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## Transformations and phase relations in Nb Ti Si ternary system at $1373 \sim 1473 \, \mathrm{K}^{\odot}$

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[Abstract] The isothermal sections of the Nb Tr Si ternary system at 1473 K and 1373 K were determined by means of diffusion triple technique and electron microprobe analysis. By analyzing the diffusion layers in the diffusion couples, the titanium silicides and niobium silicides forming in this system were identified. The results show that no ternary compounds formed in the Nb Tr Si ternary system at the test temperatures. The phase transformations occurring on cooling from 1473 K to 1373 K were discussed.

[Key words] Nb Tr Si ternary system; diffusion triple technique; isothermal sections; phase transformations; phase relations

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#### 1 INTRODUCTION

Because of their high strength, good plasticity and corrosion resistance, especially low density and excellent fracture toughness, the titanium and its alloys have found wide applications in many fields<sup>[1]</sup>.

In the Nb Si binary system, an in situ composite can form with Nb<sub>5</sub>Si<sub>3</sub> phase as reinforcement and (Nb) phase as matrix. Even at a temperature high up to 2757 K, the Nb<sub>5</sub>Si<sub>3</sub> still has enough strength (hardness) to act as reinforcement<sup>[2]</sup>. The addition of titanium into the Nb Si system can further improve the properties of the matrix as well as the Nb Si compounds, which is due to the fact that continuous solid solution can form between niobium and titanium in a certain range of temperature and sometimes niobium and titanium can replace each other to form silicides.

Schlesinger et al<sup>[3]</sup> made a detailed evaluation on the Nb Si binary system, and found there are three compounds namely Nb<sub>3</sub>Si, Nb<sub>5</sub>Si<sub>3</sub> and NbSi<sub>2</sub> in this system. Many other researchers<sup>[4,5]</sup> have also done a lot of work on this system, respectively.

The NbTi binary system is relatively simple. Murray  $^{[6]}$  made an evaluation on it, and he believed that above  $1\,155\,\mathrm{K}$ , a continuous solid solution forms between niobium and titanium.

Murray<sup>[7]</sup>, Vahlas<sup>[8]</sup>, Seifert et al<sup>[9]</sup> made thermodynamic evaluation or optimized calculation on the TirSi binary system, and found there are five compounds namely Ti<sub>3</sub>Si, Ti<sub>5</sub>Si<sub>3</sub>, Ti<sub>5</sub>Si<sub>4</sub>, TiSi and TiSi<sub>2</sub> in this system.

The experimented data of the Nb-TrSi ternary system are very limited. Bewlay et al<sup>[10]</sup> determined the liquid face and the Nb-TrSi isothermal section of the Nb-TrSi ternary system through studying 29 al-

loys by means of directional solidification. ZHANG et al<sup>[11]</sup> made a thermodynamic evaluation on the liquid face of the Nb-Tr-Si ternary system. In this paper, the authors aim to determine isothermal sections of the Nb-Tr-Si ternary system at 1473 K and 1373 K by means of diffusion triple technology and electron microprobe analysis.

### 2 EXPERIMENTAL

The raw materials for preparing the NbTrSi diffusion triple are 99.9% purity niobium and titanium ingots and 99.99% purity silicon ingot. The dimensions of niobium, titanium and silicon ingots are 5 mm × 5 mm × 10 mm, 5 mm × 5 mm × 10 mm and 5 mm × 10 mm, respectively. The schematic diagram of construction of the NbTrSi ternary system is shown in Fig. 1. After long time annealing at 1473 K and 1373 K, the transformations and phase relations were determined by analyzing the diffusion layers using electron microprobe analysis technique.

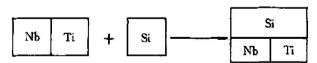


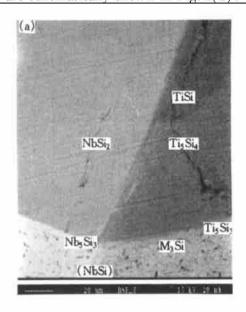
Fig. 1 Schematic diagram of construction of NbTrSi diffusion triple

### 3 RESULTS AND DISCUSSION

### 3. 1 Isothermal section of Nb Ti Si ternary system at 1473 K

Fig. 2(a) shows the back-scattered electron image of the Nb-Tr-Si diffusion triple after annealing for 360 h at 1473 K. It can be found in Fig. 2(a) that there are seven diffusion layers, of which four are Tr-

Si compounds, two are NbSi compounds and one is a compound forming between (NbTi) and Si. The characteristics of phase distribution and the phase relations are schematically shown in Fig. 2(b).



