

# Calculation of activity coefficients for components in ternary Ti alloys and intermetallics as matrix of composites<sup>①</sup>

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**Abstract:** Based on Kohler's ternary solution model and Miedema's model for calculating the formation heat of binary solution, the integral equation was established for calculating the activity coefficients in ternary alloys and intermetallics. The activity coefficients for components in alloy Ti5Al2.5Sn, Ti6Al4V and intermetallics TiAl, Ti<sub>3</sub>Al and Ti<sub>2</sub>AlNb were calculated with the equations. The calculated data coincide well with the experimental ones found in literatures. According to the calculated activity coefficients and activities, it can be predicted that the interfacial reaction in SiC/Ti<sub>3</sub>Al composite is more severe than that in composites SiC/Ti<sub>2</sub>AlNb and SiC/TiAl.

**Key words:** activity coefficient; thermodynamics; Ti alloy; intermetallics; interfacial reaction

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## 1 INTRODUCTION

Activity and activity coefficient can be directly used to calculate the evaporation loss of alloying elements during melting<sup>[1, 2]</sup> and to calculate thermodynamic properties in studying interfacial reaction mechanism of metal-matrix composites<sup>[3]</sup>. Because there are some difficulties to measure the activity coefficient experimentally<sup>[4, 5]</sup>, it is necessary to get it by calculation. Some research works have been done to predict or calculate activities and activity coefficients in systems of a number of components<sup>[6]</sup>. However, up to now, for the solution systems consisting of elements more than 3, it is still difficult to get those activity coefficients precisely. Although two main ternary solution models and formulas have been set up, the calculated data from one set formulas do not coincide well with the experimental ones found in Ref. [7] and there are some mistakes in another set formulas<sup>[8]</sup>. Besides, although thermocalc software can be used in calculating activity coefficients in ternary alloys, the price of the data base is expensive and the number of elements in the data base are limited.

In this paper, based on Kohler's ternary solution model and Miedema's model of calculating formation heat,  $\Delta H_{ij}$ , of binary system, the integral equation was established for calculating the activity coefficients

in ternary alloys and intermetallics with thermodynamic methods. The activity coefficients for components in ternary alloy Ti5Al2.5Sn, Ti6Al4V and intermetallics TiAl, Ti<sub>3</sub>Al and Ti<sub>2</sub>AlNb were obtained with the equation, and compared with experimental data in literatures. The calculated results agree with experimental values although there are a little flaws in both of the starting models. Interfacial reaction tendency of SiC/TiAl, SiC/Ti<sub>3</sub>Al and SiC/Ti<sub>2</sub>AlNb composites was predicted according to the activity coefficients.

## 2 CALCULATION MODEL

For  $i-j-k$  ternary system, under the condition of certain temperature and pressure, the activity coefficient of solute  $i$  would be expressed as<sup>[9]</sup>:

$$\ln \gamma_i = \ln \gamma_i^0 + \sum_{j=2}^3 \xi_j x_j + \sum_{j=2}^3 \varphi_j x_j^2 + \sum_{i=2}^3 \sum_{j=2}^3 \varphi_{ij} x_i x_j + o(x^3) \quad (1)$$

The definition of each coefficient is as follows:

$$\ln \gamma_i^0 = \frac{1}{RT} (\bar{G}_i^E)_{x_j=0, x_k \rightarrow 1} \quad (2)$$

$$\xi_i = \left[ \frac{\partial \ln \gamma_i}{\partial x_i} \right]_{x_j=0, x_k \rightarrow 1} = \frac{1}{RT} \left[ \frac{\partial \bar{G}_i^E}{\partial x_i} \right]_{x_j=0, x_k \rightarrow 1} \quad (3)$$

