

## Influence of $\text{Yb}_2\text{O}_3$ addition on microstructure and corrosion resistance of $10\text{Cu}/(10\text{NiO}-\text{NiFe}_2\text{O}_4)$ cermets

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**Abstract:**  $10\text{Cu}/(10\text{NiO}-\text{NiFe}_2\text{O}_4)$  cermets doped with  $\text{Yb}_2\text{O}_3$  were prepared by conventional powder metallurgy technique. The effects of  $\text{Yb}_2\text{O}_3$  content and sintering temperature on the relative density, phase composition, microstructure of the sintered cermets and the corrosion resistance to  $\text{Na}_3\text{AlF}_6\text{-Al}_2\text{O}_3$  melts were investigated by sintered density test, XRD analysis and SEM.  $\text{YbFeO}_3$  phase, which distributes in the ceramics grain boundary as particles or film, is produced by the reaction between  $\text{Yb}_2\text{O}_3$  and ceramics. The addition of  $\text{Yb}_2\text{O}_3$  accelerates the sintering process of ceramics matrix, eliminates pores in the boundary and results in coarsened crystalline grain. The relative density of the cermets with about 1% (mass fraction)  $\text{Yb}_2\text{O}_3$  sintered at  $1275^\circ\text{C}$  increases to above 95%. Addition of about 1.0%  $\text{Yb}_2\text{O}_3$  can inhibit obviously the corrosion of  $\text{NiFe}_2\text{O}_4$  grain boundary and Cu phase in  $\text{Na}_3\text{AlF}_6\text{-Al}_2\text{O}_3$  melts.

**Key words:**  $10\text{Cu}/10\text{NiO}-\text{NiFe}_2\text{O}_4$  cermets;  $\text{Yb}_2\text{O}_3$  addition; sintering densification; microstructure; corrosion resistance; aluminum electrolysis

### 1 Introduction

The use of inert anodes for replacement of consumable carbon anodes in Hall-Heroult aluminum electrolysis cells has been a technical and commercial goal for many decades. In the present process, when consumable carbon anodes are used, the anode product is  $\text{CO}_2$ . With an inert anode, the anode product is  $\text{O}_2$ . The basic requirements for an inert anode are: 1) low corrosion rate in molten cryolite, 2) ability to produce commercially pure aluminum, 3) good electric conductivity, 4) being thermally stable up to electrolysis temperature, 5) adequate resistance to thermal shock and 6) being economically feasible[1]. Recently, the materials studied as inert anodes mainly concentrated on the alloy and the cermets.  $\text{NiFe}_2\text{O}_4$ -based cermets, which possess not only low solubility of ceramic to molten cryolite, but also high electrical conductivity of metal, are very promising as inert anode for aluminum electrolysis[2–3]. However, there are selective solubility of metal phase and grain-boundary corrosion of the  $\text{NiFe}_2\text{O}_4$ -based cermets in molten cryolite, resulting in

the penetration of electrolyte into the inert anode, the reduction of service life and the increase of impurity in metal Al[4]. Sintering additive is effective to improve the microstructure and properties such as relative density, electric conductivity and corrosion resistance to cryolite melts[5–6]. JIAO et al[7] reported a gradual improvement in sintered density for nickel ferrite products by addition of  $\text{TiO}_2$  up to 1%. LIU et al[8] pointed out that addition of  $\text{MnO}_2$  increased the sintering density of nickel ferrite and improved the corrosion resistance to molten cryolite. The refined grains and improved thermal shock resistance were also obtained in cermet samples containing  $\text{MnO}_2$ . TIAN et al[9] have gained that the addition of  $\text{SnO}_2$  decreased the activation energy and improved the electric conductivity of  $\text{NiFe}_2\text{O}_4$ -based cermets. ZHANG et al[10] found that proper addition of  $\text{Sm}_2\text{O}_3$  into  $\text{FeAl}_2\text{O}_4$ -based cermet leads to bigger crystal of  $\text{FeAl}_2\text{O}_4$  phase, higher anti-oxidation and electric conductivity. XI et al[11] found that the addition of  $\text{V}_2\text{O}_5$  into  $\text{NiFe}_2\text{O}_4$ -based cermets is beneficial to improving obviously corrosion resistance to molten cryolite. In the  $x\text{Cu}/\text{NiFe}_2\text{O}_4$  cermets, proper  $\text{Y}_2\text{O}_3$  could improve the wettability between Cu