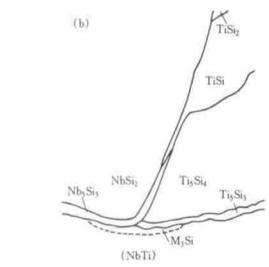


Fig. 2 Back-scattered electron image (a) and schematic diagram of phase distribution (b) of NbTrSi diffusion triple after annealing for 360 h at 1473 K

The chemical compositions of the equilibrium phases were determined by the electron microprobe analysis. Part of the typical data are listed in Table 1. The diffusion layers in the diffusion triple can be demarcated based on the experimental data. In the  $T\dot{r}$  Si side, the diffusion layers are  $TiSi_2$ , TiSi,  $Ti_5Si_4$  and  $Ti_5Si_3$  successively from silicon to titanium. In the Nb·Si side, the diffusion layers are NbSi2 and Nb<sub>5</sub>Si<sub>3</sub> successively from silicon to niobium.

In Nb-Si side, there are mainly two layers of compounds namely Nb<sub>5</sub>Si<sub>3</sub> and NbSi<sub>2</sub>. The largest solubility of titanium in Nb<sub>5</sub>Si<sub>3</sub> is 3.0% (mole fraction). While that in NbSi<sub>2</sub> is only 1.2% (mole fraction).

**Table 1** Tie lines and tie triangles of Nb-Ti-Si diffusion triple at 1473 K (mole fraction)

Tie lines	Nb	Si	Ti	Nb	Si	Тi
Ni <sub>5</sub> Si <sub>3</sub> / NbSi <sub>2</sub>	0. 634 0. 620 0. 616	0. 366 0. 371 0. 360	0 0. 009 0. 024	0. 348 0. 346 0. 343	0. 652 0. 654 0. 651	0 0 0.006
Nb <sub>5</sub> Si <sub>3</sub> /Nb(Ti)	0. 639 0. 658 0. 640	0. 331 0. 342 0. 342	0. 030 0 0. 018	0. 945 0. 990 0. 987	0.055 0.010 0.013	0 0 0
TiSi /TiSi <sub>2</sub>	0. 133 0. 096 0. 056	0. 512 0. 514 0. 495	0. 355 0. 390 0. 449	0. 110 0. 043 0. 027	0. 651 0. 650 0. 654	0. 239 0. 307 0. 319
${ m TiSi}_2 \ /  { m NbSi}_2$	0. 052 0. 052 0. 110	0. 657 0. 651 0. 651	0. 291 0. 297 0. 239	0. 338 0. 330 0. 318	0. 650 0. 661 0. 665	0. 012 0. 009 0. 027
T i <sub>5</sub> Si <sub>3</sub> / T i <sub>5</sub> Si <sub>4</sub>	0. 200 0. 166 0. 105	0. 360 0. 364 0. 354	0. 540 0. 470 0. 541	0. 171 0. 135 0. 073	0. 421 0. 423 0. 427	0. 408 0. 442 0. 500
Tie triangles	Nb	Si	Тi	Nb	Si	Тi
TiSi/ NbSi <sub>2</sub> /TiSi <sub>2</sub>	0. 133 0. 110 0. 318	0. 512 0. 651 0. 665	0. 355 0. 239 0. 027	0. 318 0. 133 0. 110	0. 665 0. 512 0. 651	0. 027 0. 355 0. 239
T i <sub>5</sub> Si <sub>4</sub> / T iSi	0. 171 0. 135 0. 073	0. 421 0. 423 0. 427	0. 408 0. 442 0. 500	0. 133 0. 098 0. 055	0. 512 0. 504 0. 495	0. 355 0. 398 0. 450
T i( Nb) / T i <sub>5</sub> Si <sub>3</sub>	0. 078 0. 098 0. 001	0. 028 0. 019 0. 045	0. 875 0. 883 0. 954	0. 031 0. 033 0. 001	0. 357 0. 357 0. 347	0. 612 0. 610 0. 652
TiSi/ NbSi <sub>2</sub> / Nb <sub>5</sub> Si <sub>3</sub>	0. 133 0. 617 0. 318	0. 512 0. 360 0. 665	0. 355 0. 023 0. 027	0. 318 0. 133 0. 617	0. 665 0. 512 0. 360	0. 027 0. 355 0. 023
M <sub>3</sub> Si/Nb <sub>5</sub> Si <sub>3</sub> /Ti <sub>5</sub> Si <sub>3</sub>	0. 0273 0. 617 0. 200	0. 246 0. 360 0. 360	0. 481 0. 023 0. 540	0. 617 0. 200 0. 273	0. 360 0. 360 0. 246	0. 023 0. 540 0. 481
T iSi/ T i <sub>5</sub> Si <sub>4</sub> / Nb <sub>5</sub> Si <sub>3</sub>	0. 133 0. 171 0. 617	0. 512 0. 421 0. 360	0. 355 0. 408 0. 023	0. 171 0. 617 0. 133	0. 421 0. 360 0. 512	0. 408 0. 023 0. 355
( NbT i) / M <sub>3</sub> Si / Nb <sub>5</sub> Si <sub>3</sub>	0. 898 0. 273 0. 617	0. 043 0. 246 0. 360	0. 059 0. 481 0. 023	0. 273 0. 617 0. 898	0. 246 0. 360 0. 043	0. 481 0. 023 0. 059
T i <sub>5</sub> Si <sub>4</sub> / Nb <sub>5</sub> Si <sub>3</sub> / T i <sub>5</sub> Si <sub>3</sub>	0. 171 0. 617 0. 200	0. 421 0. 360 0. 360	0. 408 0. 023 0. 540	0. 617 0. 200 0. 171	0. 360 0. 360 0. 421	0. 023 0. 540 0. 408
(TiNb)/M <sub>3</sub> Si /Ti <sub>5</sub> Si <sub>3</sub>	0. 098 0. 191 0. 041	0. 019 0. 253 0. 357	0. 883 0. 556 0. 602	0. 191 0. 041 0. 098	0. 253 0. 357 0. 019	0. 556 0. 602 0. 883