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$$\dot{\xi}_i = \left[ \frac{\partial \ln \gamma_i}{\partial x_j} \right]_{x_k \rightarrow 1} = \frac{1}{RT} \left[ \frac{\partial \bar{G}_i^E}{\partial x_j} \right]_{x_k \rightarrow 1} \quad (4)$$

$$\dot{\beta}_i = \frac{1}{2} \left[ \frac{\partial^2 \ln \gamma_i}{\partial x_i^2} \right]_{x_j=0, x_k \rightarrow 1} = \frac{1}{2RT} \left[ \frac{\partial^2 \bar{G}_i^E}{\partial x_i^2} \right]_{x_j=0, x_k \rightarrow 1} \quad (5)$$

$$\ddot{\beta}_i = \frac{1}{2} \left[ \frac{\partial^2 \ln \gamma_i}{\partial x_j^2} \right]_{x_k \rightarrow 1} = \frac{1}{2RT} \left[ \frac{\partial^2 \bar{G}_i^E}{\partial x_j^2} \right]_{x_k \rightarrow 1} \quad (6)$$

$$\ddot{\beta}_i = \frac{1}{RT} \left[ \frac{\partial^2 \bar{G}_i^E}{\partial x_i \partial x_j} \right]_{x_k \rightarrow 1} \quad (7)$$

In the ternary system, there exist the equations as:

$$\begin{cases} \bar{G}_i^E = G^E - x_j \frac{\partial G^E}{\partial x_j} + (1-x_i) \frac{\partial G^E}{\partial x_i} \\ \bar{G}_j^E = G^E - x_i \frac{\partial G^E}{\partial x_i} + (1-x_j) \frac{\partial G^E}{\partial x_j} \\ \bar{G}_k^E = G^E - x_i \frac{\partial G^E}{\partial x_i} - x_j \frac{\partial G^E}{\partial x_j} \end{cases} \quad (8)$$

where  $x_i$ ,  $x_j$  and  $x_k$  are the mole fractions of components  $i$ ,  $j$  and  $k$ ;  $\bar{G}_i^E$ ,  $\bar{G}_j^E$ ,  $\bar{G}_k^E$  are the partial mole excess free energy of components  $i$ ,  $j$ ,  $k$ ;  $\dot{\xi}_i$ ,  $\dot{\xi}_j$  are the first-order international coefficients between  $i$  and  $i$ ,  $j$ ;  $\dot{\beta}_i$ ,  $\dot{\beta}_j$ ,  $\dot{\beta}_k$  are the second-order international coefficients between  $i$  and  $i$ ,  $j$ ;  $G^E$  is the excess free energy of ternary system;  $\gamma_i^0$  is the activity coefficient of component  $i$ ;  $\gamma_i^0$  is the zero-order activity coefficient of component  $i$ . From Eqns. (1)–(8), it is obvious that, if the relationship between  $G^E$  and  $x_i$ ,  $x_j$  and  $x_k$  can be built up, the activity coefficient of component  $i$  can be calculated. The following Kohler's equation just serves the purpose:

$$\begin{aligned} G^E = & (x_i + x_j)^2 G_{ij}^E \left\{ \frac{x_i}{1-x_i-x_j}, \frac{x_j}{1-x_i-x_j} \right\} + \\ & (1-x_j)^2 G_{ik}^E \left\{ \frac{x_i}{1-x_j}, \frac{1-x_i-x_j}{1-x_j} \right\} + \\ & (1-x_i)^2 G_{jk}^E \left\{ \frac{x_j}{1-x_i}, \frac{1-x_i-x_j}{1-x_i} \right\} \end{aligned} \quad (9)$$

$$\begin{aligned} G_{ij}^E = & H_{ij}^E - TS_{ij}^E = \Delta H_{ij} - TS_{ij}^E \\ G_{ik}^E = & H_{ik}^E - TS_{ik}^E = \Delta H_{ik} - TS_{ik}^E \\ G_{jk}^E = & H_{jk}^E - TS_{jk}^E = \Delta H_{jk} - TS_{jk}^E \end{aligned} \quad (10)$$

where  $G_{ij}^E$ ,  $G_{ik}^E$ ,  $G_{jk}^E$  are the excess free energies of these binary systems ( $i$ ,  $j$ ), ( $j$ ,  $k$ ) and ( $i$ ,  $k$ ).  $\Delta H_{ij}$ ,  $\Delta H_{ik}$ ,  $\Delta H_{jk}$  are the formation heats during the alloying process between component  $i$  and  $j$ ,  $i$  and  $k$ ,  $j$  and  $k$ , respectively;  $S_{ij}^E$ ,  $S_{ik}^E$ ,  $S_{jk}^E$  are the excess entropies of component  $i$  and  $j$ ,  $i$  and  $k$ ,  $j$  and  $k$ , respectively.