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and  $\text{NiFe}_2\text{O}_4$  phase and enhance the bending strength and the thermal shock resistance[12]. However, those additives mentioned above could not improve the microstructure and properties at the meantime.

In this study, the  $10\text{Cu}/(10\text{NiO}-90\text{NiFe}_2\text{O}_4)$  cermets doped with  $\text{Yb}_2\text{O}_3$  were prepared. The influence of  $\text{Yb}_2\text{O}_3$  addition on phase composition, microstructure, relative density and corrosion resistance to molten cryolite was investigated.

## 2 Experimental

### 2.1 Preparation of $10\text{Cu}/(10\text{NiO}-\text{NiFe}_2\text{O}_4)$ cermets doped with $\text{Yb}_2\text{O}_3$

$10\text{Cu}/10\text{NiO}-\text{NiFe}_2\text{O}_4$  cermets doped with  $\text{Yb}_2\text{O}_3$  ( $<3\%$ ) were prepared by the conventional powder metallurgy method with raw materials (reagent grade) including copper powder,  $\text{Fe}_2\text{O}_3$ ,  $\text{NiO}$  and  $\text{Yb}_2\text{O}_3$ . The mixture of  $\text{Fe}_2\text{O}_3$  and  $\text{NiO}$  in the molar ratio of 1.35:1 was calcined in a muffle furnace at  $1\,200\text{ }^\circ\text{C}$  for 6 h in air and then the  $10\text{NiO}-\text{NiFe}_2\text{O}_4$  ceramic powder was obtained. X-ray diffraction result of  $10\text{NiO}-\text{NiFe}_2\text{O}_4$  ceramics is shown in Fig.1. The synthesized  $10\text{NiO}-\text{NiFe}_2\text{O}_4$  ceramic powder, Cu and  $\text{Yb}_2\text{O}_3$  powder were ground in the media containing dispersant and adhesive. The dried mixture was cold pressed into cylindrical blocks ( $d20\text{ mm}\times40\text{ mm}$ ) at the pressure of 200 MPa. Then the samples were sintered at  $1\,250\text{--}1\,325\text{ }^\circ\text{C}$  for 4 h in nitrogen atmosphere.

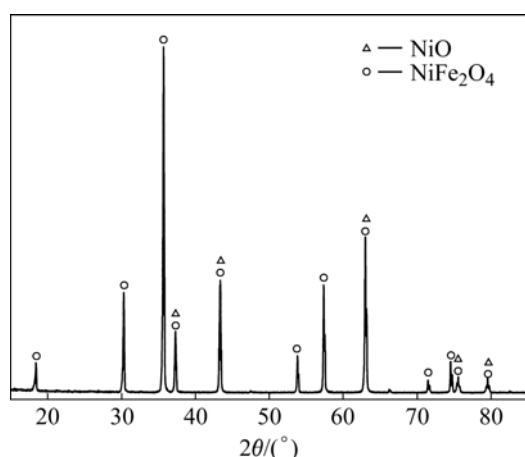


Fig.1 XRD patterns of  $10\text{NiO}-\text{NiFe}_2\text{O}_4$  ceramics

### 2.2 Characterization

The phase compositions were identified by X-ray diffraction analysis using Philips PW1390 X-ray diffractometer with  $\text{Cu K}_\alpha$  radiation. Microstructure was analyzed by scanning electron microscope (SEM) (JSM-6360LV) and energy dispersive spectrometer (EDS) connected to the SEM. Bulk density and relative density were tested according to the Archimedes' method.