By analyzing Fig. 2 and Table 1, the following three phase equilibria in the Nb-Tr-Si ternary system at 1473 K can be determined:

- 1)  $NbSi_2 + TiSi_2 + TiSi_3$ ;
- 2)  $NbSi_2 + TiSi + Nb_5Si_3$ ;
- 3) Nb<sub>5</sub>Si<sub>3</sub>+ TiSi+ Ti<sub>5</sub>Si<sub>4</sub>;
- 4)  $Nb_5Si_3 + Ti_5Si_4 + Ti_5Si_3$ ;
- 5)  $Nb_5Si_3 + Ti_5Si_3 + M_3Si_5$ ;
- 6)  $Nb_5Si_3 + (NbTi) + M_3Si;$
- 7)  $Ti_5Si_3 + (NbTi) + M_3Si$ .

Referring to Table 1, the tie lines and tie triangles of equilibrium phases in the concentration triangle can be obtained, thus the isothermal section of the NbrTrSi ternary system of 1473 K is established, as shown in Fig. 3.

### 3. 2 Isothermal section of Nb Ti Si ternary system at 1373 K

Fig. 4(a) shows the back-scattered electron image of NbTrSi diffusion triple after annealing for 480 h at 1 373 K. It can be seen that there form seven diffusion layers, of which five are TrSi compounds, and two are NbSi compounds. The characteristics of phase distribution and phase relations are shown in Fig. 4(b).

The chemical compositions of different equilibrium phases were determined by means of the electron microprobe analysis. Part of the typical data are listed in Table 2. By combining Table 2 with Fig. 4, the

TiSi compounds can be determined to be TiSi<sub>2</sub>, TiSi, Ti<sub>5</sub>Si<sub>4</sub>, Ti<sub>5</sub>Si<sub>3</sub> and Ti<sub>3</sub>Si, among which TiSi<sub>2</sub>, TiSi and Ti<sub>5</sub>Si<sub>4</sub> diffusion layers are obvious and relatively thick, but Ti<sub>5</sub>Si<sub>3</sub> diffusion layer is relatively thin. By local magnification, it can be found that there also exists a very thin diffusion layer, see Fig. 5. Through the electron microprobe analysis, this thin layer is identified to be Ti<sub>3</sub>Si. The Nb-Si compounds are divided into two layers, namely Nb<sub>5</sub>Si<sub>3</sub> and NbSi<sub>2</sub>.