$$S_{ij}^E = 0.1 \times \Delta H_{ij} \left[ \frac{T_{mi} + T_{mj}}{T_{mi} T_{mj}} \right] \quad (11)$$

$$G_{ij}^E = \alpha_{ij} \times \Delta H_{ij} \quad (12)$$

where  $\alpha_{ij} = 1 - 0.1T \left[ \frac{T_{mi} + T_{mj}}{T_{mi} T_{mj}} \right]$ ,  $T_{mi}$  and  $T_{mj}$

are the melting points of the components  $i$  and  $j$ , respectively. According to the Miedema's Model<sup>[10]</sup>,  $\Delta H_{ij}$  could be written as:

$$\Delta H_{ij} = \frac{f_{ij} x_i [1 + \mu_i x_j (\phi_i - \phi_j)]}{M_{ij}} \cdot \frac{x_j [1 + \mu_j x_i (\phi_j - \phi_i)]}{M_{ij}} \quad (13)$$

where

$$\begin{aligned} f_{ij} = & 2p V_i^{2/3} V_j^{2/3} \times \left[ \frac{9.4 [(n_{ws}^{1/3})_j - (n_{ws}^{1/3})_i]^2}{(n_{ws}^{1/3})_i^{-1} + (n_{ws}^{1/3})_j^{-1}} - \right. \\ & \left. \frac{(\phi_i - \phi_j)^2 + a(r/p)}{(n_{ws}^{1/3})_i^{-1} + (n_{ws}^{1/3})_j^{-1}} \right] \\ M_{ij} = & x_i V_i^{2/3} [1 + \mu_i x_j (\phi_i - \phi_j)] + \\ & x_j V_j^{2/3} [1 + \mu_j x_i (\phi_j - \phi_i)] \end{aligned}$$

where  $V_i$ ,  $V_j$ ,  $V_k$  are the mole volume of  $i$ ,  $j$  and  $k$ ;  $(n_{ws})_i$  the electron densities of  $i$  and  $j$ ;  $\phi_i$ ,  $\phi_j$ ,  $\phi_k$  the electron-negativity of  $i$ ,  $j$  and  $k$ ;  $p$ ,  $r$ ,  $\mu$  the experimental constants. Substituting Eqns. (12), (13) into Eqns. (2–8), and making  $F_{ij} = \alpha_{ij} \times f_{ij}$ , we can get the following expressions.

$$\ln \gamma_i^0 = \frac{1}{RT} A \quad (14)$$

$$\ln \gamma_j^0 = \frac{1}{RT} D \quad (15)$$

$$\dot{\xi}_i = \frac{1}{RT} (2B - 2C) \quad (16)$$

$$\dot{\xi}_j = \frac{1}{RT} (2B' - 2C') \quad (17)$$

$$\dot{\beta}_i = \dot{\xi}_i = \frac{1}{RT} (-A - D) \quad (18)$$

$$\dot{\beta}_j = \frac{1}{2RT} (-E + J) \quad (19)$$

$$\dot{\beta}_k = \frac{1}{2RT} (-F + J') \quad (20)$$

$$\dot{\beta}_i = \frac{1}{2RT} (-F) \quad (21)$$

$$\dot{\beta}_j = \frac{1}{2RT} (-F') \quad (22)$$

$$\dot{\beta}_k = \dot{\beta}_j = \frac{1}{RT} (A + D) \quad (23)$$

$$A = F_{ik} [1 + \mu_i (\phi_i - \phi_k)] / V_k^{2/3} \quad (24)$$

$$B = F_{ik} [-1 - (2\mu_i + \mu_k)(\phi_i - \phi_k) - \mu_i \mu_k (\phi_i - \phi_k)^2] / V_k^{2/3} \quad (25)$$

$$B' = F_{jk} [-1 - (2\mu_j + \mu_k)(\phi_j - \phi_k) - \mu_j \mu_k (\phi_j - \phi_k)^2] / V_k^{2/3} \quad (26)$$

$$\begin{aligned} C = & F_{ik} [1 + \mu_i (\phi_i - \phi_k)] \cdot \\ & \{ V_i^{2/3} [1 + \mu_i (\phi_i - \phi_k)] + \\ & V_k^{2/3} [-1 + \mu_k (\phi_k - \phi_i)] \} / (V_k^{2/3})^2 \end{aligned} \quad (27)$$

