99. (in Chinese)
- [10] DENG Li-qin, XU Qian, ZHAI Yu-chun, HUANG Zhen-qi. Preparation of niobium by direct electrochemical reduction of solid Nb₂O₅[J]. The Chinese Journal of Nonferrous Metals, 2005, 15(4): 541–545. (in Chinese)
- [11] HU Xiao-feng, XU Qian. Preparation of tantalum by electro-deoxidization in a CaCl₂-NaCl melt[J]. Acta Metallurgica Sinica, 2006, 42(3): 285–289. (in Chinese)
- [12] DU Ji-hong, XI Zheng-ping, LI Qing-yu, LI Zhen-xian, TANG Yong. Process of reduction of TiO₂ using electrodeoxidation[J]. Rare Metal Materials and Engineering, 2006, 35(7): 1045–1049. (in Chinese) (Edited by YANG Hua)