The compositions of silicon in TrSi compounds are in a narrow range, which agrees with those reported in references. But the solubility of niobium in

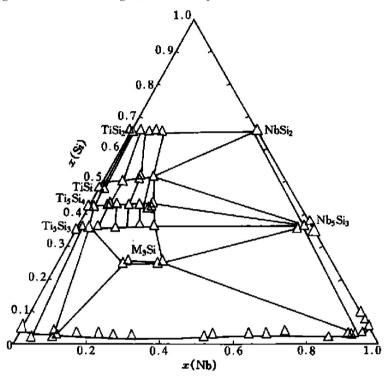


Fig. 3 Isothermal section of Nb-TrSi ternary system at 1473 K

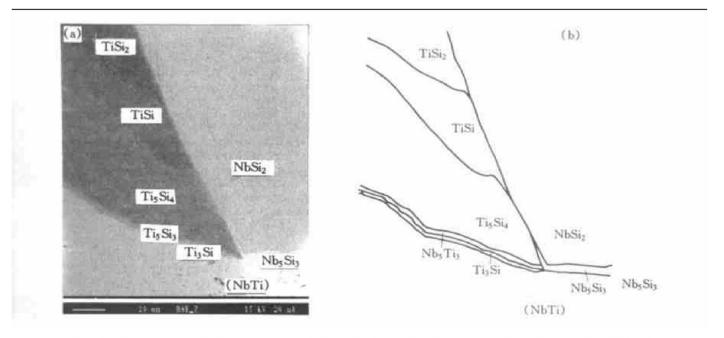


Fig. 4 Back-scattered electron image (a) and schematic diagram of phase distribution (b) of Nb-Ti-Si diffusion triple after annealing for 480 h at 1373 K

**Table 2** Tie lines and tie triangle of Nb Tr Si ternary triple at 1.373 K (mole fraction)

ternary triple at 1 373 K (mole fraction)						
Tie lines	Nb	Si	Тi	Nb	Si	Ti
Nb <sub>5</sub> Si <sub>3</sub> /NbSi <sub>2</sub>	0. 604 0. 620 0. 635	0. 360 0. 358 0. 363	0. 036 0. 022 0. 001	0. 351 0. 362 0. 347	0. 648 0. 638 0. 652	0. 001 0 0. 001
Nb <sub>5</sub> Si <sub>3</sub> / Nb( T i)	0. 604 0. 630 0. 632	0. 360 0. 358 0. 367	0. 036 0. 012 0. 001	0. 964 0. 996 0. 998	0. 021 0. 002 0	0. 015 0. 002 0. 002
TiSi /TiSi <sub>2</sub>	0. 117 0. 108 0. 085	0. 481 0. 490 0. 480	0. 402 0. 402 0. 435	0. 105 0. 081 0. 062	0. 656 0. 646 0. 655	0. 239 0. 273 0. 283
TiSi <sub>2</sub> /NbSi <sub>2</sub>	0. 105 0. 081 0. 062	0. 656 0. 646 0. 655	0. 239 0. 273 0. 283	0.311 0.308 0.315	0. 661 0. 668 0. 665	0. 028 0. 024 0. 020
T i <sub>5</sub> Si <sub>3</sub> / T i <sub>5</sub> Si <sub>4</sub>	0. 155 0. 103 0. 073	0. 364 0. 361 0. 366	0. 481 0. 536 0. 561	0. 132 0. 107 0. 063	0. 431 0. 430 0. 423	0. 437 0. 463 0. 514
Tie triangles	Nb	Si	Τi	Nb	Si	Ti
Ti <sub>5</sub> Si <sub>4</sub> /TiSi	0. 132 0. 097 0. 063	0. 431 0. 440 0. 423	0. 437 0. 463 0. 514	0. 117 0. 108 0. 085	0. 481 0. 490 0. 480	0. 402 0. 402 0. 435
TiSi/ NbSi <sub>2</sub> / TiSi <sub>2</sub>	0. 117 0. 311 0. 105	0. 481 0. 661 0. 656	0. 402 0. 028 0. 239	0. 311 0. 105 0. 117	0. 661 0. 656 0. 481	0. 028 0. 239 0. 402
Ti(Nb) /Ti <sub>3</sub> Si	0. 980 0. 322 0. 127	0. 004 0. 008 0. 011	0. 016 0. 670 0. 862	0. 145 0. 129 0. 068	0. 258 0. 253 0. 255	0. 597 0. 618 0. 677
TiSi/ NbSi <sub>2</sub> / Ti <sub>5</sub> Si <sub>4</sub>	0. 117 0. 311 0. 132	0. 481 0. 661 0. 431	0. 402 0. 028 0. 437	0. 311 0. 132 0. 117	0. 661 0. 431 0. 481	0. 028 0. 437 0. 402
T i <sub>3</sub> Si/ Nb <sub>5</sub> Si <sub>3</sub> / T i <sub>5</sub> Si <sub>3</sub>	0. 145 0. 604 0. 155	0. 258 0. 360 0. 364	0. 597 0. 036 0. 481	0. 604 0. 155 0. 145	0. 360 0. 364 0. 258	0. 036 0. 481 0. 597
T i <sub>5</sub> Si <sub>4</sub> / NbSi <sub>2</sub> / Nb <sub>5</sub> Si <sub>3</sub>	0. 132 0. 311 0. 604	0. 431 0. 661 0. 360	0. 437 0. 028 0. 036	0. 311 0. 604 0. 132	0. 661 0. 360 0. 431	0. 028 0. 036 0. 437
(TiNb)/Ti <sub>3</sub> Si /Nb <sub>5</sub> Si <sub>3</sub>	0. 980 0. 145 0. 604	0. 004 0. 258 0. 360	0. 016 0. 597 0. 036	0. 145 0. 604 0. 980	0. 258 0. 360 0. 004	0. 597 0. 036 0. 016
T i <sub>5</sub> Si <sub>4</sub> / Nb <sub>5</sub> Si <sub>3</sub> / T i <sub>5</sub> Si <sub>3</sub>	0. 132 0. 604 0. 155	0. 431 0. 360 0. 364	0. 437 0. 036 0. 481	0. 604 0. 155 0. 132	0. 360 0. 364 0. 431	0. 036 0. 481 0. 437