$$\begin{aligned} C' = & F_{jk} [1 + \mu_j (\phi_j - \phi_k)] \cdot \\ & \{ V_j^{2/3} [1 + \mu_j (\phi_j - \phi_k)] + \\ & V_k^{2/3} [-1 + \mu_k (\phi_k - \phi_j)] \} / (V_k^{2/3})^2 \end{aligned} \quad (28)$$

$$D = F_{jk} [1 + \mu_j (\phi_j - \phi_k)] / V_k^{2/3} \quad (29)$$

$$\begin{aligned} E = & F_{ik} [(4\mu_i + 2\mu_k)(\phi_k - \phi_i)] - \\ & 2 - 2\mu_i \mu_k (\phi_k - \phi_i)^2 / V_k^{2/3} - \\ & 2 [1 + \mu_i (\phi_i - \phi_k)] / V_i^{2/3} (1 + \end{aligned}$$

$$\mu_i(\phi_i - \phi_k) + V_k^{2/3}(-1 + \mu_k(\phi_k - \phi_i))/V_k^{2/3})^2 \quad (30)$$

$$F = F_{jk}\{[(4\mu_j + 2\mu_k)(\phi_k - \phi_j)] - 2 - 2\mu_j\mu_k(\phi_k - \phi_j)^2]/V_k^{2/3} - 2[1 + \mu_j(\phi_j - \phi_k)]/V_j^{2/3}(1 + \mu_j(\phi_j - \phi_k)) + V_k^{2/3}(-1 + \mu_k(\phi_k - \phi_j))/V_k^{2/3})^2 \quad (31)$$

$$F' = F_{ik}\{[(4\mu_i + 2\mu_k)(\phi_k - \phi_i)] - 2 - 2\mu_i\mu_k(\phi_k - \phi_i)^2]/V_k^{2/3} - 2[1 + \mu_i(\phi_i - \phi_k)]/V_i^{2/3}(1 + \mu_i(\phi_i - \phi_k)) + V_k^{2/3}(-1 + \mu_k(\phi_k - \phi_i))/V_k^{2/3})^2 \quad (32)$$

$$J = F_{ik}\{6\mu_i(\phi_i - \phi_k)/V_k^{2/3} + 6\mu_i\mu_k(\phi_i - \phi_k)^2/V_k^{2/3} + 6\mu_i(\phi_i - \phi_k)Z/V_k^{2/3})^2 + 6[1 + \mu_i(\phi_i - \phi_k)]\mu_k(\phi_i - \phi_k)/V_k^{2/3} + 6[1 + \mu_i(\phi_i - \phi_k)]Z/V_k^{2/3})^2 + 6[1 + \mu_i(\phi_i - \phi_k)]\mu_k(\phi_i - \phi_k)Z/V_k^{2/3})^2 - 3[1 + \mu_i(\phi_i - \phi_k)]Y/V_k^{2/3})^2 + 6[1 + \mu_i(\phi_i - \phi_k)]Z^2/V_k^{2/3})^3 \quad (33)$$

$$J' = F_{jk}\{6\mu_j(\phi_j - \phi_k)/V_k^{2/3} + 6\mu_j\mu_k(\phi_j - \phi_k)^2/V_k^{2/3} + 6\mu_j(\phi_j - \phi_k)Z'/V_k^{2/3})^2 + 6[1 + \mu_j(\phi_j - \phi_k)]\mu_k(\phi_j - \phi_k)/V_k^{2/3} + 6[1 + \mu_j(\phi_j - \phi_k)]Z'/V_k^{2/3})^2 + 6[1 + \mu_j(\phi_j - \phi_k)]\mu_k(\phi_j - \phi_k)Z'/V_k^{2/3})^2 - 3[1 + \mu_j(\phi_j - \phi_k)]Y'/V_k^{2/3})^2 + 6[1 + \mu_j(\phi_j - \phi_k)]Z'^2/V_k^{2/3})^3 \quad (34)$$

$$Z = V_i^{2/3}[1 + \mu_i(\phi_i - \phi_k)] + V_k^{2/3}[-1 + \mu_k(\phi_k - \phi_i)] \quad (35)$$

$$Y = V_i^{2/3}[-2\mu_i(\phi_i - \phi_k)] + V_k^{2/3}[-2\mu_k(\phi_k - \phi_i)] \quad (36)$$

$$Z' = V_j^{2/3}[1 + \mu_j(\phi_j - \phi_k)] + V_k^{2/3}[-1 + \mu_k(\phi_k - \phi_j)] \quad (37)$$

$$Y' = V_j^{2/3}[-2\mu_j(\phi_j - \phi_k)] + V_k^{2/3}[-2\mu_k(\phi_k - \phi_j)] \quad (38)$$