### 2.3 Electrolysis tests

The electrolyte was made up of  $\text{Na}_3\text{AlF}_6$  and  $\text{AlF}_3$  of reagent grade,  $\text{CaF}_2$  and  $\text{Al}_2\text{O}_3$  of technical grade; the CR ( $\text{NaF}/\text{AlF}_3$  molar ratio) was 2.3; and the concentrations of  $\text{CaF}_2$  and  $\text{Al}_2\text{O}_3$  are both kept to be about 5%(mass fraction). All were added prior to electrolysis. The temperature was controlled at  $960\text{ }^\circ\text{C}$  with superheat of  $10\text{ }^\circ\text{C}$ .

The sketch map of the experimental cell is shown in Fig.2. Alumina sleeve was set in the graphite crucible. About 400 g electrolyte was contained in the graphite crucible. The cell with the anode was placed in a vertical furnace and heated to the desired temperature and kept for 2 h before the anode was immersed into the electrolyte. The anode was immersed into the electrolyte by 20 mm. The current density of anode bottom was  $1\text{ A}/\text{cm}^2$  and the current was kept constant throughout the experiment, which lasted 10 h. By calculating the consumption rate of  $\text{Al}_2\text{O}_3$  when cathode current efficiency was 85%, the concentration of  $\text{Al}_2\text{O}_3$  in the bath decreased by 2.0% if  $\text{Al}_2\text{O}_3$  was not added during the test. Moreover, alumina sleeve also contributes to keep the concentration of  $\text{Al}_2\text{O}_3$ . So,  $\text{Al}_2\text{O}_3$  was not added into the bath during test. After electrolysis, the anode was raised out of the melt to prevent reduction of the anode material by dissolved metal aluminum. The cell was left to cool naturally with the anode resting above the electrolyte. The anodes tested were sectioned, polished, and analyzed by SEM (JSM-6360LV). The electrolyte after electrolysis test was analyzed by X-ray fluorescence spectrum (Philips 8424 TW2424) and the concentrations of Ni, Fe and Cu in electrolyte and

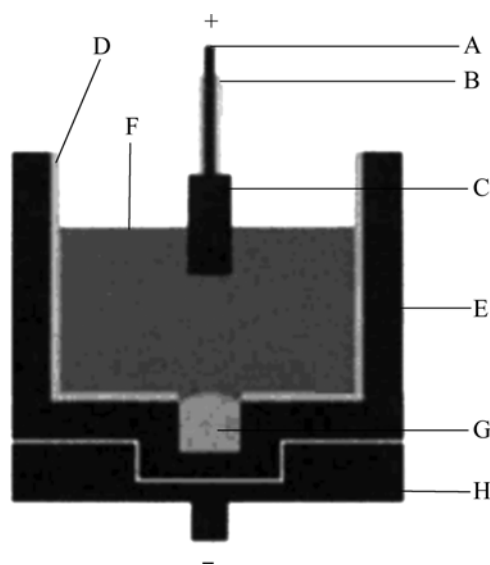


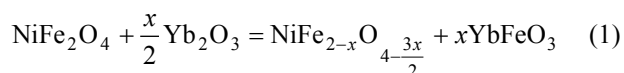
Fig.2 Sketch map of electrolysis experimental cell: A—Stainless steel anode rod; B— $\text{Al}_2\text{O}_3$  sleeve; C—Cermet inert anode; D— $\text{Al}_2\text{O}_3$  liner; E—Graphite crucible; F—Electrolyte; G—Metal aluminum; H—Graphite mechanical support

cathode aluminum were obtained, respectively.