 $T\,\dot{r}\,Si$  compounds at  $1\,373\,K$  is layer; it is  $10.\,5\%$  in  $T\,iSi_2,\ 11.\,7\%$  in  $T\,iSi,\ 13.\,2\%$  in  $T\,i_5Si_4,\ 15.\,5\%$  in  $T\,i_5Si_3$  and  $14.\,5\%$  in  $T\,i_3Si$  (mole fraction), respectively.

The solubility of titanium in Nb-Si compounds is relatively small. The largest value in NbSi<sub>2</sub> is 2.77% (mole fraction) and that in Nb<sub>5</sub>Si<sub>3</sub> is 3.6% (mole fraction). The solubility of silicon in (NbTi) is very small.

There can form continuous solid solution between niobium and titanium at 1373 K, but the Nb/Ti interface can be clearly seen in the diffusion couple.

It can be known from Fig. 4 and Table 2 that there exist the following three phase equilibria in the Nb-Tr-Si ternary system at 1373 K:

- 1) NbSi<sub>2</sub>+ TiSi<sub>2</sub>+ TiSi;
- 2)  $NbSi_2 + TiSi_2 + Ti_5Si_4$ ;

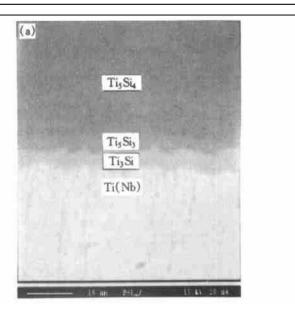


Fig. 5 Local magnification of Ti<sub>3</sub>Si thin layer

- 3) NbSi<sub>2</sub>+ Ti<sub>5</sub>Si<sub>4</sub>+ Nb<sub>5</sub>Si<sub>3</sub>;
- 4) Nb<sub>5</sub>Si<sub>3</sub>+ Ti<sub>5</sub>Si<sub>4</sub>+ Ti<sub>5</sub>Si<sub>3</sub>;
- 5) Nb<sub>5</sub>Si<sub>3</sub>+ Ti<sub>5</sub>Si<sub>3</sub>+ Ti<sub>3</sub>Si;
- 6)  $Nb_5Si_3 + Ti_3Si + (NbTi)$ .