In order to obtain the activity of each component in intermetallics with higher solute concentration, the following Eqn. [11] can be used to calculate the activity coefficient of solvent:

$$\ln \gamma_k = \left| \frac{\xi x_i^2 + \xi x_{ixj} + \xi x_j^2}{(x_i + x_j)^2} \right| (x_i + x_j) + \ln x_k + 2 \left[ \ln x_k + (x_i + x_j) + \frac{(x_i + x_j)^2}{2} \right] \cdot \left| \frac{x_i^3 \bar{\rho} + x_i x_j^2 \bar{\rho}}{(x_i + x_j)^3} + \frac{x_i^2 x_j \bar{\rho} + x_j x_i^2 \bar{\rho} + x_i^2 x_j \bar{\rho} + x_j^2 x_i \bar{\rho}}{(x_i + x_j)^3} \right| \quad (39)$$

The activity coefficients of solutes  $i$  and  $j$  are calculated according to the following formulae which were derived from Eqn. (1):

$$\ln \frac{\gamma_i}{\gamma_i^0} = \ln \gamma_k + \xi x_i + \xi x_j + \rho x_i^2 + \rho x_j^2 + \rho x_i x_j \quad (40)$$

$$\ln \frac{\gamma_j}{\gamma_j^0} = \ln \gamma_k + \xi x_i + \xi x_j + \rho x_i^2 + \rho x_j^2 + \rho x_i x_j \quad (41)$$

### 3 RESULTS AND DISCUSSION

Using Eqns. (39), (40) and (41), the activity coefficients,  $\gamma_i$  of Ti5Al2.5Sn, Ti6Al4V alloy and Ti<sub>3</sub>Al, TiAl intermetallics are calculated and compared with experimental data. For calculation, the values of electron densities  $\phi$ , electron negativity  $n_{ws}$ , mole volumes  $V$  and some parameters ( $r/p$ ,  $\mu$ ,  $q/p$  and  $P$ ) come from Ref. [12]. Due to the lower solute concentration in Ti5Al2.5Sn alloy, it is not necessary to consider the revision term  $\ln \gamma_k$  in Eqns. (40) and (41). The activity coefficients of components in Ti5Al2.5Sn melt at 2100 K and Ti6Al4V melt at 2100 K are computed with empirical constant  $a = 0.73$ <sup>[12]</sup>, as listed in Table 1 together with the measured ones<sup>[13]</sup>. It can be found that the calculated and the measured are in close agreement with each other. For Ti<sub>3</sub>Al and TiAl in solid state, if  $a = 1$  as reported in Refs. [7, 12], a large deviation can be found between the calculated and the measured data. However, by the way of try and error, if  $a = 0.32$ , as shown in Table 1, the calculated activity coefficients of Ti in Ti<sub>3</sub>Al and TiAl are in good accordance with the measured ones<sup>[14]</sup>. The difference of activity coefficient between melt and solid depends on empirical constant  $a$ , which in some system may be larger than that in others by calculating with different  $a$ .