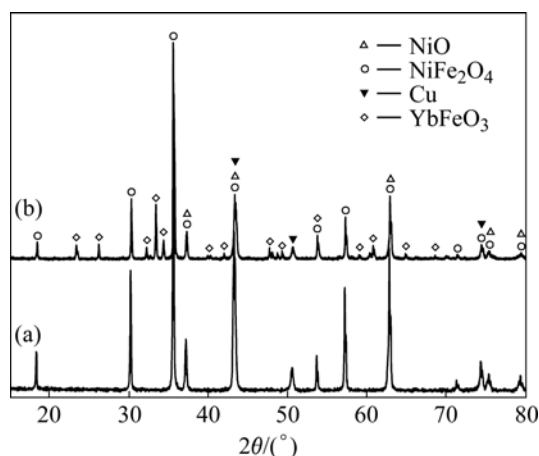
### 3 Results and discussion

#### 3.1 Phase composition and microstructure

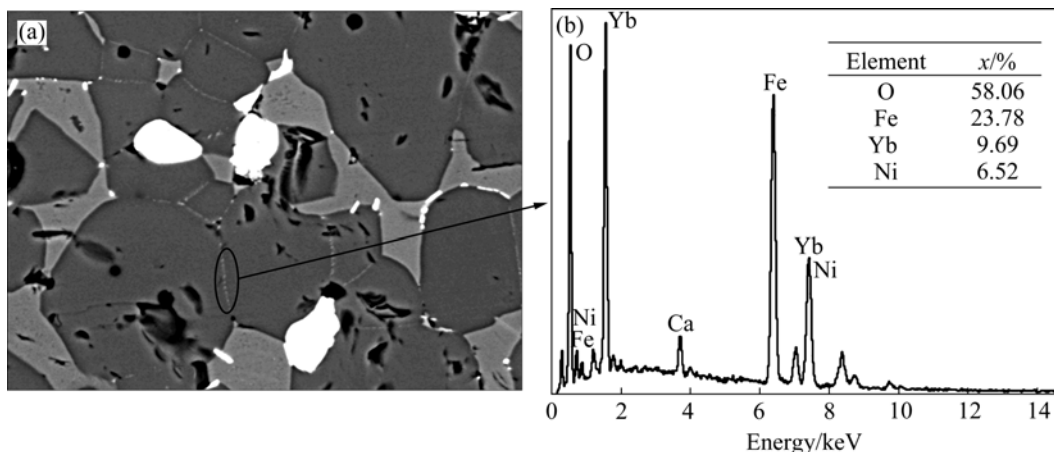
The XRD patterns of 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) samples doped with Yb<sub>2</sub>O<sub>3</sub> and sintered at 1 275 °C are shown in Fig.3. It can be seen from Fig.3 that only NiFe<sub>2</sub>O<sub>4</sub>, NiO and Cu phases exist in the 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermet samples. Besides the NiFe<sub>2</sub>O<sub>4</sub>, NiO and Cu phases, a new phase YbFeO<sub>3</sub> appeared in the 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermet doped with 2.0% Yb<sub>2</sub>O<sub>3</sub>. However, Yb<sub>2</sub>O<sub>3</sub> was not detected in the cermets doped with Yb<sub>2</sub>O<sub>3</sub>. This phenomenon indicates that the Yb<sub>2</sub>O<sub>3</sub> in the sample has completely become YbFeO<sub>3</sub> in the sintering process. According to the phase diagram of Fe<sub>2</sub>O<sub>3</sub>-Yb<sub>2</sub>O<sub>3</sub> system[13], the YbFeO<sub>3</sub> was possibly generated by the reaction:



This reaction and formation of YbFeO<sub>3</sub> may lead to some Fe and O vacancies in the NiFe<sub>2</sub>O<sub>4</sub> lattice.