Referring to Fig. 4 and Table 2, the tie lines and tie triangles of the equilibrium phases in the concentration triangle can be obtained, thus the isothermal section of the Nb-Ti-Si ternary system at 1373 K is established, as shown in Fig. 6.

# 3. 3 Phase transformations and phase relations from 1473 K to 1373 K in Nb-Ti-Si ternary system

The three-phase equilibria of the Nb-Ti-Si ternary system at 1373 K and 1473 K are summarized in Table 3. It is clear from Table 3 and the isothermal sections in Fig. 3 and Fig. 5 that there are two main changes from 1473 K to 1373 K. 1) There occurs a peritectoid-eutectoid reaction: Nb<sub>5</sub>Si<sub>3</sub>+ TiSi → Ti<sub>5</sub>Si<sub>4</sub>+ NbSi<sub>2</sub>; 2) With decreasing temperature, the Ti<sub>5</sub>Si<sub>3</sub>+ M<sub>3</sub>Si+ (NbTi) phase field moves to the Ti-Si side and disappears at some temperature between 1473 K and 1373 K, and correspondingly M<sub>3</sub>Si compound changes to Ti<sub>3</sub>Si compound.

Table 3 Three phase equilibria in Nb-TrSi ternary system at 1473 K and 1373 K

Equilibria

T/K

		1)	$1.151_2 + 1.151_2 + 1.151_1$	5)	$Nb_5S_{13} + T_{15}S_{13} + M_3S_1$
1 472	1 472	2)	NbSi <sub>2</sub> + TiSi+ Nb <sub>5</sub> Si <sub>3</sub>	6)	Ti <sub>5</sub> Si <sub>3</sub> + M <sub>3</sub> Si+ (NbTi)
	14/3	3)	Nb <sub>5</sub> Si <sub>3</sub> + TiSi+ Ti <sub>5</sub> Si <sub>4</sub>	7)	T i <sub>5</sub> Si <sub>3</sub> + M <sub>3</sub> Si+ (NbTi) Nb <sub>5</sub> Si <sub>3</sub> + M <sub>3</sub> Si+ (NbTi)
			Nb <sub>5</sub> Si <sub>3</sub> + Ti <sub>5</sub> Si <sub>4</sub> + Ti <sub>5</sub> Si <sub>4</sub>		
		1)	TiSi. I NhSi. I TiSi	4)	Nh-Si-   Ti-Si-   Ti-Si-

1 373 2) NbSi<sub>2</sub>+ TiSi+ Ti<sub>5</sub>Si<sub>4</sub> 5) Nb<sub>5</sub>Si<sub>3</sub>+ Ti<sub>5</sub>Si<sub>3</sub>+ Ti<sub>3</sub>Si 3) Nb<sub>5</sub>Si<sub>3</sub>+ Ti<sub>5</sub>Si<sub>4</sub>+ NbSi<sub>2</sub> 6) Nb<sub>5</sub>Si<sub>3</sub>+ Ti<sub>3</sub>Si+ (NbTi)

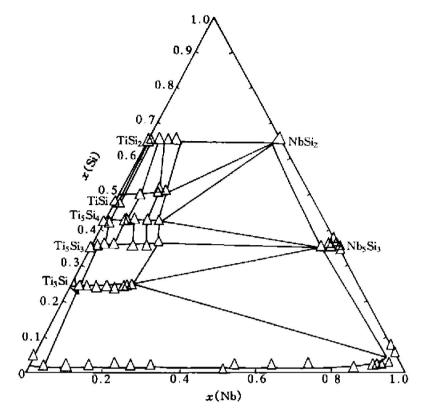


Fig. 6 Isothermal section of Nb-TrSi ternary system at 1373 K

### 4 CONCLUSIONS

- 1) No ternary compounds form in the Nb-Tr-Si ternary system at 1473 K and 1373 K.
- 2) There occurs a peritectoid eutectoid reaction:  $Nb_5Si_3 + TiSi \longrightarrow Ti_5Si_4 + NbSi_2$  in the Nb TiSi ternary system on cooling from 1473 K and 1373 K.
- 3) With decreasing temperature, the  $Ti_5Si_3 + M_3Si + (NbTi)$  phase field moves to the  $T\dot{r}Si$  side and disappears at some temperature between 1473 K and 1373 K, and correspondingly  $M_3Si$  compound changes to  $Ti_3Si$  compound.

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