Article ID: 1003 - 6326(2000)02 - 0253 - 04

Galvanic effects on electrochemical behaviors of bare surface of 304 stainless steel[®]

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Abstract: The bare surface of 304 stainless steel is produced by the fast fracture method. The influence of the surface galvanic cell on the electroche mical behaviors of bare surface of 304 stainless steel has been investigated in H_2SO_4 solutions with different concentrations. The results show that the solution corrosivity level and the area ratio influence the surface galvanic effects caused by the inhomogeneity between the free-film surface of alloy and the passive surface. The surface galvanic effects can speed up the dissolution rate of the bare surface of the alloy and will change the electroche mical behavior of bare surface. With the increase of the area ratio between passive surface and fractured surface, the galvanic potential becomes more positive and, in the range of passive potential, both galvanic current and the peak fracture current increase evidently.

Key words: stainless steel; alloy; galvanic effect; electroche mistry

Document code: A

1 INTRODUCTION

There are many ways to produce fresh bare surface of metals at a certain speed by the dynamic methods $[1^{-9}]$ and the fast fracture method is one of these. The electrochemical information about the bare surface in corrosive solution can be obtained by this method to study the repassivation kinetics of bare metals.

When the surface film of passive metals is removed in the electrolyte, anodic dissolution, passivation and/or repassivation processes instantly carry out on the exposed bare surface^[10,11]. The repassivation ability and the repassivation kinetics of the bare surface of metals are the keys to the study of local corrosion and passivation theory.

On the surface of the metals, especially the passive alloy, in the fast fracture process in electrolyte, due to the nonexistence of the passive film on the fractured surface (bare surface), the electrochemical inhomogeneity can be induced between its bare surface and its passive surface.

The fractured surface, with lower potential, can be called anode. In contrast, the passive surface around is cathode with higher potential. Thus surface galvanic cell comes into being between the cathode area and anode area on the surface of same alloy^[12]. In general, the area of fracture surface is much less than that of the passive surface around. So the area effect of the galvanic cell will have an important influence on the electroche mical behaviors of the bare sur

face of alloy. So, the study to the electroche mical behaviors of alloy in the fast fracture process has both theoretical and practical importance.

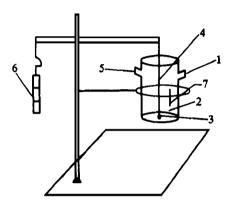
2 EXPERI MENTAL

In this article, the galvanic effect on the electroche mical behaviors of the bare surface of 304 stainless steel in the fast fracture process was investigated respectively at the open circuit potential and under the applied potential. In experiment, the area of the fractured surface was defined as $S_{\rm A}$, the area of the coupled passive surface was called $S_{\rm C}$.

The wire of 304 stainless steel (d = 0.3 mm) was used as electrode, its che mical content(mass fraction, %) was: 18.160 Cr, 8.620 Ni, 0.027 S, 0.026 P, $1.240\,Mn\,,~0.510\,Si\,,~0.052\,C$ and Fe balance. The experimental solutions were 0.1 mol/L, 0.2 mol/L, $0.5 \text{ mol/} L H_2 S O_4$. Before the experiment, a notch was prepared as the preformed crack for the wire of stainless steel. Then the part of sample A, which submerges into solution, was insulated using glue(Nanda 704) except for the preformed crack. Sample C prepared by the same material, with changeable surface area, was used as the galvanic couple to sample A in parallel. Their distance was 1 cm. In applied potential fast fracture experiment, sample A and C were electrically short-circuited by a wire. Fig.1 is the sche matic of the fast fracture apparatus. Under the rapid action of the lever, sample A fractured at the notch instantly under certain load.

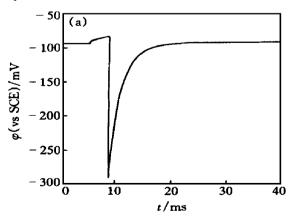
The maximum galvanic potential between sample A and C during the fracture process was defined as $\varphi_{\rm Lg}$ and the maximum galvanic current was called $I_{\rm Lg}$. The fractured galvanic current density $J_{\rm Lg}$ can be obtained when the area of the fractured surface divides $I_{\rm Lg}$. In applied potential experiment, the peak value of the fractured current recorded was $I_{\rm peak}$.

The CR-3 multifunction corrosion measuring equipment was used in experiment to monitor the change of $\varphi_{\rm Lg}$ and $I_{\rm Lg}$. PS-I potentialstat was used to apply potential on the specimen and record the $I_{\rm peak}$ in the fast fracture process . The data was collected by the AD/DA interface and corresponding electrochemical measuring software .