**Fig.3** XRD patterns of 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) samples sintered at 1 275 °C: (a) Without Yb<sub>2</sub>O<sub>3</sub>; (b) With 2.0% Yb<sub>2</sub>O<sub>3</sub>



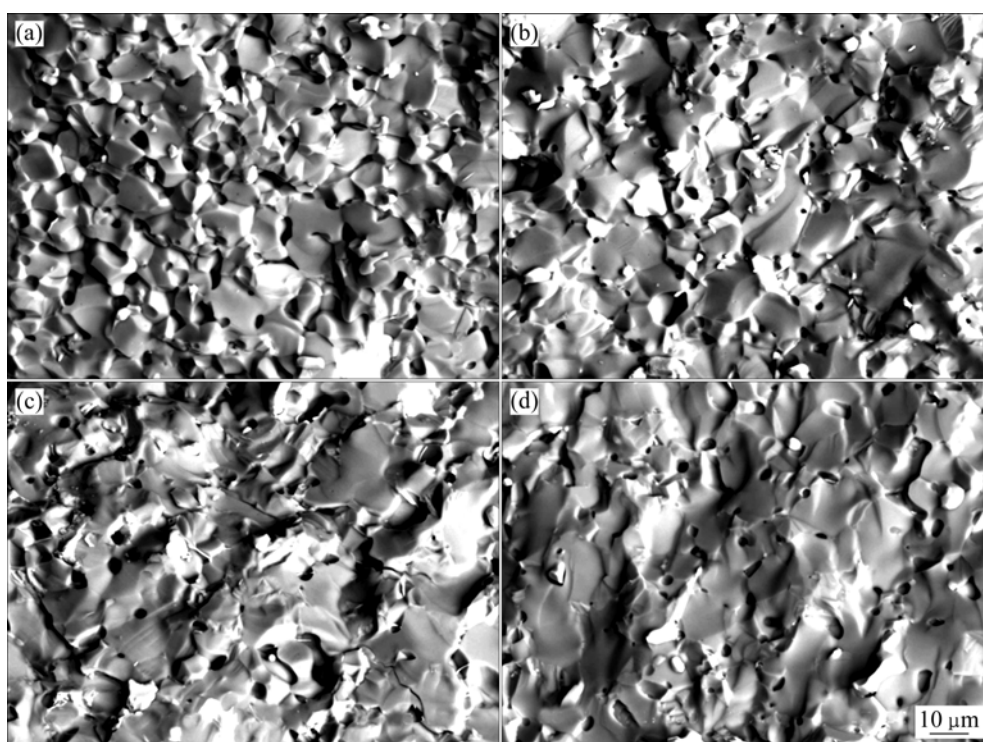
**Fig.4** SEM photograph (a) and EDS (b) of 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets doped with 2.0% Yb<sub>2</sub>O<sub>3</sub>

The SEM photograph of the polished surface of 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermet doped with 2.0% Yb<sub>2</sub>O<sub>3</sub> is shown in Fig.4. It can be seen that the YbFeO<sub>3</sub> phase distributed mainly along the NiFe<sub>2</sub>O<sub>4</sub> grain boundary, which may be helpful for purification and strengthening of NiFe<sub>2</sub>O<sub>4</sub> grain boundary.

The SEM photographs of the cermets sintered at 1 275 °C and doped with different contents of Yb<sub>2</sub>O<sub>3</sub> are shown in Fig.5. It can be seen from Fig.5(a) that there are many pores in the sample without addition of Yb<sub>2</sub>O<sub>3</sub> and the grain size is small, being 8–10 μm. As shown in Fig.5(b)–(d), the number of the pores in the sample decreased obviously and the grain size became large with the addition of Yb<sub>2</sub>O<sub>3</sub> into 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets. In addition, tighter combination between the NiFe<sub>2</sub>O<sub>4</sub> grains was achieved by addition of Yb<sub>2</sub>O<sub>3</sub> to cermets due to removing of pores. This indicates that proper addition of Yb<sub>2</sub>O<sub>3</sub> into 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets can accelerate the sintering process and improve properties of the cermets.