**Table 1** Comparison of calculated activity coefficients with measured ones

Alloy	Elements	Calculated	Measured
Ti5Al2.5Sn (2100 K, $a = 0.73$ )	Al	0.014	0.016
	Sn	0.052	0.066
Ti6Al4V (2100 K, $a = 0.73$ )	Al	0.1179	0.011
	V	0.8913	0.713
TiAl (1000 K, $a = 0.32$ )	Ti	0.1769	0.157
	Ti <sub>3</sub> Al (1000 K, $a = 0.32$ )	0.6353	0.716

The activity coefficients of intermetallics Ti<sub>3</sub>Al, TiAl and Ti<sub>2</sub>AlNb at different temperatures are then calculated as shown in Figs. 1, 2 and 3, respectively. The results show that the activity coefficients of Ti and Al in the three TiAl intermetallics are less than 1. The activity coefficients of Ti in both TiAl and Ti<sub>2</sub>AlNb are much smaller than those in Ti<sub>3</sub>Al, indicating that the atomic binding energy in both TiAl and Ti<sub>2</sub>AlNb is larger. So it can be forecasted that

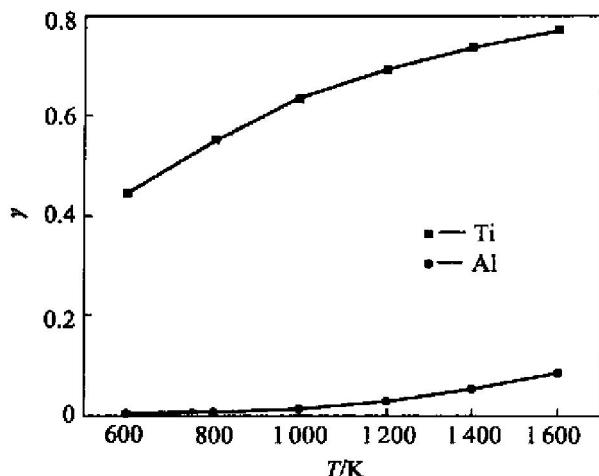


Fig. 1 Activity coefficient of elements in  $\text{Ti}_3\text{Al}$  at different temperatures

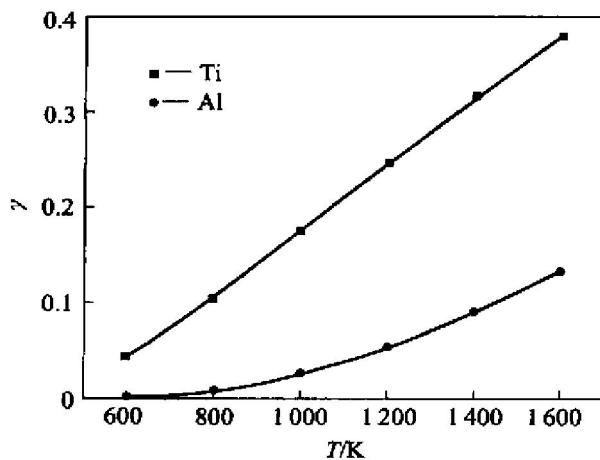


Fig. 2 Activity coefficient of elements in  $\text{TiAl}$  at different temperatures

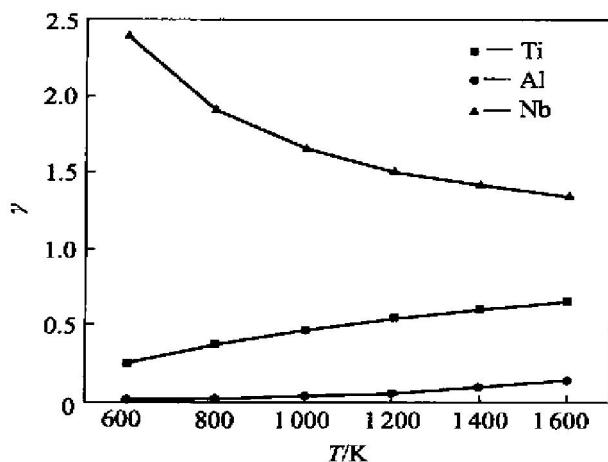


Fig. 3 Activity coefficient of elements in  $\text{Ti}_2\text{AlNb}$  at different temperatures

Ti atoms in  $\text{Ti}_3\text{Al}$  are more active.