 $\label{eq:Fig.1} \begin{array}{ll} \textbf{Fig.1} & \textbf{Sche matic of testing apparatus} \\ 1 - Pt \ ; \ 2 - H_2 S O_4 \ solution \ ; \\ 3 - Plug \ ; \ 4 - Working \ electrode \ (Sample \ A) \ ; \\ 5 - Reference \ electrode \ (SCE) \ ; \ 6 - Load \ ; \ 7 - Sample \ C \end{array}$

Fig.2 shows the typical $\varphi-t$, I-t curves recorded in the experiment. It is seen from the curves that there are peaks of the potential and current in the $\varphi-t$ and I-t curves during the fast fracture process of 304 stainless steel in $\rm H_2SO_4$ solutions. These peak values are the important electroche mical parameters of the bare surface of alloy. At the same time, the recorded $\varphi-t$, I-t curves represent the process of passive film's removal, dissolution and the subsequent repassivation occurred on the fresh surface, and



the setup of new dynamic equilibrium on the surface of 304 stainless steel. Thus the change of $\varphi_{\rm Lg}$, $J_{\rm Lg}$ and $I_{\rm peak}$ represents the electrochemical behaviors of 304 stainless steel bare surface.

3 RESULTS AND DISCUSSION

The bare surface of 304 stainless steel is produced in the fast fracture process. Because the potential at the interface between the bare surface and solution is lower than that at the interface between the passive film surface and solution^[12], there exists an electroche mical inho mogeneity for the same material due to the different surface states, which results in the appearance of surface galvanic cell. Fig. 3 and Fig .4 show the relation between the fractured galvanic potential and the logarithm area ratio (S_C/S_A) and between the logarith m fractured galvanic current density and the logarithm area ratio (S_C/S_A) in H_2SO_4 solutions with different concentrations. The curves indicate that φ_{Lg} is directly in proportion to $\lg(S_C/$ $S_{\rm A}$) and, on the other hand, the relationship between $lg(J_{Lg})$ and $lg(S_C/S_A)$ is also linear. As increasing the area ratio($S_{\rm C}/$ $S_{\rm A}$) , $\varphi_{\rm Lg}$ approaches to the steady potential of sample C, and the anodic current density grows evidently, which leads to speed up the dissolution rate of bare surface. The results obtained above are similar to that, deduced by Mansfeld[12], about the relationship between the galvanic current and the area ratio coupled as well as between the galvanic potential and the area ratio for the galvanic couple of different metals. It proves that the galvanic effect can exist on the surfaces, with different electroche mical states, of the same material, and this kind of galvanic effect is similar to that between two different metals. So the existence of surface galvanic cell must have great effect on the electrochemical behaviors of the bare surface of alloy.

Seen from Fig .3 and Fig .4, it is obvious that the effect of area on the surface galvanic cell is the weakest in 0.1 mol/ L $H_2S\,O_4$, the strongest in 0.2

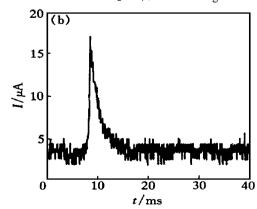


Fig.2 Typical variation of potential φ and current I with time during surface breaking process

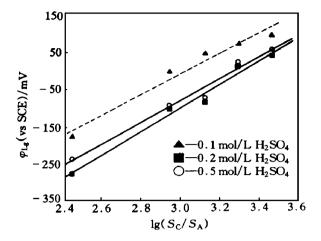


Fig.3 Relationship of φ_{Lg} — $lg(S_C/S_A)$ of 304 stainless steel in H_2SO_4 solutions with different concentrations

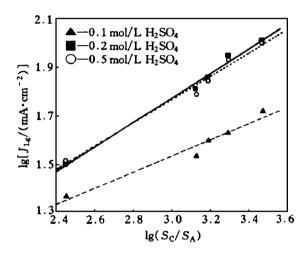


Fig.4 Relationship of $\lg(J_{\rm Lg})$ — $\lg(S_{\rm C}/S_{\rm A})$ of 304 stainless steel in $\rm H_2SO_4$ solutions with different concentrations

mol/L $\rm H_2SO_4$. By comparing the polarization curves of 304 stainless steel in the $\rm H_2SO_4$ solutions with three different concentrations, it is shown that 304 stainless steel has the greatest corrosion current density in 0.2 mol/L $\rm H_2SO_4$, the lowest that in 0.1 mol/L $\rm H_2SO_4$. So it can be concluded that the extent of galvanic effect is related to the corrosivity of the experimental solutions.