#### 3.2 Effect of Yb<sub>2</sub>O<sub>3</sub> addition on relative density of 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets

The relative densities of 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets doped with different contents of Yb<sub>2</sub>O<sub>3</sub> sintered at 1 250–1 300 °C are listed in Table 1. It can be seen from Table 1 that the relative density of 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets sintered at 1 250 °C increased obviously with the addition of 0.5%–3% Yb<sub>2</sub>O<sub>3</sub>, which indicates that the presence of Yb<sub>2</sub>O<sub>3</sub> in 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets could promote the sintering process. When the sintering temperature was 1 300 °C, there was little difference in relative density with the increase of Yb<sub>2</sub>O<sub>3</sub> content. It is possible that the presence of Fe and O vacancies in the 10Cu/10NiO-NiFe<sub>2</sub>O<sub>4</sub> cermets, resulting from the formation of YbFeO<sub>3</sub>, leads to more point defects in the material,



**Fig.5** SEM fractographs of samples with different contents of  $\text{Yb}_2\text{O}_3$ : (a) Without  $\text{Yb}_2\text{O}_3$ ; (b) 0.5%  $\text{Yb}_2\text{O}_3$ ; (c) 1.0%  $\text{Yb}_2\text{O}_3$ ; (d) 2.0%  $\text{Yb}_2\text{O}_3$

**Table 1** Relative densities of 10Cu/10NiO-NiFe<sub>2</sub>O<sub>4</sub> cermets with different contents of  $\text{Yb}_2\text{O}_3$

Content of $\text{Yb}_2\text{O}_3/\%$	Relative density/%		
	1 250 °C	1 275 °C	1 300 °C
0	86.92	93.16	94.40
0.5	89.75	95.58	94.89
1.0	90.62	94.85	95.23
1.5	88.42	95.72	94.74
2.0	89.99	95.42	95.06
3.0	91.82	94.76	93.51

helps the transportation of oxygen and Fe ions, and accelerates atomic diffusion and mass transportation, which finally promotes the sintering densification of 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets[14].

### 3.3 Corrosion resistance

LAI et al[15] pointed out that it took 5–6 h for the concentration of Cu, Ni or Fe in electrolyte during electrolysis test to reach steady state, which was about the solubility of Cu, Ni or Fe in the cryolite melts. Therefore, all electrolysis experiments lasted 10 h in this study. The concentration and mass change of Cu, Ni and Fe in electrolyte after electrolysis test are listed in Table 2. The results listed in Table 2 show that the steady-state concentration of Cu in the bath decreased from 0.0211% to 0.0178% as the  $\text{Yb}_2\text{O}_3$  content changes

from 0 to 1.0%, which is a little lower than its solubility (0.0226%) reported in the previous study[16]. The  $\text{Yb}_2\text{O}_3$  addition has little effect on the concentration of Ni and Fe in the cryolite melts. The concentration of Ni was all close to the solubility (0.0125%) by LAI et al[16] (bath ratio 1.15, melt with 5%  $\text{Al}_2\text{O}_3$ , 965 °C) and the concentration of Fe was all far lower than the solubility(0.0580%) mentioned by DEYOUNG[17] (bath ratio 1.1, melt with 6.5%  $\text{Al}_2\text{O}_3$ , 1 000 °C). The total mass change of Cu, Fe and Ni in the electrolyte decreased from 0.90 to 0.774 g as the  $\text{Yb}_2\text{O}_3$  content increased from 0 to 1.0%.

The content and mass change of Cu, Ni and Fe in metal aluminum after electrolysis test are listed in Table 3. It is obvious that the contents of impurities Cu, Ni and Fe in metal aluminum all decreased when 0.5%  $\text{Yb}_2\text{O}_3$  was added into the 10Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets. More  $\text{Yb}_2\text{O}_3$  addition has little effect on the impurities Cu, Ni and Fe in metal aluminum after electrolysis. This should be because the higher density of cermets inert anode by addition of  $\text{Yb}_2\text{O}_3$  and the  $\text{YbFeO}_3$  phases, which distribute in the  $\text{NiFe}_2\text{O}_4$  grain boundary, could prevent the penetration of cryolite melt into the inert anode. In addition, the molar ratio of Fe to Ni in metal aluminum is 4.14–14.8, which is much more than that in cermets. That is to say, the transferring velocity of Fe from electrolyte to molten aluminum is more than that of Ni. This phenomenon is same with the results by

TIAN et al[4].