For a chemical reaction, according to thermodynamic criterion,  $\Delta_r G = \Delta_r G^\ominus + RT \ln J_a$ , in which  $\Delta_r G^\ominus$  is determined,  $\Delta_r G$  depends on  $J_a$  which is in inverse proportion with activities, namely the larg-

er the activities are, the more negative  $\Delta_r G$  are, and the easier the reaction occurs. Based on some experiments on the interfacial reaction of  $\text{SiC}/\text{Ti}_3\text{Al}$  matrix composites<sup>[14-16]</sup>, there exist some binary products of Ti and C and of Ti and Si and ternary products of Ti, C and Al at the interface. Considering the influence of Ti and Al in matrix, according to  $\gamma_i$  values calculated in this paper and the formula  $a_i = \gamma_i x_i$ , the activities of Ti and Al in  $\text{Ti}_3\text{Al}$ ,  $\text{TiAl}$  and  $\text{Ti}_2\text{AlNb}$  are  $a(\text{Ti}) = 0.4765$ ,  $a(\text{Al}) = 0.002844$ ;  $a(\text{Ti}) = 0.08845$ ,  $a(\text{Al}) = 0.01357$ ;  $a(\text{Ti}) = 0.2318$ ,  $a(\text{Al}) = 0.00707$ , respectively, at 1000 K. One can find that the activities of Ti in  $\text{Ti}_3\text{Al}$ ,  $\text{Ti}_2\text{AlNb}$  and  $\text{TiAl}$  decrease in turn. Therefore, it can be predicted that the interfacial reaction in  $\text{SiC}/\text{Ti}_3\text{Al}$  composite is more severe than that in composites  $\text{SiC}/\text{Ti}_2\text{AlNb}$  and  $\text{SiC}/\text{TiAl}$ , in other words, the thickness of interfacial reaction products in composites  $\text{SiC}/\text{Ti}_2\text{AlNb}$  and  $\text{SiC}/\text{TiAl}$  should be thinner than that in  $\text{SiC}/\text{Ti}_3\text{Al}$ . Fig. 4 shows the thickness of the interfacial reaction zone in SCS-6  $\text{SiC}/\text{Ti}_3\text{Al}$  and in SCS-6  $\text{SiC}/\text{Ti}_2\text{AlNb}$  exposed at 800 °C, which was measured with transmission electron microscope by one of the present authors<sup>[15, 16]</sup>. It is in good accordance with the predictions. The above results mean that the thermodynamic prediction may be helpful for choosing the matrix alloys of the composites.

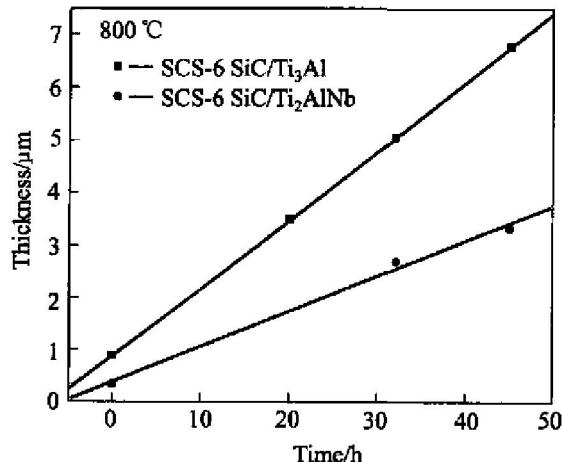


Fig. 4 Thickness of interfacial reaction zone in SCS-6  $\text{SiC}/\text{Ti}_3\text{Al}$  and SCS-6  $\text{SiC}/\text{Ti}_2\text{AlNb}$

#### 4 CONCLUSIONS

1) The formulae of calculating activity coefficient of ternary alloy and intermetallic have been developed, which are suitable to calculate activity coefficient of binary, ternary alloys and intermetallics. The empirical constants ( $a$ ) of 0.73 for ternary melt and 0.32 for solid ternary system were determined.

2) The calculated activity coefficients of Ti and Al in  $\text{Ti}_3\text{Al}$ ,  $\text{TiAl}$  and  $\text{Ti}_2\text{AlNb}$  are less than 1, indicating a negative deviation from Raoult law and at-

tractive force between atoms.

3) By comparing the calculated activities in  $Ti_3Al$ ,  $TiAl$  and  $Ti_2AlNb$ , it can be predicted that the interfacial reactions in  $SiC/Ti_3Al$  composite at about 1 000 K are more severe than those in  $SiC/Ti_2AlNb$  and  $SiC/TiAl$ , namely the thickness of interfacial reaction layer in  $SiC/Ti_2AlNb$  and  $SiC/TiAl$  composites is thinner than those in  $SiC/Ti_3Al$  composite.

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