In applied potential experiment, the potential of the fractured surface of alloy will deviate from the potential applied instantly when the fracture occurred. At this moment, the galvanic cell forms between the fractured surface and the passive surface around, which has influence on the electroche mical behaviors of the fractured alloy. Fig.5 shows the variation of $I_{\rm peak}$ of 304 stainless steel with $S_{\rm C}/S_{\rm A}$ under the applied passive potential 200 mV. It can be seen that, in all the $\rm H_2SO_4$ solutions of different concentration, as increasing $S_{\rm C}/S_{\rm A}$, $I_{\rm peak}$ grows and anodic dissolutions of disso

tion rate becomes faster. But the extents of galvanic effect in these solutions are much different. The degrees of galvanic effect (area ratio effect) in 0.2 mol/L $\rm H_2S\,O_4$ and 0.5 mol/L $\rm H_2S\,O_4$ are evidently higher than that in 0.1 mol/L $\rm H_2S\,O_4$. This indicates that, for the passive alloy, the stronger the aggressiveness of the solutions, the more obvious the galvanic effect with the change of the area ratio $S_{\rm C}/S_{\rm A}$.

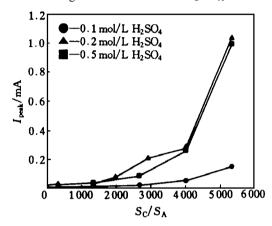


Fig.5 Relationship of $I_{\rm peak} - S_{\rm C}/S_{\rm A}$ of 304 stainless steel in H₂SO₄ solutions with different concentrations under applied potential 200 m V(vs SCE)

Fig. 6 shows that the relationship between the breaking current $I_{\rm peak}$ of 304 stainless steel in 0.2 mol/L $\rm H_2SO_4$ solution and the applied potential under the condition of certain area ratio. It can be seen that $I_{\rm peak}$ grows slowly as the increase of $S_{\rm C}/S_{\rm A}$, which indicates that, in the passive range, the potential has a little effect on the peak current. Comparing with the result under $S_{\rm C}/S_{\rm A}=0$, the bigger the area ratio, the greater the fractured current, from which it is learnt that the galvanic cell formed due to the surface electrochemical inhomogeneity has important

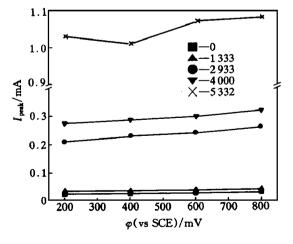


Fig.6 Relationship between $I_{\rm peak}$ and applied potential φ of 304 stainless steel in 0.2 mol/ L $\rm H_2SO_4$ solution with different values of $S_{\rm C}/S_{\rm A}$

influence on the fractured current of alloy in the fast fracture process.

From the results above, the galvanic effect, formed on the 304 stainless steel surfaces with different surface states, affects greatly the electrochemical behavior during the fracture process of 304 stainless steel. Because the fractured surface area is so small, the area ratio of $S_{\rm C}$ to $S_{\rm A}$ can reach a very high value. So the area effect on the experimental results can not be ignored when studying the electrochemical kinetics of the alloy's bare surface which is produced through fast fracture method.

The analysis above demonstrates that there exists surface galvanic effect between the fractured surface and the passive surface around during the fast fracture process. The existence of the area ratio effect evidently influences the electrochemical kinetics of the bare surface of alloy. This leads to the increase of dissolution rate of alloy.

4 CONCLUSIONS

- 1) There is surface galvanic cell in the fast fracture process, due to the electroche mical inhomogeneity between the bare surface of alloy and the passive surface of alloy. And the effect of area ratio (passive surface to fractured surface) on the electroche mical galvanic behavior is particularly important and evident.
- 2) The surface galvanic effect has obvious effect on the electrochemical behaviors of the bare surface of 304 stainless steel. It speeds up the dissolution rate of the bare surface.
- 3) With the increase of $S_{\rm C}/S_{\rm A}$, $\varphi_{\rm Lg}$ moves towards positive direction and $I_{\rm Lg}$ increases substantially.
- 4) Under the passive potential of 200 mV, the fractured current $I_{\rm peak}$ increases with $S_{\rm C}/S_{\rm A}$. The applied potential doesn't have strong effect on the $I_{\rm peak}$.
- 5) In passive system, the stronger the corrosivity of the corrosion medium, the more obvious the sur-

face area effect on the galvanic behavior.

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(Edited by YUAN Sai-qian)