To obtain more information about the effect of  $\text{Yb}_2\text{O}_3$  addition on the corrosion resistance to cryolite melt, the cermet inert anodes were sectioned, mounted and polished after electrolysis tests. The SEM photographs of anodes cross-section after electrolysis test are shown in Fig.6. According to Fig.6(a), the metal phase Cu is leached preferentially by cryolite melt, and there are many obvious holes and pores in the corrosion layer, which is about 200  $\mu\text{m}$  after electrolysis test for 10

h. From Figs.6(b)and (c), it is found obviously that only less than 20  $\mu\text{m}$  in the exterior surface of anodes was corroded when the  $10\text{Cu}/(10\text{NiO-NiFe}_2\text{O}_4)$  cermets were added with 0.5% or 1%  $\text{Yb}_2\text{O}_3$ . By comparing Figs.6(a) with (b) and (c), the addition of  $\text{Yb}_2\text{O}_3$  into  $10\text{Cu}/(10\text{NiO-NiFe}_2\text{O}_4)$  cermets has improved obviously the corrosion resistance of the inert anode to the cryolite melts. This result corresponds with the content change of Cu, Ni and Fe in electrolyte and metal aluminum after electrolysis test.

**Table 2** Concentration and mass change of impurities in electrolyte after electrolysis test

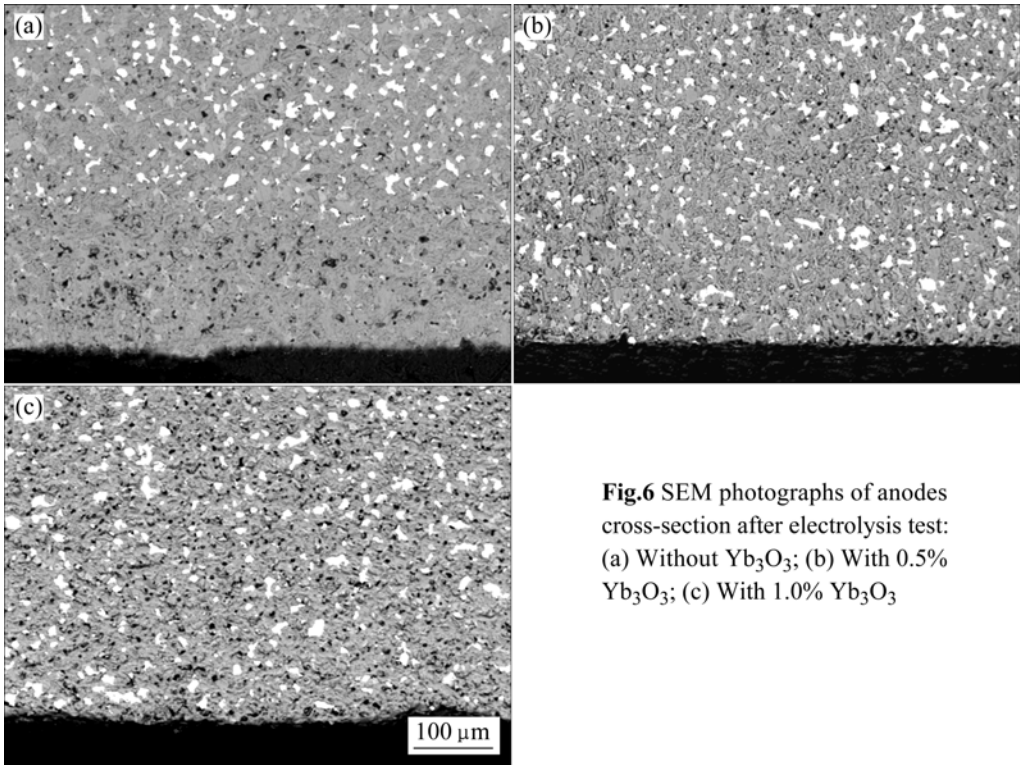
$\text{Yb}_2\text{O}_3$ content/%	$w(\text{Cu})/\%$	$w(\text{Ni})/\%$	$w(\text{Fe})/\%$	$\Delta(w(\text{Cu})+w(\text{Fe})+w(\text{Ni}))/\%$	$\Delta(w(\text{Cu})+w(\text{Fe})+w(\text{Ni}))/\text{g}$
0	0.021 1	0.012 1	0.027 0	0.030 0	0.90
0.5	0.020 7	0.012 1	0.026 9	0.029 5	0.885
1.0	0.017 8	0.0117	0.026 5	0.025 8	0.774
1.5	0.018 1	0.012 3	0.027 7	0.027 9	0.837

Note: The concentrations of Cu, Fe and Ni in electrolyte before electrolysis were  $6.6 \times 10^{-5}$ ,  $1.95 \times 10^{-4}$  and  $4.1 \times 10^{-5}$ , respectively

**Table 3** Content and mass change of impurities in metal aluminum after electrolysis test

$\text{Yb}_2\text{O}_3$ content/%	$m/\text{g}$		$w(\text{Cu})/\%$	$w(\text{Fe})/\%$	$w(\text{Ni})/\%$	$\Delta(m(\text{Cu})+m(\text{Fe})+m(\text{Ni}))/\text{g}$
	Before test	After test				
0	67.57	73.66	0.093 5	1.088 1	0.049 5	0.292
0.5	67.84	74.5	0.076 0	1.071 3	0.042 0	0.161
1.0	68.4	76.13	0.072 0	1.070 9	0.043 0	0.168
1.5	67.92	74.62	0.073 5	1.072 0	0.044 0	0.166

Note: Contents of Cu, Fe and Ni in metal aluminum were  $5.8 \times 10^{-5}$ ,  $1.68 \times 10^{-3}$  and  $3.6 \times 10^{-5}$ , respectively



**Fig.6** SEM photographs of anodes cross-section after electrolysis test: (a) Without  $\text{Yb}_3\text{O}_3$ ; (b) With 0.5%  $\text{Yb}_3\text{O}_3$ ; (c) With 1.0%  $\text{Yb}_3\text{O}_3$

## 4 Conclusions

1)  $\text{YbFeO}_3$  phase, which distributes in the ceramics grain boundary as particles or film, is produced by the reaction between  $\text{Yb}_2\text{O}_3$  and ceramics when  $\text{Yb}_2\text{O}_3$  is added to the  $10\text{Cu}/(10\text{NiO}-\text{NiFe}_2\text{O}_4)$  cermets.

2) The addition of  $\text{Yb}_2\text{O}_3$  accelerates the sintering process of ceramics matrix, eliminates pores in the boundary and results in coarsened crystalline grain. The relative density of the cermets with about 1%  $\text{Yb}_2\text{O}_3$  sintered at  $1\,275\text{ }^\circ\text{C}$  increases to 95%.

3) Proper addition of  $\text{Yb}_2\text{O}_3$  and  $\text{YbFeO}_3$  phase can inhibit the penetration of cryolite melts into the  $\text{NiFe}_2\text{O}_4$  grain boundary, which improves the corrosion resistance of  $10\text{Cu}/(10\text{NiO}-\text{NiFe}_2\text{O}_4)$  cermets inert anode to  $\text{Na}_3\text{AlF}_6\text{-Al}_2\text{O}_3$  melts.

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